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Propellant Nonlinear Constitutive Theory Extension: Preliminary Results

August 1983

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FOREWORD

This report was submitted by United Technologies Corporation/Chemical Systems Division, 1050 E. Arques Avenue, Sunnyvale CA 94086 under Contract F04611-80-C-0052, Job Order No. 2307MIEB with the Air Force Rocket Propulsion Laboratory, Edwards AFB CA 93523. This Special Technical Report is approved for release and pulication in accordance with the distribution statement on the cover and in the DD Form 1473.

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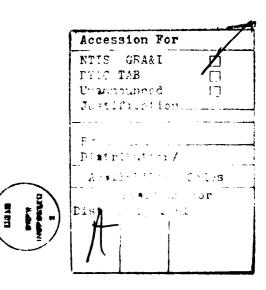
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Uniaxial r	nonisothermal	linear	damage
Biaxial c	constitutive	fracture	healing
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This rem	ort details the tech	nical effort put	forth by CSD and its subcon-
			gram for propellant nonlinear
			Phase I preliminary study
	Outplan theory was a		** ************************************

tractors during the first three phases of this program for propellant nonlinear constitutive theory extension. This includes the Phase I preliminary study in which the Quinlan theory was critiqued, alternate approaches were studied and detailed research planning accomplished. Also included are the detailed experimental evaluations of propellant during Phases II and III, the uniaxial/isothermal investigation and the two-dimensional variable temperature investigation. Detailed subcontractor theoretical development and predictions

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are presented.

This special technical report on Contract No. F04611-80-C-0052 consists of a summary of the Phase II and III laboratory propellant evaluation for UTP-3001 and UTP-19,360B propellants. The detailed experimental results were distributed in Data Packages A through G to all project personnel. Additional details are given in Section 1.0.

In performing the work required for the program, Chemical Systems Division (CSD) employed under subcontract the services of five scientists of national reputation: Drs. M. Quinlan, M. E. Gurtin, W. L. Hufferd, R. Wool, and R. A. Schapery. The program effort combined the comprehensive experience and specialized test capabilities of CSD in solid propellant mechanical properties with the theoretical expertise of these scientists. In addition, Dr. J. E. Fitzgerald was retained as a consultant to participate in the periodic technical reviews of the program status.

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1.0 PROGRAM OBJECTIVES AND OVERVIEW

The objective of the Propellant Nonlinear Constitutive Theory Extension Program is to develop and demonstrate an accurate and usable nonlinear thermal-mechanical constitutive law for solid rocket propellants. The program has been conducted through three phases of a four phase project. These four phases are summarized below:

Phase I - Preliminary Study

A variety of nonlinear theories were considered and five methods were selected for further study.

- Modified Swanson Theory
- Russian (Hufferd) Theory
- Schapery Theory
- Gurtin Theory
- Quinlan Theory.

Dr. Richard Wool presented a review of available micro-mechanics theories which could be considered for inclusion within the nonlinear constitutive theories.

Phase II - Uniaxial Isothermal Investigation

A series of unaxial tests were conducted with two materials - a PBAN and an HTPB propellant. The data was fit to each of the nonlinear theories. Then the material constants derived for each analytic method were used in a predictive calculation of a complex laboratory test history which included some typical long time rocket motor mechanical bonding sequences. A review of the analytic methods and predictive results showed that all of the theories could be adapted to give better correlations with realistic motor loading conditions. Other test histories were also suggested that would permit more direct evaluation of the pertinent nonlinear material behavior required for each theory. Numerical difficulties were encountered with some theories and refinements were determined

that would hopefully eliminate these problems. All five theories were considered acceptable and were to be further developed in the next phase.

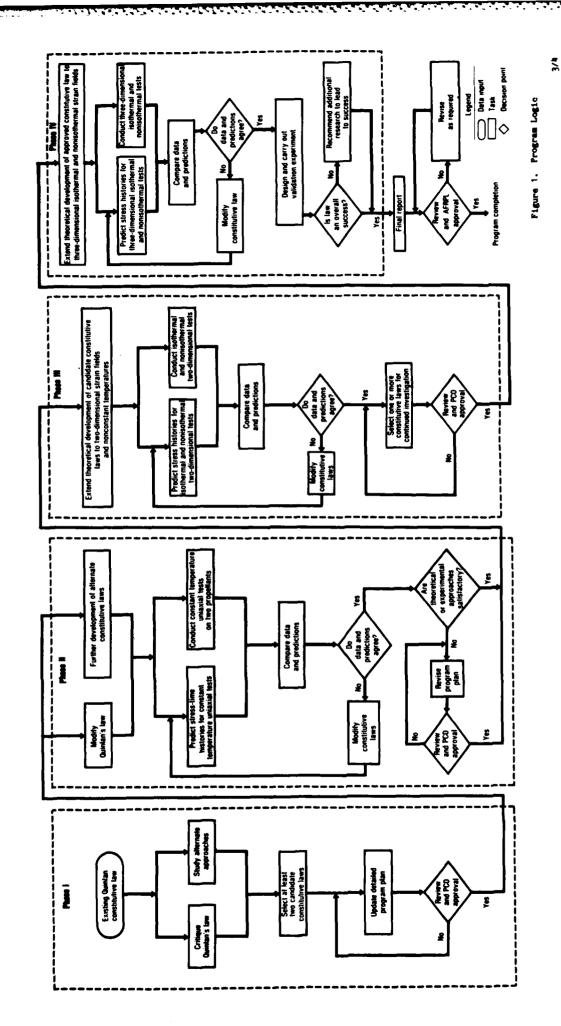
Phase III - Two-Dimensional and Variable Temperature Investigation

The analytic and numerical improvements were made with each theory. These refined methods were then evaluated with the test data from phase II plus the additional recommended tests from the phase II review meetings. Most of the analytic methods looked very good with the final refinements. This effort, using the latest data, used most of the scheduled time for this phase. In addition to the suggested uniaxial mechanical tests, a variety of shear, biaxial and uniaxial (some with simultaneous thermal and mechanical loading) tests were conducted with some very complex load histories. This latest test data was only considered in a preliminary way since the theory modifications were very exstensive.

Phase IV - Three Dimensional Investigation (Not started)

The additional laboratory data developed in phase III plus some instrumented structural test vehicle data will be used with the various nonlinear constitutive theories. One or more selected theories will be determined and the computer codes will be installed and checked out on the Air Force Rocket Propulsion Laboratory's computer.

The program logic chart relating each of the phases and their respective tasks is presented in Figure 1.



WARRY CONTRACT CONTRACT

STATES OF STATES

2.0 TASK DESCRIPTION

2.1 PHASE I - PRELIMINARY STUDY

The objective of phase I was to critique the Quinlan theory, propose at least one alternate approach to the constitutive law solution, and make detailed research plans for evaluating and modifying the candidate constitutive law approaches.

2.2 PHASE II - UNIAXIAL/ISOTHERMAL INVESTIGATION

The objective of phase II was to carry out modifications to Quinlan's law and to do theoretical development of the other candidate constitutive laws. These laws were used to make stress-time predictions for uniaxial/isothermal tests. Concurrently, actual uniaxial/isothermal tests were conducted in the laboratory. A comparison of the predictions and actual data was then made, and the theoretical and experimental approaches were evaluated.

2.3 PHASE III - TWO-DIMENSIONAL AND VARIABLE TEMPERATURE INVESTIGATION

The objective of phase III was to extend the theoretical development of the candidate constitutive laws to two-dimensional and variable temperature tests. Stress-time predictions were made concurrently with actual testing in the laboratory. A comparison of predictions and actual data was made and two constitutive law candidates were selected for further investigation from those results.

2.4 PHASE IV - THREE-DIMENSIONAL INVESTIGATION

The objective of the final phase is to extend the theoretical development of the candidate constitutive laws to three-dimensional and variable temperature history tests. Concurrently, three-dimensional, variable temperature laboratory tests are to be conducted. Data and predictions will be compared and any final modifications to the constitutive law made. A validation experiment is to be conducted, and stress-time predictions made with the finalized version of the nonlinear constitutive law. A final assessment of the overall success of the new law will be made and recommendations will be presented for further avenues of research. Recommendations for utilizing the new law in existing solid rocket motor structural analysis techniques will also be made.

3.0 LABORATORY TESTS, RESULTS, AND SUMMARY

The laboratory testing was divided into the categories of uniaxial/isothermal, two-dimensional and variable temperature, and three-dimensional investigations.

The two propellants selected for the program were (1) a PBAN currently being used in the first stage of the Titan missile system (UTP-3001-750/7768) and (2) a HTPB propellant developed for the IUS motor (UTP-19,360B-400/1777). The first numbers are the propellant designation, the next the mixer size, and the last a batch number.

In each of the uniaxial or biaxial groups, a specific test of each type was selected to show the test details in graphic and tabular form. While tests were run on both propellants, only one is shown. The details of test temperatures and rates are discussed with each test type.

3.1 UNIAXIAL/ISOTHERMAL INVESTIGATION

Testing uniaxial specimens of UTP-3001 and UTP-19,360B propellants, in phase II of the contract, was done for the nondamaged material as indicated in Figure 2 and for damaged material per Figures 3 and 4. Most of the tests were run with $1/2 \times 1/2 \times 6$ -in. bars with redwood end tabs. The exceptions were stress endurance (test 2) and constant rate (comparison to test No. 4) data were obtained with JANNAF Class B specimens. (1) The details of the individual test types are discussed in subsections below.

The uniaxial bars were machined from redwood boxes of propellant. The redwood was sealed then lined in the same manner as a rocket motor. After a partial cure of the liner, propellant was cast in the box and the system cured to provide a good bond to the redwood end tabs. The redwood box assembly and finished specimen are shown in Figure 5. After the specimen is mill finished,

Reference 1 - Solid Propellant Mechanical Behavior Manual, CPIA Publication Wo. 21, Section 4.3.2

Test No.	Test Description	Temperature, °F	Pressure, psig	Rate, in./min	Strain, %	Experimental Effects	Strain History
1	Constant rate	70 120 40	0 0 0	0.001	To failure	Time and rate temperature sample type	
2	Stress endurance	70 120 40	0 0	_ _	To failure	Time and temperature	1
3	Multirate	70	0	0.1-1 1.0 — 0.1	12 12	Rate change	High-low Low-high
4	Stress relaxation	70 120 40 23	0 0 0 0	1 1 1 1	3 3 3 3	Temperature	
Note:	Nominal tests w	ere run with three	samples per s	set	1		t Legend: $\epsilon = \text{strain}$ $t = \text{time}$

Figure 2. Uniaxial Isothermal Nondamaged Tests 24406R1

a 1/8 in. hole is drilled on the center line of each end tab for attachment purposes. (Attachment fixtures are shown in Figure 16).

Due to the enormous test load compared to the available time, both calendar and contract wise, the decision was made to run all tests on six-channel testers. Both the Chemical Systems Division (CSD) manufactured six-channel tester and a modified Instron were used in conjunction with a Hewlett-Packard computer to collect digitized data for the tests. (See Appendix A details on the automated data reduction system). While the CSD tester was equipped with an oscillograph as backup for relatively short duration tests, the modified Instron had no backup. In instances where a power surge occurred, the data were lost and the test had to be rerun.

			Dem	Damage Cycle	•			Test			
Test No.	Test Description	Temperature, °F	ure, Pressure, pei	Rate, in./min	Strain, %	Remarks	Temperature,	Pressure, psi	Rete, in/min	Strain, %	Strain
S	Multiple loading with rest periods:	02	0	0.1. 1,	ဗ	30-min hold between cycles	02	0	0.1, 1, 10	3 to 12	, www
မှ	Creep	70, 120, 40	0	-	1/4, 1/2, ~maxi- mum	1/4, 1/2, 3-hr creep ~ maxi- periods: first mum step only for 120 and 40 °F	20	0	-	1/4, 1/2. ~maximum	الله الله
7	Cyclic loading	02	0	_	4. 8. 12	20 cycles: return to zero stress and monitor E recovery	02	0	-	4, 8, 12	· M. Zo cycles
&	24-hr relaxation	02	0	20	4, 8, 12	Monitor the unload	02	0	-	4, 8, 12	Ů.
6	Predamage relaxation (1 hr)	0.2	0	0.1	6. 12	30-min hold between loading: monitor unload	02	0	20, 1	3. 4. 8.	, W
01	Complex multiple load	20	0	5.	12. 8, 4	High strain followed by low strain; return to zero stress and monitor strain	70	0	5.1, 1,	12. 8. 4	00
Note	Nominal tests	Note. Nominal tests were run with three samples per set	ee sambles t	er set							Legend. c = Strain t = Ime

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Figure 3. Uniaxial/Isothermal Damaged Tests

	Test Description	Damage Cycle Test	Strain Cycle
	Quinlan complex history	Single samples clamped in rigid Instron jaws were run on UTP-3001 and UTP-19,360B. In returning to zero strain the samples were put into compression. Data are reported in "Data Package F".	
	Similitude	Similitude tests were run with 6-in. bar samples of UTP-3001 and UTP-19,360B. Data are reported in "Data Package E" and the September meeting handout.	
	Three step relaxation	Short and long time tests were run with 6-in. bar samples of UTP-3001 and UTP-19,360B. Data are reported in "Data Package E" and the September meeting handout.	الم
	Nominal tests were	Nominal tests were run with three samples per set	Legend: <pre></pre>

Figure 4. Uniaxial/Isothermal Damaged Propellant Tests

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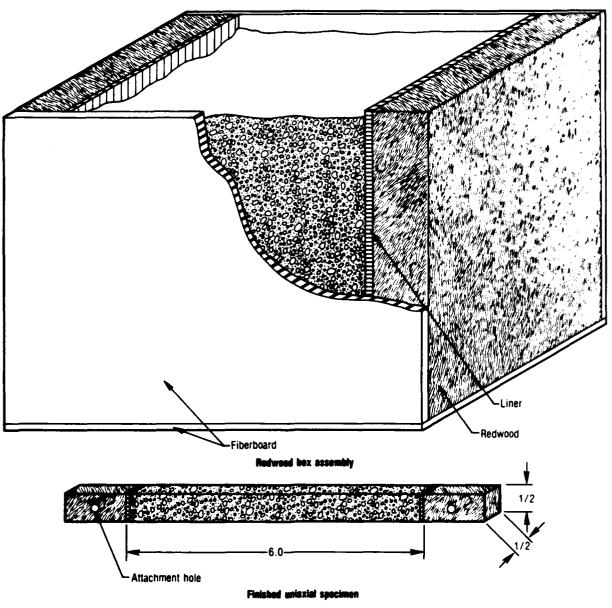


Figure 5. Uniaxial Bar Specimen

28804

The attachment linkages on both testers were such that the specimens could not be put into compression when the crosshead was returned to an equivalent zero strain position. The strain measurement was done with a linear potentiometer attached to the crosshead; consequently, the data had to be modified to reflect the propellant strain relaxation behavior after the stress had returned to zero (free hanging specimen).

Strain relaxation was measured on samples in some of the tests during the final unload cycle. Cathetometer measurements were made periodically and strain versus time data were plotted. These data were used to estimate the relaxation behavior on cyclic tests where there was insufficient time for measurements.

A data modification was made to estimate the peak or minimum stress and strain points, which were not recorded by the digitized data acquisition system. The sampling rate limited the crosshead rate that could be used and still obtain enough points to adequately define a ramp. The available computer memory also influenced the sampling rate in some of the long tests.

3.1.1 Constant Rate Test No. 1

Uniaxial constant rate tests to failure were run on 6-in. bars of UTP-3001 and UTP-19,360B. The 70°F tests were at crosshead rates of 10, 1, 0.1, 0.01, and 0.001 in./min. while 40 and 120°F tests were at 10, 1, and 0.1 in./min. A typical load-time curve is shown in Figure 6 for the UTP-3001 at 75°F and 10 in./min., and tabular data are given in Table 1. Because of the computer

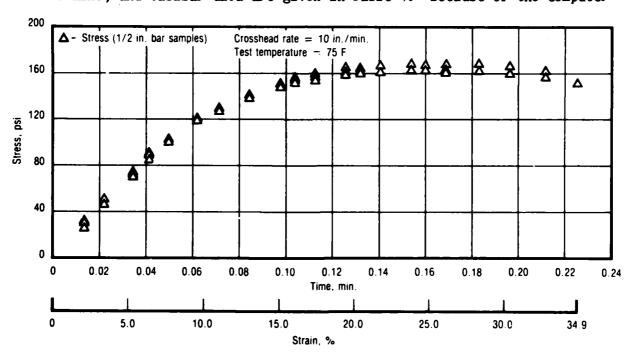


Figure 6. Test No. 1 - Straining to Failure for UTP-3001 750/7768 28387

TABLE 1. 1/2-IN. BAR STRAINING TO FAILURE

PROPELLANT: REQUESTOR: NOR:	UTP 3001 750/7768 Ceriten	•		DATE	MATE: 4/21/81	181		
DEFINITIONS: Time 6	OB Start	Test (min)		# T # T	TIONSHIPS o = Force f = Sampl	/Area e Extension	on/Length	
T(air) T(prop)	Verent Hodolog (p. 18st Air Temperat Test Propellant I	ure (F) emperature (F)		IMON	NAI VALUE Lest Temp Sage Leng XHD Rate	S: 75 F th = 6,10 in	e:	
CALIBRATION Cal Wt = Load Cal Uf set Pet Cal	### ### ##############################	44.12.	-3-11 -0.05	9AMPLE 3	4 0.0- 0.0-			
AREAS (sq in	n):	0.252	0.250	0.249	0.250			
SECANT MODUL 11 3500E - 82	US (psi): T(prop) T(air)	6:\0		1788.15	1.E. 3	926	1757.16	
4. 456/E - 15. 4. 9858E - 10. 4. 9858E - 10.				1537.49 1537.49		1500 1500 1500 1500 150 150 150 150 150	0.000 000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.	
7.1442E-U 7.1442E-U 8.4617E-U 9.7825E-U		*~WiD-0		1297 13 1224 03 1147 70	h h	+ 1	12.00 17.15.00 11.42.55	
2. 4. 4. 4. 4. 4. 4. 4. 4. 4. 4. 4. 4. 4.		70-0-10 440-40€		0.000 0.000	1039.79 978.79 948.63 849.99		2000 2000 2000 2000 2000 2000 2000 200	
1.0890H - 01 1.6800H - 01 1.9627H - 01 2.1155H - 01		60000000000000000000000000000000000000	852,16 814,05 760,62 71,01 653,60	2827.64 788.76 788.76 683.16 531.85 683.16 683.16	1		634 36 794 38 794 38 697 18 697 18	12.546 12.546 15.799 15.379

data sampling rate, the 10-in./min. tests were run one propellant at a time. Sample 4 (Table 1) shows the load cell reached the limit of its adjustment so did not record the specimen failure. The 6-in. bar specimens always fail below what would be obtained from JANNAF specimens. Since response properties are what is of interest, particularly in the small strain region, the uniform cross sectional area specimen does what it is supposed to do. The continual changing effective gage length of the JANNAF dogbone is avoided.

These data were also reduced to secant modulus $(\lambda \sigma/\epsilon)$ as shown in Figure 7. The data are compared to the stress relaxation modulus from test No. 4 later.

3.1.2 Uniaxial Stress Endurance Test No. 2

Stress endurance tests were run on the two propellants using JANNAF Class B dogbones. The ultimate failure properties were of importance in this test rather than small strain response hence the dogbones. This is a constant load test with plastic extensometer to monitor the strain increase with time (also known as a creep test). They were run at 23, 70, and 120°F. The data shifted

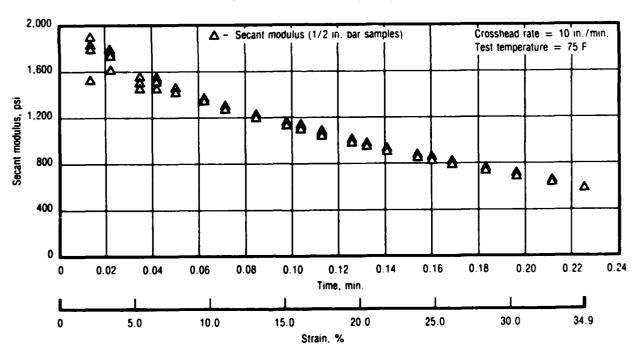


Figure 7. Test No. 1 - Secant Modulus for UTP-3001 750/7768 28751

to a master endurance curve as shown in Figure 8 for UTP-19,360B with typical 70°F data given in Table 2. The strain creep effect on stress is shown in Figure 9 where the engineering stress (F/A_O) is multiplied by the extension ratio (1+ ϵ) to account for the sample's necking down. The secant modulus (λ_{G}/ϵ) for the 70°F data are shown in Figure 10 and data are given in Table 3. This same type of data were generated for UTP-3001.

3.1.3 Multirate Test No. 3

Constant rate tests were run on the two propellants and the rate was changed in the middle of the test. Because of the different response for the high-low compared to the low-high, an example of both is given. The test was included with the nondamaged tests because there was no rest or reversal in the crosshead direction. The first leg of the test could be considered the damage. The 1.0 to 0.1 in./min. rate change is shown in Figure 11 for UTP-19,360B and the 0.1 to 1.0 in./min. is shown in Figure 12. The corresponding data for the first sample of each group are given in Table 4 and 5, respectively.

3.1.4 Stress Relaxation Modulus Test No. 4

The stress relaxation modulus tests were run at a nominal 3% strain using $1/2 \times 1/2 \times 6$ -in. samples of propellant bonded to redwood end tabs for both propellants. The samples were loaded to 3% strain at a crosshead rate of 1 in./min. for temperatures of 20, 43, 73, and 122°F. The load was monitored with time. Strain was determined by cathetometer measurements on the samples. The relaxation modulus (E_R) was then calculated at specific time intervals by the following equation:

$$E_R = \frac{F}{A_0} \frac{\lambda}{\epsilon}$$

where F = force as measured by the load cell

An = initial area

 $\lambda = 1 + \epsilon$

 ϵ = strain

The master stress relaxation modulus data for UTP-3001 are presented in Figure 13 and typical data at 73°F are given in Table 6.

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- Master Uniaxial Stress Endurance Curve for UTP-19,360B-400/1777 Test No. 2 Figure 8.

Failure stress (F/A $_0$) (1 + ϵ), psi

TEST NO. 2 - UTP-19,360B 400/1777 CONSTANT ENDURANCE TEST

THE BATA REDUCTION COMBINES of 7 LEGENTAL STATEMENT OF THE STATEMENT OF TH		 ZANGE CONTROL OF THE	AST SQUARE FITS: In(t) vs In(s) and In(t) vs In(c) OTHORING HOPERAMENTALLY DETERMINED) ENTERS (1+0) to EXPERIMENTAL FAILURE TIME US LEAST SQUARE VALUES OF CORPELATION COEFFICIENT OF LEAST SQUARES LINE OF CORPELATION COEFFICIENT OF LEAST SQUARES LINE OF CORPELATION COEFFICIENT OF LEAST SQUARES LINE OF CORPER VALUES OF THESE FAILURE TIMES	FAILURE TIME FAILURE TIME FAST SQUARES	1916				
	0.000 0.000	1.00 (UE (p s 1)	A month of the control of the contro	e segments segments segments	R 25/2-25/3	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1		
	A. 0801	 90.	8(F) 6.6543 0.0070 0.0070	8 00 00 00 00 00 00 00 00 00 00 00 00 00		186.91 66.41	200 200 300 200 200 200	20.00 20.00	5 47 64 50 50 50 50 50 50 50 50 50 50 50 50 50

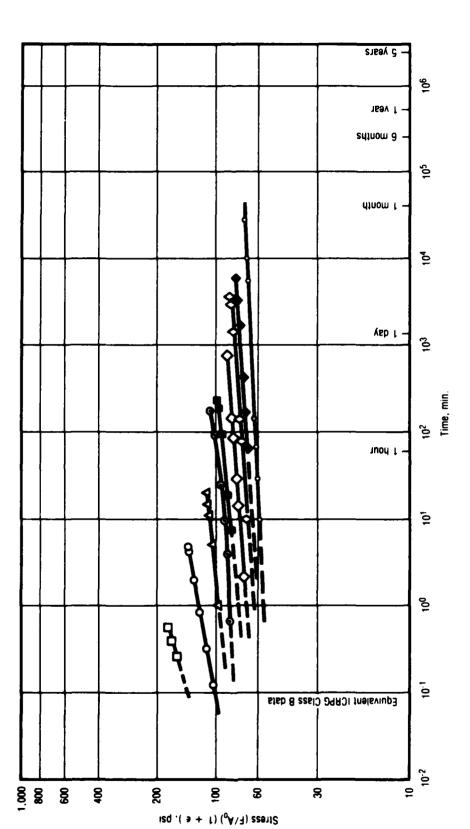
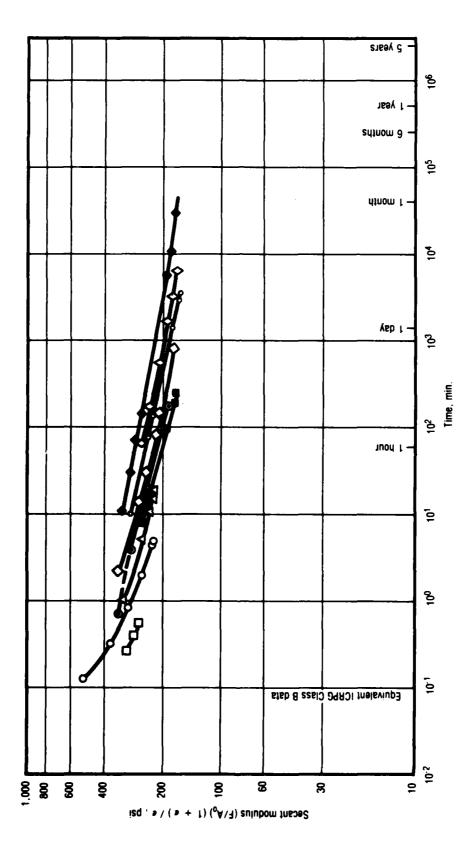


Figure 9. Test No. 2 - Uniaxial Stress Endurance Creep Behavior UTP-19,360B-400/1777 at 700F



Test No. 2 - Uniaxial Stress Endurance Creep Secant Modulus Behavior for UTP-19,360B-400/1777 at 70°F Figure 10.

TABLE 3. TEST NO. 2 - UNIAXIAL STRESS ENDURANCE CREEP BEHAVIOR FOR UTP-19,360B-400/1777 at 70°F

(SHEET 1 OF 2)

T8706

Temper- ature,	Sample	Time, min.	Strain,	Load,	Area, in. ²	Stress (F/A), psi	λσ, psi	Secant Modulus $(\lambda \sigma/\epsilon)$, psi
70	1	0.13 0.33 0.87 2.0 4.4 5.0	20 30 40 50 60	7,180	0.185	85.5	102.6 111.2 119.7 128.3 136.8 137.7	513 370 299 257 228 226
70	2	0.27 0.41 0.55	50 60 64	8,720	0.184	104.4	156.6 167.1 171.2	313 279 268
70	3	1.0 5.3 11.0 14.9 21.0	30 40 46 50 50	6,000	0.179	73.8	95.9 103.3 107.7 110.7	320 258 234 221 221
70	4	7.5 18.0 95.0 196.0 247.0	33 38 49 56 56	5,200	0.182	62.9	83.6 86.8 93.7 98.1 98.1	253 228 191 175 175
70	5	65.0 190.0 440.0 1,640.0 3,325.0 6,086.0	27 30 35 40 45 47	4,480	0.180	54.8	69.6 71.2 73.9 76.7 79.5 80.6	258 237 211 192 177 172
70	6	2.2 14.0 30.0 84.0 150.0 833.0	21 29 32 37 38 50	4,750	0.177	59.1	71.5 76.2 78.0 80.9 81.6 88.7	340 263 244 219 214 177

TABLE 3. TEST NO. 2 - UNIAXIAL STRESS ENDURANCE CREEP BEHAVIOR FOR UTP-19,360B-400/1777 at 70°F

(SHEET 2 OF 2)

T8706

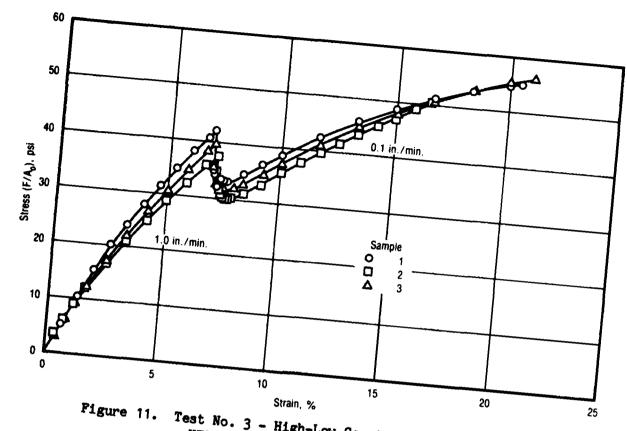
Temper- ature, F	Sample	Time,	Strain,	Load,	Area, in. ²	Stress (F/A), psi	λσ, psi	Secant Modulus $(\lambda \sigma/\epsilon)$, psi
70	7	0.7	25	5,630	0.183	67.8	84.8	339
		4.0	30	•			88.1	294
		10.0	35				91.5	261
		26.0	42				96.3	229
		92.0	51				102.4	200
		177.0	57				106.5	187
70	8	10.0	24	4,660	0.184	55.8	69.2	288
		77.0	30				72.5	242
		140.0	32				73.7	230
		1,418.0	44				80.4	182
		2,900.0	49				83.1	169
		3,500.0	49				83.1	169
70	9	10.0	19	4,030	0.178	49.9	59.4	312
		30.0	20	•			59.9	299
		70.0	22				60.9	277
		140.0	24				61.9	258
		5,496.0	35				67.4	193
		10,040	37				68.4	185
		Unbroken 28,600	≈ 41				70.4	172

3.1.4.1 Constant Rate Modulus

Constant rate modulus tests were run on the 6-in. bar samples described above and on JANNAF Class B specimens. The 6-in. bar samples use the wood-to-wood distance as the effective gage length while the JANNAF's were analyzed using 2.70-in. effective gage length through the test even though it is varying. Strain was determined by the crosshead travel. The constant rate modulus (F(t)) is calculated from the following equation at specific strain levels through the tests.

$$F(t) = \frac{F}{A_0} \frac{\lambda}{\epsilon(t)}$$

where: $\epsilon(t)$ = strain at time t



Test No. 3 - High-Low Constant Rate Tests of UTP-19,360B-400/1777 at 70°F

Constant rate modulus data for UTP-3001, with the curve drawn through the small strain portion of the results, are compared to the relaxation modulus in Figure 14. Tabular data for the relaxation and constant rate moduli for 6-in. bars and JANNAF's are given in Table 7.

3.1.5 Multiple Loading Test No. 5

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The multiple loading tests on 6-in. bars of UTP-3001 and UTP-19,360B were run five cycles with increasing strain levels for each cycle and with a rest period between cycles. All tests were at 70°F and at crosshead rates of 5, 1, and 0.1 in./min. An attempt was made with UTP-3001 to run at 10 in./min. (i.e., planned rate instead of five) but the data sampling rate did not provide sufficient data points to clearly define the load-time curve, particularly on the first low strain cycle. As previously mentioned, this data had to be

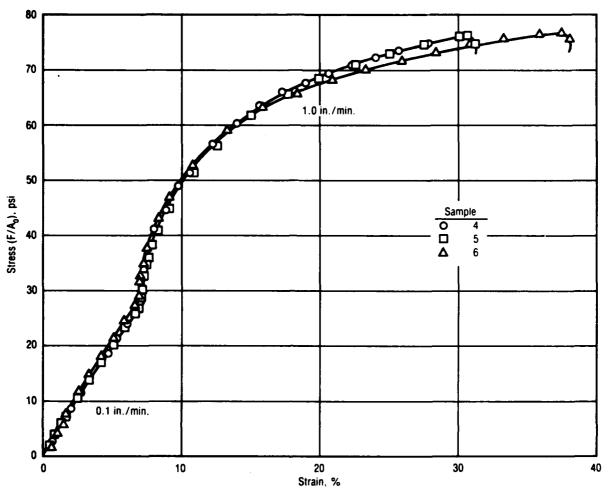


Figure 12. Test No. 3 - Low-High Constant Rate Tests of UTP-19,360B-400/1777 at 70°F

The rework of the data consisted of estimating maximum and minimum stress and strain points that were not picked up by the digitized data acquisition system. The data reduction system computed strain from crosshead travel. This was satisfactory except at and below zero stress. The sample linkage attachment was such that specimens would not be put into compression with the exception of test No. 11, which was run in an Instron with rigid clamp jaws. The actual propellant strain decay was estimated from other tests where strain recovery was monitored by cathetometer measurements for the part of the tests at zero stress (i.e., no load on the samples). The 5-in./min. crosshead test for UTP-19,360B was selected as typical and shown in Figure 15. While the load-unload ramps

TABLE 4. TEST NO. 3 - HIGH-LOW CONSTANT RATE TESTS OF UTP-19,360B-400/1777 AT 70°F WITH 1/2 x 1/2 x 6-IN. BAR SAMPLES

T8707

Sample No. 1

Crosshead speed, in./min. 1.0 and 0.1 Chart speed, in./min. 10 and 1

Load scale 5 lb/in.

Area, in. 0.500 x 0.501

Gage length, in. 6.00

istance, n.	Load Signal, in.	Strain,	Stress (F/A _O), psi
0	0	0	0
0.4	0.26	0.667	5.190
0.8	0.52	1.333	10.379
1.2	0.77	2.00	15.369
1.6	1.00	2.667	19.960
2.0	1.20	3.333	23.952
2.4	1.39	4.00	27.745
2.8	1.57	4.667	31.337
3.2	1.74	5.333	34.731
3.6	1.90	6.00	37.924
4.0	2.04	6.667	40.719
4.15	2.10	6.917	41.916
Change to 0.			
4.20	1.74	7.00	34.731
4.25	1.70	7.083	33.932
4.30	1.68	7.167	33.533
4.40	1.66	7.333	33.134
4.50	1.65	7.50	32.934
4.60	1.65	7.667	32.934
5.0	1.73	8.33	34.531
5.5	1.83	9.167	36.527
6.0	1.93	10.00	38.523
7.0	2.12	11.667	42.315
8.0	2.29	13.333	45.709
9.0	2.43	15.00	48.503
ó.o	2.55	16.667	50.898
1.0	2.66	18.33	53.094
2.0	2.74	20.00	54.691
2.3	2.76	70.50	55.090

TABLE 5. TEST NO. 3 - LOW-HIGH CONSTANT RATE TESTS OF UTP-19,360B-400/1777 AT 70°F WITH 1/2 x 1/2 x 6-IN. BAR SAMPLES

T8708

Sample No. 4

Crosshead speed, in./min. 0.1 and 1.0 Chart speed, in./min.

1.0 and 1.0

Load scale

5 lb/in.

Area, in. Gage length

 0.502×0.500

6.00

	istance,	Load Signal, in.	Strain,	Stress (F/A _O), psi
0	}	0	0	0
0	.4	0.14	0.667	2.789
0	.8	0.28	1.333	5.578
1	.2	0.43	2.00	8.566
1	.6	0.57	2.667	11.355
2	•0	0.69	3.333	13.745
2	.4	0.83	4.00	16.534
	.8	0.94	4.667	18.725
3	.2	1.07	5.333	21.315
3	.6	1.19	6.00	23.705
4	•0	1.30	6.667	25.896
4	.18	1.36	6.967	27.092
C	hange to 1 i	n./min.		
4	.2	1.68	7.300	33.466
4	-25	2.05	8.134	40.837
4	•3	2.23	8.967	44.422
4	•35	2.45	9.800	48.805
4	.4	2•57	10.634	51.195
	•5	2.82	12.300	56.175
	.6	3.03	13.967	60.359
	•7	3.18	15.634	63.347
	.8	3.30	17.300	65.737
	•9	3•39	18.967	67.530
	.0	3.48	20.634	69.323
	•1	3.55	22.300	70.717
	.2	3.62	23.967	72.112
5	•3	3.68	25.634	73.307
5	.43	3.75	27.80	74.701

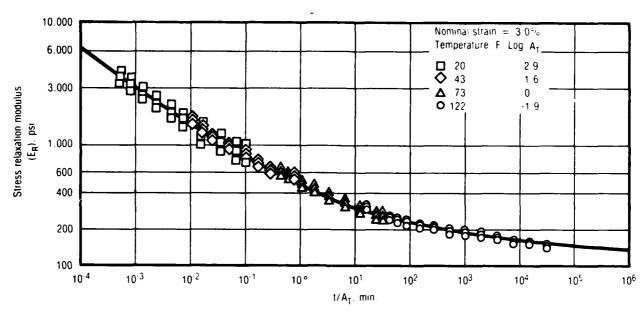


Figure 13. Master Modulus Data for UTP-3001-750/7768 with Experimental Shift

are not clearly shown in the figure because of the scale necessary to show the total test duration, the detailed data are given in Table 8.

3.1.6 Creep Test No. 6

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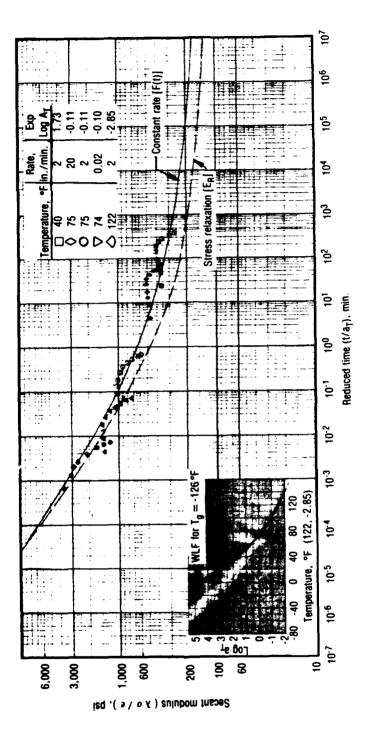
Creep tests were run on 6-in. bars of UTP-3001 and UTP-19,360B by hanging weights on the samples and allowing them to deform under load. The loading was done by setting the weight on the Instron crosshead and then lowering the crosshead away from them as shown in Figure 16. Proper spacing of the segmented weight provided a means of incrementally unloading the samples. The two propellants were run separately to allow access for spacer insertion without disturbing the samples. The procedure for unloading part of the load and still maintaining the balance as a constant load creep test is shown in Figures 17. Tests were run at the maximum load the sample would be sure to survive, then at half and quarter loads. Tests were repeated for 70, 120, and 40° F. The full stress creep test for UTP-19,360B at 71°F is shown as typical. The strain time curve from crosshead and cathetometer measurements is shown in Figure 18 with engineering stress, corrected stress, and secant modulus shown in Figures 19, 20, and 21, respectively. The test data are given in Table 9 and summarized in Table 10 with cathetometer and secant modulus data.

(Text continued on page 29.)

TABLE 6. TEST NO. 4 - 1/2-IN. BAR STRESS RELAXATION DATA REDUCTION

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	Test (min) re (F) mperature (F)	-0.789 -1.004 -1.016 -0.163 -0.120 -0.149	99105 73535 73535 98095 88255 88255 73080 3.05 3.05	0.2505 0.2495 0.2499	44 DE CONTRACTOR CONTR	544.05 575.05 583.26 583.26 583.26 583.20 583.05
ROPELLANT: UTP 3001 750/7768 EQUESTOR: Carlton OR:	DEFINITIONS: Time From Start of Time s Stress (DSI) s Strain (%) s Strain (%) s Strain (%) s Strain s	CALIBRATION DATA; Pretest: Col Wt = 5 lbs Difflection (lbs/volt) Zero Offset (volt,	CATMATOMETER STRAIN DATA: Initial Lower Final Upper Final Lower Strain (%)	AREAS (sq in)	LOAD DISPLACEMENT DATA: (011)) 4.14E-01 4.05E-01 5.42E-00 5.42E-00 5.58E-01 3.03E-01 3.03E-01	MODULUS DATA (psi): 1(air) 1.1HE



Constant Rate Master Modulus Data for UTP-3001-750/7768 JANNAF Specimens Figure 14.

TABLE 7. MASTER MODULUS CURVES FOR UTP-3001 STRESS RELAXATION AND CONSTANT RATE TESTS

(SHEET 1 OF 2)

T8709

Reduced Time, t/A _T , min.	6-in. Bar, E _R , psi	6-in. Bar, F(t), psi	JANNAF, F(t), psi
1 x 10 ⁻⁵	-	•	-
2 x 10 ⁻⁵	-	-	-
4 x 10-5	8,750*	10,000#	8,750*
6 x 10-5	7,600*	8,900*	7,800*
8 x 10 ⁻⁵	6,850#	8,200	7,150#
1 x 10 ⁻⁴	6,350#	7,700*	6,700*
2 x 10 ⁻⁴	5,000#	6,300*	5,400#
4 x 10 ⁻⁴	4,000#	5,200#	4,450*
6 x 10 ⁻⁴	3,500	4,650#	3,950*
8 x 10 ⁻⁴	3,170	4,300	3,630
1 x 10-3	3,000	4,070	3,400
2 x 10 ⁻³	2,400	3,350	2,760
4 x 10-3	1,930	2,760	2,300
6 x 10 ⁻³	1,720	2,520	2,070
8 x 10 ⁻³	1,570	2,320	1,920
1 x 10 ⁻²	1,470	2,180	1,800
2 x 10 ⁻²	1,220	1,830	1,500
4 x 10 ⁻²	1,020	1,550	1,260
6 x 10 ⁻²	910	1,400	1,130
8 x 10 ⁻²	840	1,300	1,070
1 x 10 ⁻¹	800	1,240	1,020
2 x 10 ⁻¹	675	1,060	870
4 x 10 ⁻¹	580	905	765
6 x 10 ⁻¹	520	835	705
8 x 10 ⁻¹	490	790	665
1 x 100	470	755	640
2 x 10 ⁰	412	660	570
4 x 100	362	580	510
6×10^{0}	340	540	475
8 x 10 ⁰	320	510	460
1 x 101	308	490	440
2 x 10 ¹	278	445	400
4 x 10 ¹	256	403	370
6×10^{1}	243	380	352
8 x 10 ¹	235	365	340

TABLE 7. MASTER MODULUS CURVES FOR UTP-3001 STRESS RELAXATION AND CONSTANT RATE TESTS

(SHEET 2 OF 2)

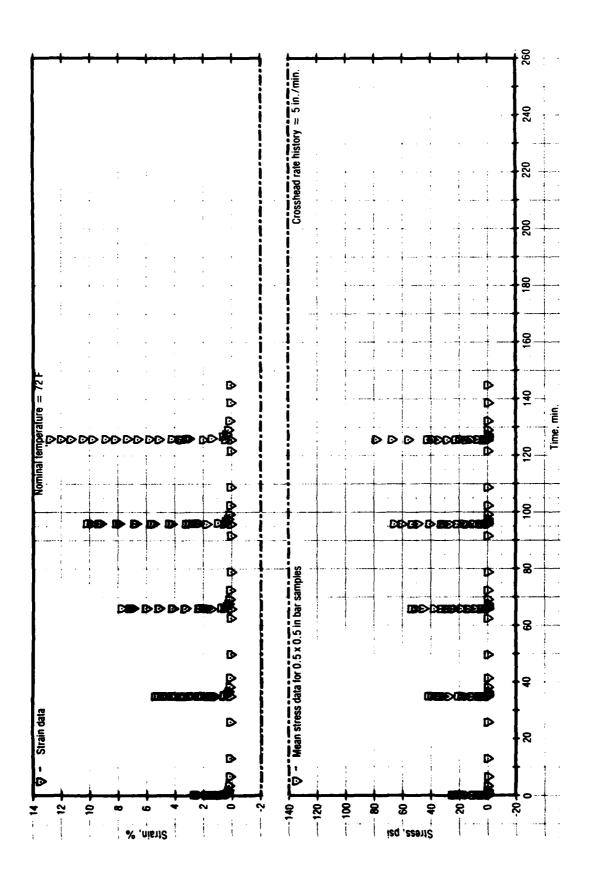
T8709

Reduced Time, t/A _T , min.	6-in. Bar, E _R , psi	6-in. Bar, F(t), psi	JANNAF, F(t), psi
1 x 10 ²	230	353	332
2×10^{2}	218	326	310
4 x 10 ²	207	300	290*
6×10^{2}	201	290	280*
8 x 10 ²	198	280	273*
1 x 103	193	275	268#
2 x 10 ³	185	258#	255 *
4×10^3	182	245#	242*
6×10^{3}	179	236*	235#
8 x 10 ³	172	232*	230*
1 x 10 ⁴	169	228#	227#
2 x 10 ⁴	163	219#	218#
4 x 10 ⁴	157#	210#	210*
6 x 10 ⁴	154 *	205#	205*
8 x 10 ⁴	152*	203*	202#
1 x 10 ⁵	151#	200#	200#
2 x 10 ⁵	147*	193#	194#
4 x 10 ⁵	143#	188 #	188*
6 x 10 ⁵	141#	184#	185 *
8 x 10 ⁵	140#	182*	183*
1 x 106	139#	181*	182*
2 x 10 ⁰	136*	177*	178*
4 x 10°	133*	174*	174#
6×10^{6}	131#	171 *	172
8 x 10 ⁶	130*	169*	170*

^{*} Extrapolated data

3.1.7 Cyclic Loading Test No. 7

Cyclic loading tests were run on 6-in. bars of UTP-3001 and UTP-19,360B propellants at ambient temperature. The cycling was for 20 cycles at nominal strain levels of 4, 8, and 12% for UTP-19,360B with UTP-3001 limited to 12%. At the end of the test (after unloading to zero stress) the strain was monitored on the samples with a cathetometer. The test at a nominal 12% strain is shown in Figure 22 for UTP-19,360B. These data were modified to insert the estimated maximum stress points and the propellant strain decay while at zero stress. Data for the test are given in Table 11. (Text continued on page 34.)



ASSESSED BEREEZE HERBERG (BORRESSE) RECERTOR

Test No. 5 - Stress While Cycling for UTP-19,360B-400/1777

TABLE 8. TEST NO. 5 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 1 of 3)

DATE; 5/6/81 (PERATOR: JWD	KELATIONSHIPS: office/Area office/Area cfice/Area NOMINAL VALUES: Test Temp Gage Length # 5.9 NOM. Strain # 5.9 XHD Rate # 5.5	5.850 6.098 6.118 0.049 0.045 0.093	0.251 0.253 0.255	88 44 48 89 89 89 89 89 89 89 89 89 89 89 89 89
PROPELLANT: UTP 19360B 400/1777 REQUESTOR: Carlton WOR:	DEFINITIONS: Time = Time From Start of Test (min) To = Stress (Dsi) E = Strain (Z) T(air) = Test Air Temperature (F) T(prop) = Test Propellant Temperature (F)	CALIBRATION DATA: . Cal Wt == 5.0 lbs. Cal Wt == 5.0 lbs. Und Cal (lbs/velts) Offset (volts) == Temp (f) velts ==	AREAS (sq in):	STRESS DATA (psi): 1 0 06607 2 0 01226 3 0 01226 4 0 0034009 5 0 003356 5 0 003356 6 0 003356 11 0 0 05238 11 0 0 05238 12 0 05558 13 0 05558 14 0 0 05558 15 0 05558 16 0 05796 17 1 581063 73 4 74 2 16 19 6 48156 73 4 74 1 10 0 6 73 6 73 6 74 1 10 0 6 73 6 73 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7

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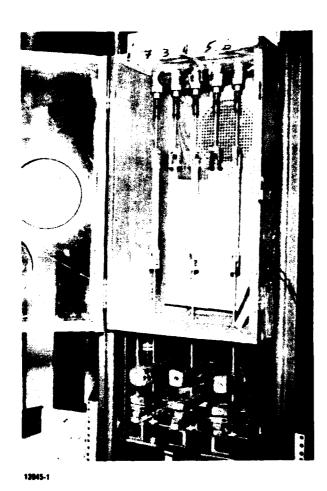


Figure 16. Test No. 6 - Creep Test with 6-in. Bar Specimens 28747

3.1.8 Relaxation Test No. 8

Stress relaxation tests were run on 6-in. bars of UTP-3001 and UTP-19,360B propellant for a 24-hr period and then monitored for strain decay after they were unloaded. The tests at ambient temperature were repeated for 4, 8, and 12% nominal strain levels for UTP-19,360B but limited to 4% for UTP-3001. They were loaded at a crosshead rate of 20 in./min. and unloaded at 1 in./min. after the 24-hr relaxation. A plot for the UTP-19,360B at 4% nominal strain is shown as typical in Figure 23, and data are given in Table 12.

3.1.9 Predamaged-Relaxation Test No. 9

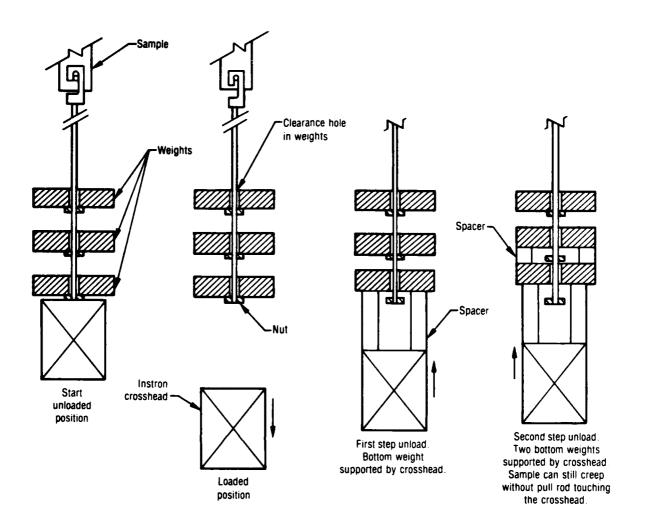
The predamage-relaxation tests were run with 6-in. bars on UTP-19,360B propellant at ambient temperature.

They were preloaded to 12% and unloaded at a crosshead rate of 0.1 in./min.,

allowed to rest, then reloaded to 8 or 4% strain at 20 in./min. After relaxing 1 hr the samples were unloaded at 1.0 in./min. and strain was monitored after unloading. These tests were repeated for a 6% predamage strain followed by a reload to 4 or 2% as above. The 12% predamage followed by 8% stress relaxation for UTP-19,360B is shown in Figure 24, and tabular data are given in Table 13. Comparison data for UTP-3001 was limited to 6% predamage followed by 3% relaxation. These data were modified to obtain the peak stress and strain relaxation after the samples were unloaded. Strain was monitored with a cathetometer after the relaxation part of the test.

3.1.10 Complex Multiple Load Test No. 10

The complex multiple load tests were run with 6-in. bars on UTP-3001 and UTP-19,360B propellants. Tests were run at crosshead rates of 5, 1, and



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MANAGERIA PARAMENTAL INDICATOR (SCHOOLS) SACRONAL

Figure 17. Procedure for Loading and Incrementally Unloading Creep Samples 28805

0.1 in./min. The test sequence was 12 to 8 to 12 to 4% strain, then unload, reload to 4% strain, and unload (four cycles) with cathetometer monitoring of strain decay. The same type of sequence was repeated with maximum strains of 8 and then 4% where the 4% maximum strain was shortened by one cycle. The 5 in./min., 12% maximum strain test on UTP-19,360B as typical is shown in Figure 25, and tabular data are given in Table 14. The data were reworked to insert maximum and minimum stress values as well as strain decay for unloaded specimens. The cathetometer strain after the final unload was incorporated into the data. The 5 in./min 12% strain test for UTP-3001 was deleted because the modulus was outside the range of the remainder of the data (i.e. carton to carton difference).

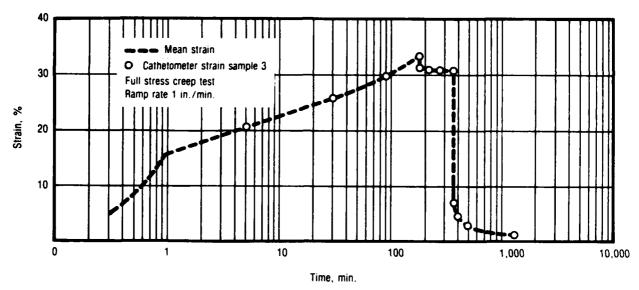


Figure 18. Test No. 6 - Strain-Time Data for Creep Test of UTP-19,360B-400/1777 at 71°F

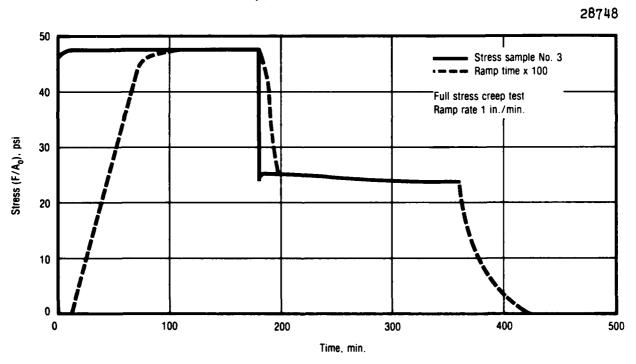


Figure 19. Stress-Time Data for Creep Test of UTP-19,360B-400/1777 at 71°F. 28749

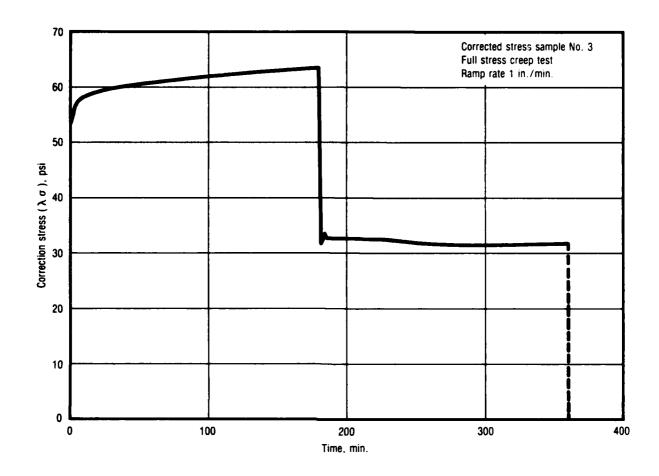


Figure 20. Corrected Stress-Time Data for Creep Test of UTP-19,360B-400/1777 at 71°F

3.1.11 Quinlan Complex History Test No. 11A

This particular test is very complex and required 2 full months to complete. It was run with single samples (one at a time) in an Instron using 6-in. bar specimens of UTP-3001 and UTP-19,360B propellant. The samples were pinned through the redwood end tabs and clamped onto pins to avoid crushing the redwood (because that would generate a compression load on the sample) during attachment. All coupling joints were heavy and tightly pinned so that the sample would be put into compression when returned to the zero strain position.

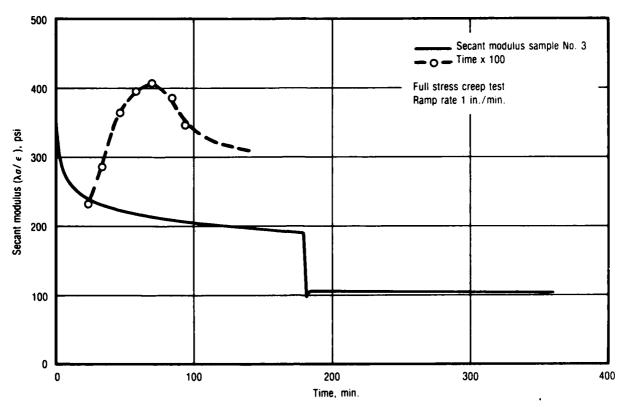


Figure 21. Secant Modulus Data for Creep Test of UTP-19,360B-400/1777 at 71°F

The test on UTP-3001 is presented but the test is so complex that it has been divided into three parts. Part 1 is described in Table 15 with data given in Table 16. Since an actual Instron trace of the load-time curve was obtained, the peaks and minimum (compression) stresses were selected data reduction points. Part 1 of the test is expanded in time scale to show some detail of the process (Figures 26 through 29). Part 2 is described in Table 17 with data given in Table 18. Test sequence for Part 2 is shown in Figures 30, 31, and 32. Part 3 (selected cycle maximum and minimum) are given in Table 19. The last figure of this part (Figure 32) is some of the cyclic loading at the end of the test. The chart speed was set such that good definition of the cycles could be recorded.

(Text continued on page 64.)

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9. TEST NO. 6	
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TABLE 9. TEST NO. 6 - 1/2	PROPELLANT: UTP 19360B 400/1777 REGUESTOR: Carlton WOR:	DEFINITIONS: Time Stress (DSI) Strain (X) T(air) # Test Air Temperatur T(prop) # Test Propellant Te	CALIBRATION DATA: Cal Wt = 5:0 lbs Cad Cal (lbs/volts) Offset (volts) Pot Cal (in/volts) = Temp (F)	AREAS (sq in):	STRESS DATA (psi): 6.57468E-03 2.3468E-03 2.4568E-03 2.4568E-01 1.58845E-01 1.5886E-01 1.5886E-01 1.5886E-01 1.5886E-01

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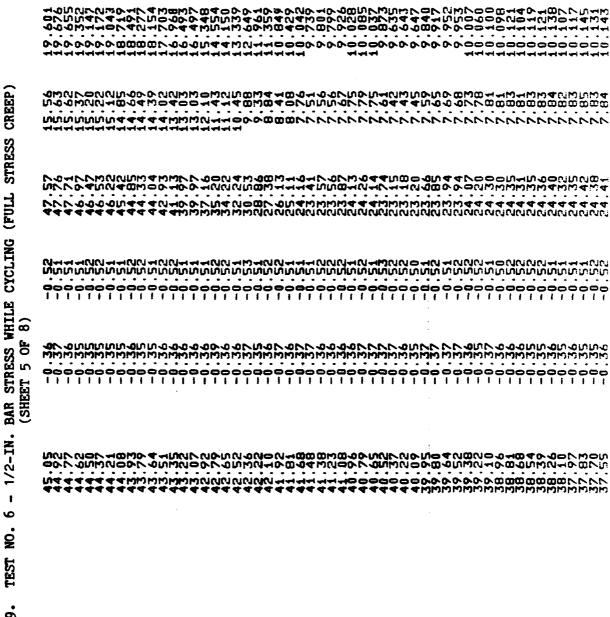
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TABLE 10. FULL STRESS CREEP TEST FOR UTP-19,3608-400/1777 at 710P

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	.	Cathetometer	r Strain,	×	Ø	Stress F(A _o),	o), pst			Secant
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9900				0.01	ŀ	0	ŀ	0	0	
0.1218				2.03	0	2.80	0	0	0	•
0.2377				3.96	N	11.41	8.95	8.95	•	235
0.3338				5.56	₹.	16.66	15.18	15.18	•	288 288
0.4643				7.74	22.43	26.24	26.32	26.32	28.36	366
0.5810				9.68	'n	33.87	34.84	34.84	•	395
0.6979				11.63	7	40.52	42.61	42.61	47.57	601
0.8146				•	ď	45.80	46.30	46.30	•	387
0.9261				15.44				46.34	53.49	346
ι.	Broke	Broke	20.43	20.43				47.50	57.20	2 8 0
18.557				24.20			47.53	47.52	59.05	244
2			25.64	25.64				47.42	59.58	233
56.958				27.80			47.32	47.32	60.47	218
8			29.81	29.81				47.39	61.52	506
108.16				30.70			94.74	47.40	61.95	202
150				32.40				47.46	62.84	₹
180			33.64	32.64				94.74	63.43	189
181.78				32.78			23.85	23.85	31.67	9.96
183.16				32.12			25.15	25.15	33.23	103.4
185			31.23	31.25				25.00	32.81	105.1
215			31.00	31.00				24.90	35.62	105.2
233.56				30.90			24.75	24.75	32.40	104.8
275			30.76	30.76				2 4 .00	31.38	102.0
284.76				30.78			24.23	24.23	31.69	102.9
360			30.83	30.83			•	24.23	31.70	102.8
363.26				12.00			•	ı		
363.32				10.00			•	•		
365			7.96	7.96			0	0		
390			4.61	4.61			0	0		
180			2.91	2.91			0 (0 (
1290			1.15	1.15			၁	2		



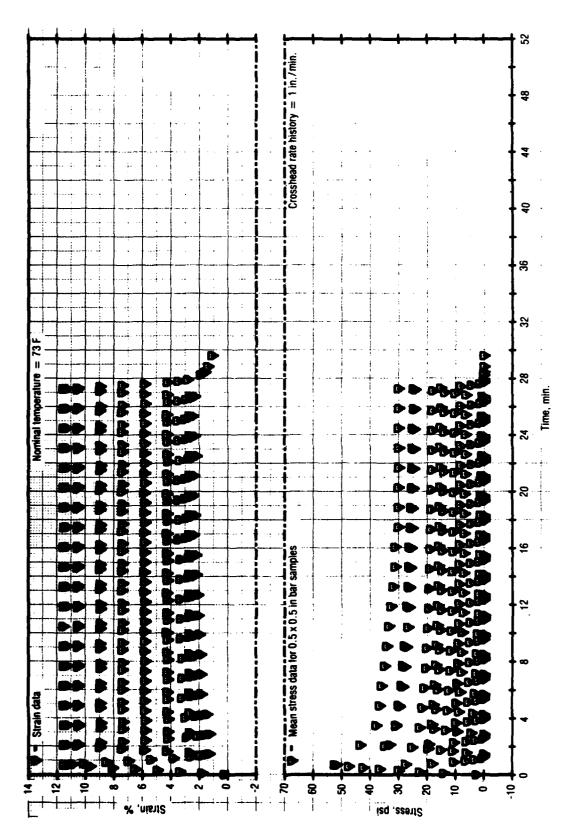


Figure 22. Test No. 7 - Stress While Cycling for UTP-19,360B-400/1777

a transport francisco (Probates) (appropri francisco)

TABLE 11. TEST NO. 7 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 1 OF 7)	1 400/1777 DATE: 6/4/8 OPERATOR: JWD	RELATIONSHIPS: psi) Force/A Force/A Force/A Fomple Force/A Fomple Force/A Fomple Force/A Fomple Force/A Fomple Force/A Fomple Force/A Force/F Force/A Force/A	SAMPLE 3 2 3 2 855 6.103 6.125 6.022 -0.024 0.073 8) = -0.389	0,249 0,250 0,250	######################################
	PROPELLANT: UTP 19360B REQUESTOR: Carlton WOR:	DEFINITIONS: Time # Stress (1 Cair) # Strain (2 Toly) # Test Air Toly (2 Toly) # Test Proj	CALIBRATION DATA: Cal Wt = 5.0 lbs. Lead Cal (lbs/velty) Offset (velts) Pot Cal (in/velt) Temp	AREAS (sq in):	C

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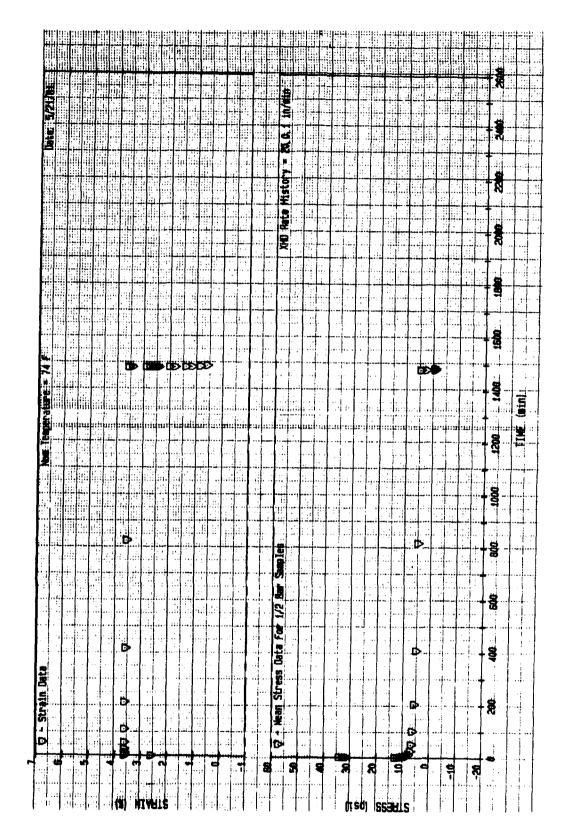
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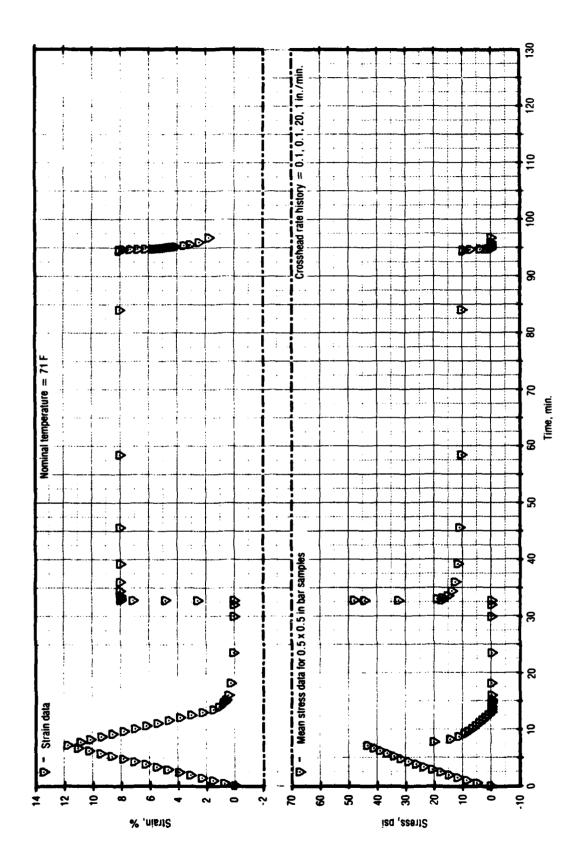


Test No. 8 - Stress While Step Straining for UTP-19,360B-400/1777 Figure 23.

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TABLE	19360B 4007	10 T T 0 T T 0 T T 0 T T 0 T T 0 T T 0 T T 0 T	Ten Ten Ten Ten	d lbs bs/velts) elts) n/velts) s	NO CONTRACTOR OF THE PROPERTY	-	1076 1076 1076	5416 7 5441 7 5469 7	55558 55594 57504 7704 7704	ひひりい B ひひのか 0 4 かの → 4	25993 25993 25993 2599 2599 2599 2599 25	50000000000000000000000000000000000000	0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0
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Test No. 9 - Stress While Step Straining for UTP-19,360B-400/1777 Figure 24.

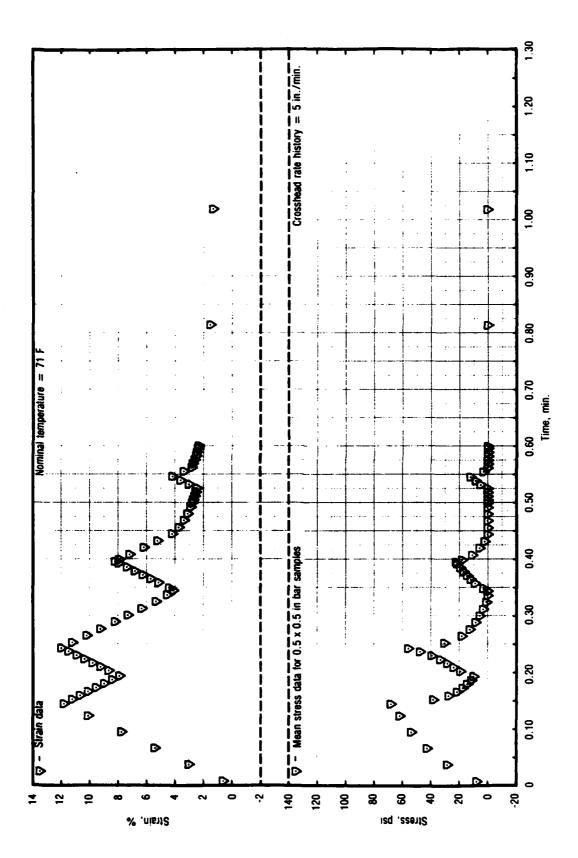
MANAGEM STATEMENT STATEMEN

TABLE 13. TEST NO. 9 - 1/2-IN. BAR STRESS WHILE STEP STRAINING (SHEET 1 OF 2)

STATES CONTRACTOR STATES CONTRACTOR

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est (min (F) erature			
Start of 1) and the start of 1) and the start of the star	3 4		► CCCCCCCCCCCCCC
##### 	DATA: 5.0 lbs (lbs/velt (velts) (in/velts (F)		2 0004400WW44RN034CVBBPPP0144 CLD44446BWBWBWCVCVCVCVA4444
DEFINITIONS: Time f f(alr) T(prop)	CALIBRATION Cal Wi Lead Cal Offset Per Cal	AREAS (sq In	888 44444444444666666666666666666666666
	KELATIONS: Time # Time From Start of Test (min) Time # Stress (psi) # Stress (psi) # Strain (%) # Strain (%) # Strain (%) # Strain (%) # Strain # 12 12 8 8 % **Non: Strain # 12 12 8 8 8 % **Non: Strain # 12 12 8 8 8 % **Non: Strain # 12 12 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8	Time = Stress (psi) F(air) = Stress (psi) F(air) = Test Air Temperature (F) F(air) = Test Air Temperature (F) F(air) = Test Pepellant Temperature (F) F(air) = Test Air Temperature (F) F(air) = Test Temp = 7.1 F F(air) = Test Temp = 12.12.8.8 F(air) = Temp = 12.12.8.8 F(air) = Test Temp = 12.12.8.8 F(air) = Test Temp = 12.12.8.8 F(air) = Temp = Temp = 12.12.8.8 F(air) = Temp = Tem	NITIONS Time From Start of Test (win)

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WHILE STEP	
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Test No. 10 - Stress While Complex Straining for UTP-19,360B-400/1777 Figure 25.

TEST NO. 10 - 1/2-IN. BAR STRESS WHILE COMPLEX STRAINING TABLE 14.

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	TE: S/15/81 ERÁTOR: JWD	LATIONSHIPS: o = Force/Area f = Sample Extension/Length MINAL VALUES: Test Temp = 71 F Gage Length = 6.00 in Nom. Strain = 128,12,4,8,0,4,0 % Nim. Strain = 51n/min	3 6 · 119 0 · 092	0.249	CONTRACTOR OF THE TOTAL OF THE
1 08 2)	Æ.	NOW.	SAMPLE 2 6.098 0.013	0.251	04004000 400040 600040 600040 6000600 6000600 6000600 6000600 6000600 6000600 6000600 6000600 60
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	PROPELLANT: UTP 19360B 400/1777 REQUESTOR: Carlton WOR:	DEFINITIONS: Time = Stress (USI)	CALIBRATION DATA: Cal Wt = 5.0 lbs Load Cal (lbs/volts) Offset (volts) Pot Cal (in/volts) = Temp (F)	AREAS (sq in):	STRESS DATA (psi): 2

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TABLE 15. TEST NO. 11, PART 1 - QUINLAN COMPLEX HISTORY FOR UTP-3001

C·	ycle	Rate, in./min.	T7859 Remarks
			
1-7	Load Unload	2 2	Approximately 15 min. rest after cycle Approximately 15 min. rest after cycle
8	Load Unload	1 1	Approximately 15 min. rest after cycle
9	Load Unload	5 5	Approximately 15 min. rest after cycle
10	Load Unload	0.5 0.5	Approximately 15 min. rest after cycle
11	Load Unload	10 10	Approximately 15 min. rest after cycle
12	Load Unload	0.2 0.2	Approximately 15 min. rest after cycle
13	Load Relax Unload	2 1/2 hr 2	Approximately 30 min. rest after cycle
14	Load Relax Unload	2 1 hr 2	4 day rest after cycle
15	Load Relax Unload	2 1 hr 2	7 day rest after cycle

The later part of the cycling is represented only by the maximum and minimum stress-strain points. Part 3 of this test, the balance of cycling to failure, is recorded in Table 19 as maxima and minima for selected cycles sufficiently close to describe the upper and lower bounds. A plot of the data would be similar to Figure 32. The strain values in Table 19 are stable, while the maximum stress shows a continual decay, and compressive (negative values) stresses become less compressive.

(Text continued on page 76)

TABLE 16. TEST NO. 11, PART 1 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 1 OF 11)

STATE OF THE PROPERTY OF THE P

DATE: 1/4/82 (PERATOR: JWD	KPLATIONSHIPS; office/Area ffice/Area ffice/Area KIMINAL VALUES; Test Temp # 70 f Gage Length # 6.00 in Nom. Strain # 3.6,8 % XHD Rate # 2.in/Min		PLE Ava St	255 275 275 275 275 275 275 275	24 25 34 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	82 5 82 0 64 0 64 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0
	of Test (min) ure (F) emperature (F)	SAMPLE 0.000 0.000 70.0	0.251 .ain	1087 1087 1083 1083 1083 1083 1083 1083 1083 1083	- Now who would not have a second not have a sec	67 33 83
PROPELLANT: UTP 3001 750/7768 REQUESTOR: Carlton	DEFINITIONS: Time From Start of G = Stress (psi) T(air) = Test Air Temperatu T(prop) = Test Propellant Te	CALIERATION DATA; Cal Wt = 5.0 lbs Load Cal (lbs/in) Offset (in) Temp (F) =	sq in): DATA (psi):		24 - 1 - 1 - 1 - 1 - 1 - 1 - 1 - 1 - 1 -	3 0 1500

TABLE 16. TEST NO. 11, PART 1 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 2 OF 11)

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TABLE 16. TEST NO. 11, PART 1 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 8 OF 11)

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TABLE 16. TEST NO. 11, PART 1 - 1/2-IN. BAR STRESS WHITE OWN TWO TABLE 16. TEST NO. 11, PART 1 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 9 OF 11)

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TABLE 16. TEST NO. 11, PART 1 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 10 OF 11)

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TABLE 16. TEST NO. 11, PART 1 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 11 OF 11)

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TABLE 17. TEST NO. 11, PART 2 - QUINLAN COMPLEX HISTORY FOR UTP-3001 T8714

C	ycle	Rate, in./min.	Remarks
16	Load	0.02	
	unload	0.02	Approximately 30 min. rest after cycle
17	Load	0.02	
	unload	0.02	Approximately 30 min. rest after cycle
18	Load	0.05	
	relax	3 hr	
	unload	0.05	2 weeks rest after cycle
19	Load	0.02	
	unload	0.02	Approximately 30 min. rest after cycle
20	Load	0.02	
	unload	0.02	Approximately 30 min. rest after cycle
21	Load	0.05	
	relax	3 hr	
	unload	0.05	1 month rest after cycle
22-42	Cycling	5	Several cycles monitored followed by several with only maximum and minimum recorded.

Note: Part 3 of this test was continuing cycling to failure. The maximum and minimum values at larger intervals have been tabulated but not incorporated into the disk files.

## 3.1.12 Similitude Test No. 12

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This similitude test and the one following it were run with the intent that only the strain-time history would be supplied to the subcontractors, who would then predict the stress-time histories from their respective predictive theories for the program review meeting held at the end of the phase II testing in September 1982.

This similitude test on the 6-in. bar specimens of UTP-3001 and UTP-19,360B propellants was: (1) a 0.01 in./min. ramp to 10% strain followed by relaxation; (2) a 1 in./min. unload to 5% strain and relaxation; and (3) 0.1 in./min. ramp to failure. The same test was repeated for ramp rates of 0.001, 0.1, and 0.01 in./min. The data as reported at the September meeting are shown for UTP-19,360B in Figure 33. The ramp rates were 0.01, 1, and 0.1 in./min. Tabular data for the test are given in Table 20.

TABLE 18. TEST NO. 11, PART 2 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 1 OF 10)

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TABLE 18. TEST NO. 11, PART 2 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 2 OF 10)

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TABLE 18. TEST NO. 11, PART 2 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 3 OF 10)

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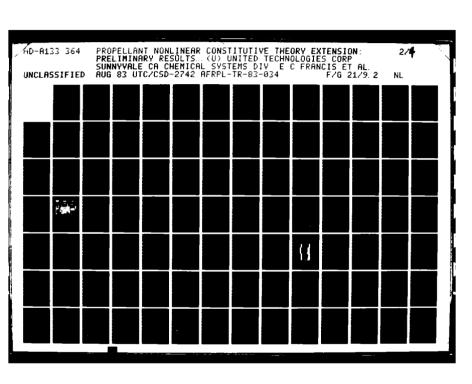
TABLE 18. TEST NO. 11, PART 2 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 4 OF 10)

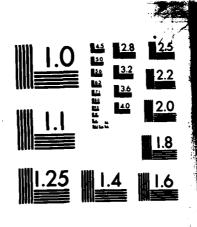
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TABLE 18. TEST NO. 11, PART 2 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 5 OF 10)

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TABLE 18. TEST NO. 11, PART 2 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 6 OF 10)

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TABLE 18. TEST NO. 11, PART 2 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 7 OF 10)

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TABLE 18. TEST NO. 11, PART 2 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 8 OF 10)

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TABLE 18. TEST NO. 11, PART 2 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 9 OF 10)

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TABLE 18. TEST NO. 11, PART 2 - 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 10 OF 10)

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TABLE 19. TEST NO. 11, PART 3 - UTP-3001-750/7768 1/2-IN. BAR STRESS WHILE CYCLING

(SHEET 1 OF 2)

T8715

Time	Strain	Stress	Remarks
74388.0479	2.40	-6.37	End of cycle 20
74390.3479	12.17	93.23	Peak of cycle 30
74390.4539	2.40	-5.98	End of cycle 30
74391.3819	12.17	92.03	Peak of cycle 34
74391.4239	2.40	<b>-5.58</b>	End of cycle 34
74391.9432	12.17	91.24	Peak of cycle 40
74391.9859	2.40	<b>-5.5</b> 8	End of cycle 40
74392.0799	12.17	89.64	Peak of cycle 50
74392.1219	2.40	-5.18	End of cycle 50
74393.0099	12.17	88.45	Peak of cycle 60
74393.0519	2.40	-5.18	End of cycle 60
74394.7799	12.17	84.86	Peak of cycle 80
74394.8219	2.40	-4.78	End of cycle 80
74395.7279	12.17	80.10	Peak of cycle 102
74395.7479	2.40	-4.78	End of cycle 102
74418.3479	12.17	76.49	Peak of cycle 164
74418.3979	2.40	-4.78	End of cycle 164
74451.5979	12.17	71.71	Peak of cycle 330
74451.8779	2.40	-4.38	End of cycle 330
74474.2779	12.17	68.92	Peak of cycle 442
74474.3179	2.40	-3.98	End of cycle 442
74698.0179	12.17	62.95	Peak of cycle 980
	2.40	-3.59	End of cycle 980
74928.0179	12.17	60.16	Peak of cycle 198
	2.40	-3.59	End of cycle 1980
75158.0179	12.17	56.97	Peak of cycle 298
	2.40	-3.59	End of cycle 2980
75388.0179	12.17	56.18	Peak of cycle 398
· · = · · · ·	2.40	-3.59	End of cycle 3980

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TABLE 19. TEST NO. 11, PART 3 - UTP-3001-750/7768 1/2-IN.
BAR STRESS WHILE CYCLING

(SHEET 2 OF 2)

T8715

Time	Strain	Stress	Remarks
75618.0179	12.17	53.39	Peak of cycle 4980
	2.40	<b>-3.59</b>	End of cycle 4980
75695.5179	12.17	51.39	Peak of cycle 5317
	2.40	-3.59 SAMPLE BROKE	End of cycle 5317

## 3.1.13 Three-Step Relaxation Test No. 13

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The three-step relaxation tests were run as similitude tests in which the strain-time data were reported to the subcontractors and they were to predict the stress-time histories. The test consisted of loading 6-in. bar specimens at 0.05 in./min. crosshead rate to 10%, relaxing 1 hr, unloading at the same rate to 7% strain, relaxing 1 hr, and repeating the process at 3% strain, then unload. The tests were then repeated with nominal 24 hr relaxation periods for both UTP-3001 and UTP-19,360B propellants.

These data were reworked to make sure that peak and minimum stress points were included in the data. Relaxation was monitored after the sample was unloaded to zero stress. Data for UTP-3001 is shown in Figure 34 with the 1 hr relaxation periods. Tabular data are given in Table 21.

(Text continued on page 103.)

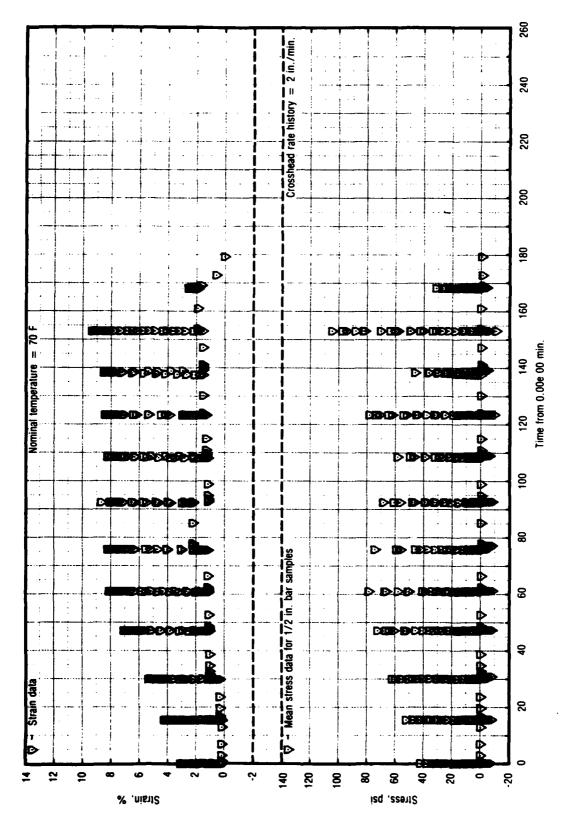
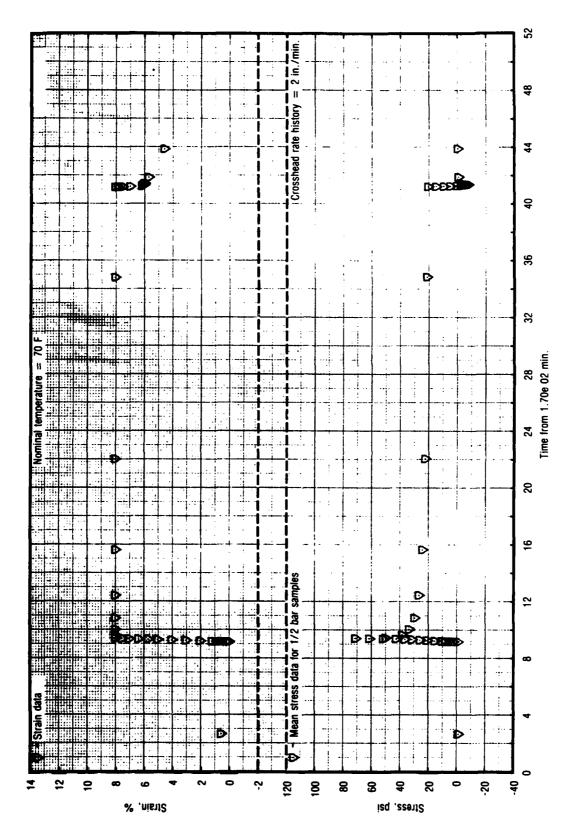
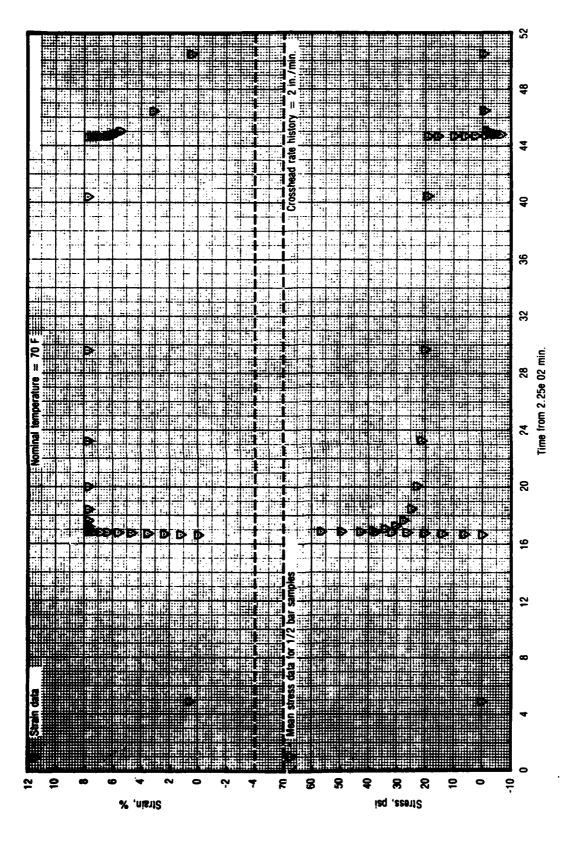


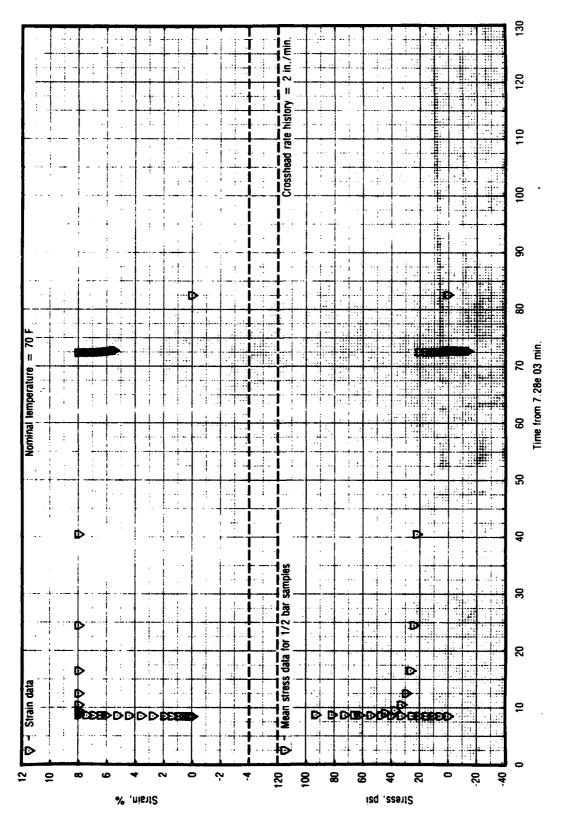
Figure 26. Test No. 11, Part 1 - Stress While Cycling for UTP-3001-750/7768



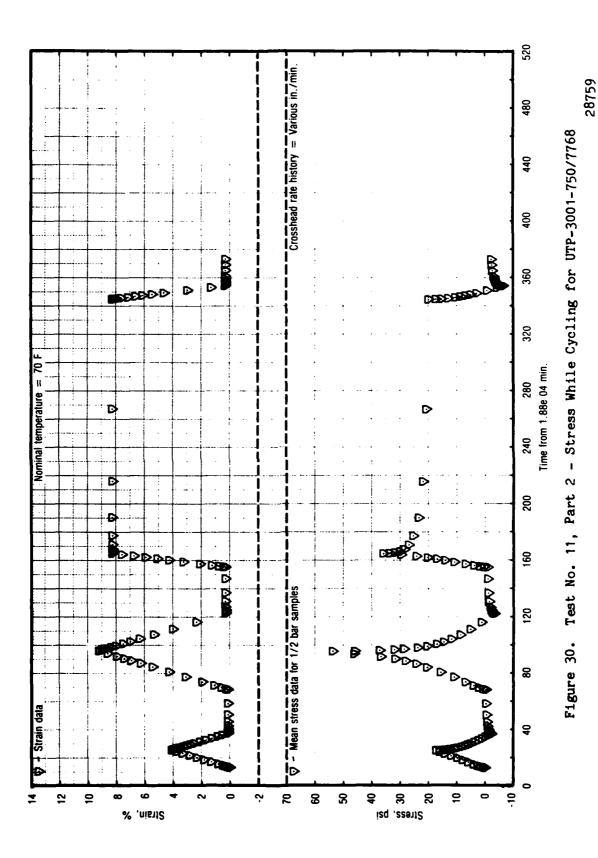
Test No. 11, Part 1 - Stress While Cycling for UTP-3001-750/7768 Figure 27.

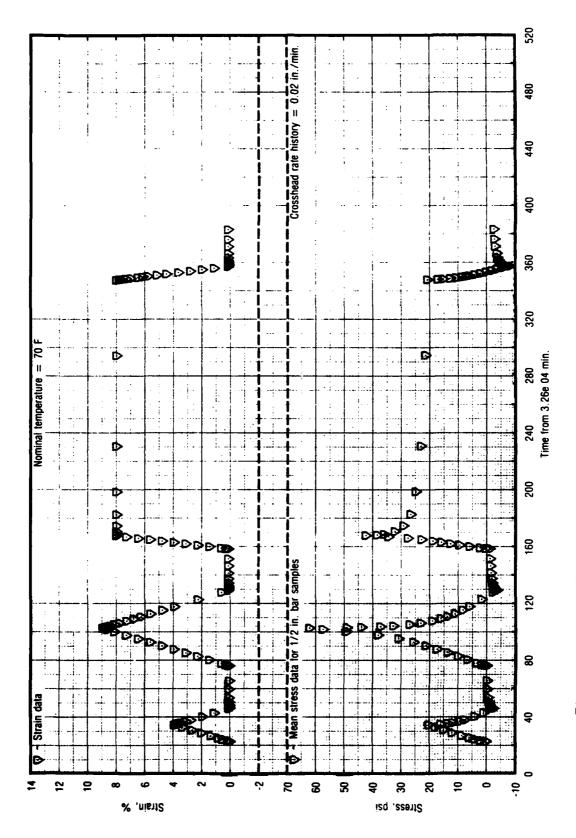


Test No. 11, Part 1 - Stress While Cycling for UTP-3001-750/7768 Figure 28.

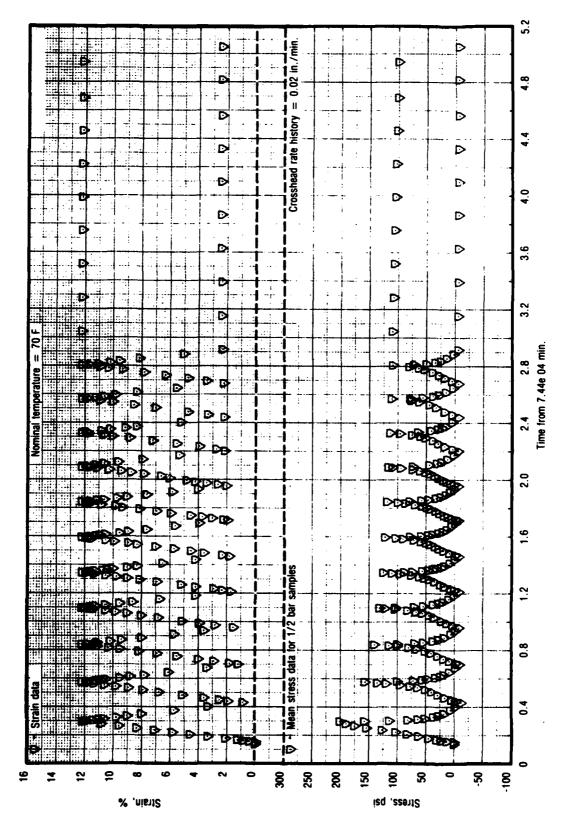


Test No. 11, Part 1 - Stress While Cycling for UTP-3001-750/7768 Figure 29.

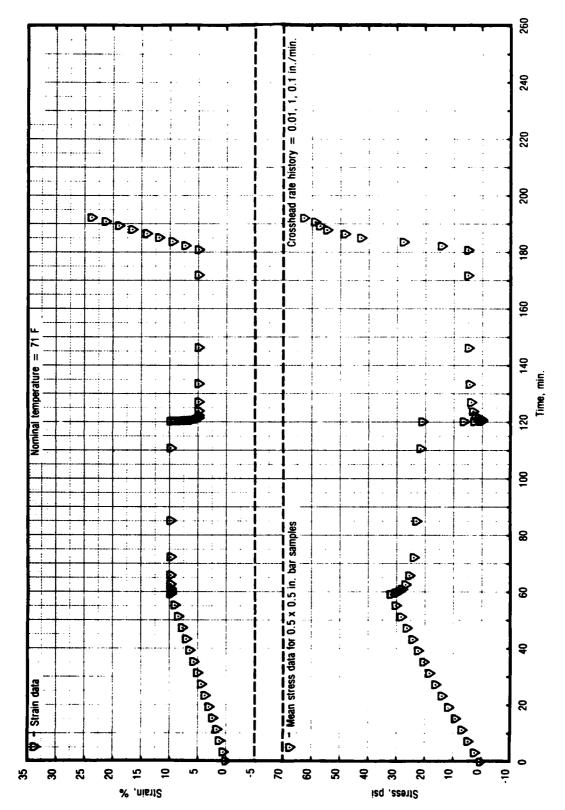




Test No. 11, Part 2 - Stress While Cycling for UTP-3001-750/7768 Figure 31.



- Stress While Cycling for UTP-3001-750/7768 28761 N Test No. 11, Part Figure 32.



Test No. 12 - Stress While Step Straining for UTP-19,360B-400/1777 Figure 33.

TABLE 20. TEST NO. 12 - 1/2-IN. BAR STRESS WHILE STEP STRAINING (SHEET 1 OF 2)

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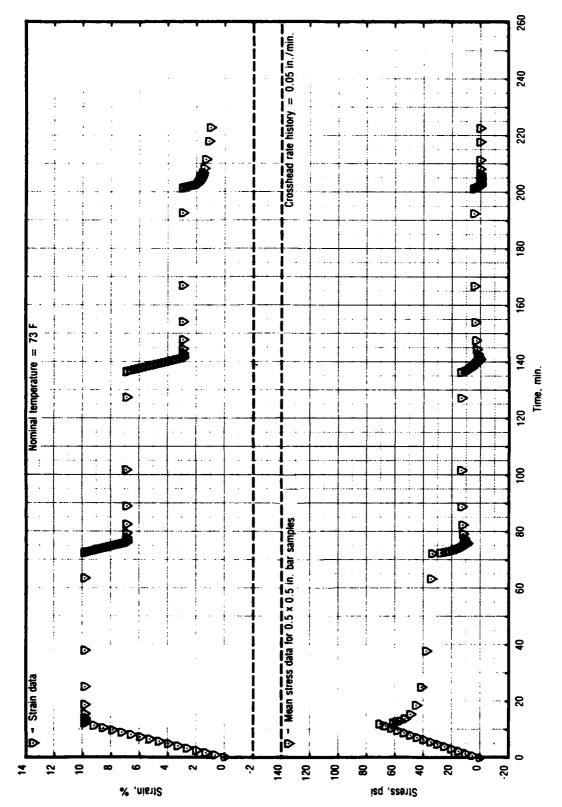


Figure 34. Test No. 13 - Stress While Step Straining for UTP-3001-750/7768

TABLE 21. TEST NO. 13 - 1/2-IN. BAR STRESS WHILE STEP STRAINING (SHEET 1 OF 3)

DATE: 6/5/81 (IPERATOR: JWD

PROPELLANT: UTP 3001 750/7768 REQUESTOR: Carlton WOR:

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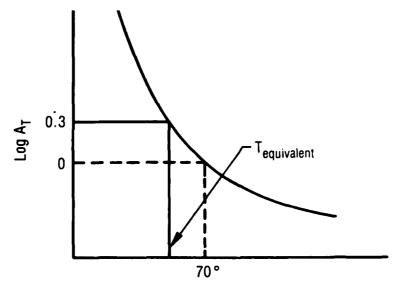
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# 3.1.14 Propellant Aging Effects During Phase II Testing

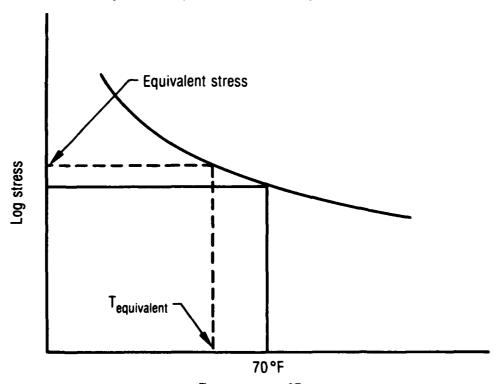
The numerous and complex tests involved in the uniaxial-isothermal evaluation of the two propellants (UTP-3001 and UTP-19,360B) covered an approximate 3-month time period. Since the 6-in. bar specimens at constant rate (test No. 1) were run first, they were used as the standard to compare the balance of the data.

Stress-strain plots of UTP-3001, test No. 1 are shown in Figures 35, 36, and 37 for 41, 75, and 124°F tests, respectively. Crosshead rates of 10, 1, and 0.1 in./min. are shown for each temperature with the ambient tests down to 0.001 in./min. The same data are shown in Figures 38, 39, and 40 for UTP-19,360B. Comparison of the other tests was based on the initial ramp loading or undamaged state. Test No. 6 was a full, one-half and one-quarter load creep test with crosshead loading rate of 1 in./min. The peak stress-strain points, when the crosshead stopped, were compared to the constant rate plots as shown in Figure 35, etc. All the data are given in Tables 22 and 23 for UTP-3001 and UTP-19,360B, respectively. When crosshead rates did not match, a time-temperature equivalent stress value was selected for comparison purposes. These data were considered approximate. Stress values that were considered reliable (i.e., very close) were marked in the tables with a single approximate sign (~) while those that left some doubt due to the shift were marked with a double approximate ( $\approx$ ). Some tests were run at 5 in./min. rather than the 10 in./min. comparison data. The 10 in./min. equivalent of the 5 in./min. data can be determined from the rate shift (i.e.,  $\log A_T = \log 10/5 = 0.30$ ). The WLF curve (in text figure) is used to pick off the equivalent temperature. The high rate stress value being sought is equivalent to a lower temperature at the lower rate (5 in./min.).



Temperature, °F

The equivalent temperature is then used to obtain the equivalent stress value from a stress-temperature plot (in text Figure).



Temperature, °F

The stress-temperature plot can also be used to obtain equivalent stress values when the temperatures do not match at the same rate.

After the initial ramp loading, the propellant was considered damaged. In most instances the tests were of short enough duration to neglect any aging effect during test. Since bulk storage at controlled ambient conditions has not shown a significant aging effect over 3 months, this was also neglected. Rather it was lumped into the between carton difference. The sample handling procedure established was to machine one carton at a time, hold all samples overnight in a nitrogen flushed dry box (<10% RH) before testing, and then test all samples from the dry box before machining additional specimens. The data in Table 22 for UTP-3001 indicate little difference between boxes 1 and 2 but a one-third higher average stress in box 3. The other significant change was within a box due to the amount of storage time in the dry box. During the 13 days the residual of box 2 was in the dry box, the stress changed from 30% below the mean to 10% above the mean. The UTP-3001 tests from box 3 have been replaced with another carton which was sample tested before doing the program tests. This provided a reasonable assurance that the results would be of the same family as the original constant rate data (test No. 1). The data from UTP-19,360B in Table 23 do not show as much effect for dry box storage or between box differences compared to the UTP-3001 propellant.

Both propellants have indicated that dry box storage will increase the stress capability. This is assumed to be due to loss of moisture and it is reversible as is shown in Figure 41 for UTP-3001. In this example, specimens held in the dry box for 4 months were exposed to 58 and 85% relative humidity for 1 week. All six samples were tested simultaneously in a CSD multistation tester.

The one test which may need some special treatment is the Quinlan complex history No. 11 (see section 3.1.11). After 12 cycles and 2 relaxation periods, the samples were removed for 4 days storage in a dry box. They were then followed by (1) relaxation - 7 days in a dry box, (2) two cycles - a relaxation period - 2 weeks in a dry box, (3) two cycles - a relaxation period - 1 month in a dry box, and (4) cycling to failure. At this point it is not clear just how to separate the stress increase due to rehealing from the dehydration effect of

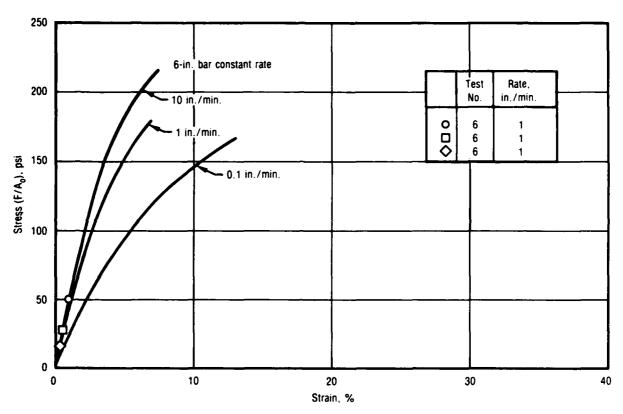


Figure 35. Initial Ramp for UTP-3001-750/7768 39°F Tests Compared to the 6-in. Bar Constant Rate Data

the dry box particularly for the 1 month storage period. The test times at approximately 50% RH were too short to have any significant effect on propellant stress capability. Both propellants showed reasonable correspondence of the initial ramp to the reference constant rate data.

### 3.2 TWO-DIMENSIONAL AND VARIABLE TEMPERATURE INVESTIGATION

The biaxial and nonisothermal testing was conducted on specimens of UTP-3001 and UTP-19,360B propellants as detailed in Figure 42. The biaxial samples were cast into prelined redwood boxes with a 1.25-in. gage length by 6-in. wide and machined flat to a 0.25-in. thickness. The response properties rather than failure properties were of interest so the discontinuity at the redwood interface did not affect the desired behavior. The  $1/2 \times 1/2 \times 6$ -in. specimens were used for straining-cooling and cyclic strain tests. Shear relaxation tests

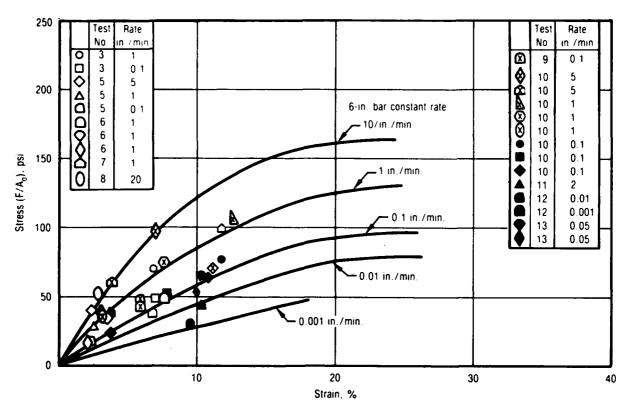


Figure 36. Initial Ramp for UTP-3001-750/7768 75°F Tests Compared to the 6-in. Bar Constant Rate Data

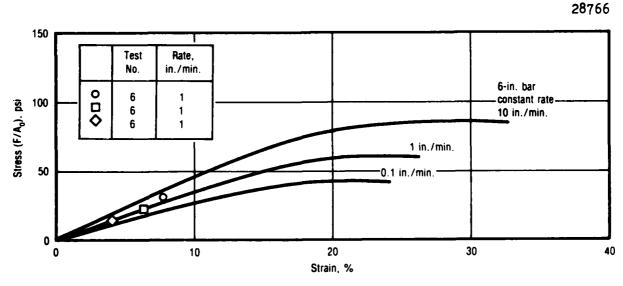


Figure 37. Initial Ramp for UTP-3001-750/7768 124 $^{\rm O}$ F Tests Compared to the 6-in. Bar Constant Rate Data

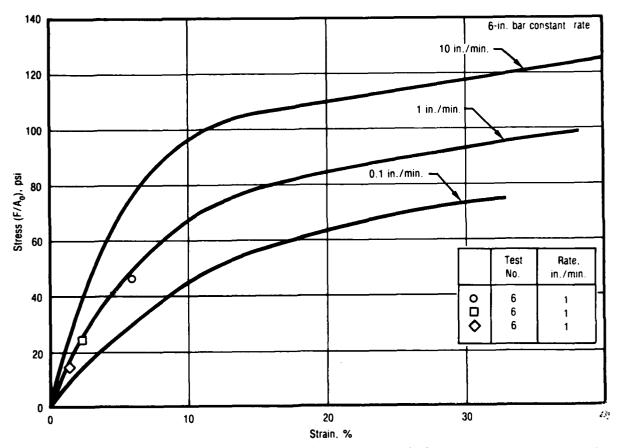


Figure 38. Initial Ramp for UTP-19,360B-40C/1777 40°F Tests Compared to the 6-in. Bar Constant Rate Data

were run with  $1 \times 1 \times 3$ -in. specimens bonded directly to steel anvils. Details are given in later sections. The equipment discussed in section 3.1 was utilized for this testing; however, only three biaxial specimens could be tested at once because of space limitations in the oven.

The biaxial specimens used in this part of the program were cast into redwood boxes similar to that shown in Figure 5. The space between redwood blocks was 1.25-in. instead of the 6-in. for the uniaxial bars. A mill finished specimen is shown in Figure 43. The propellant was left flat rather than necking it down as with standard JANNAF biaxial specimens. The gage length was designated at the wood to wood distance for strain evaluation. Response properties rather than failure properties were of interest so the boundary

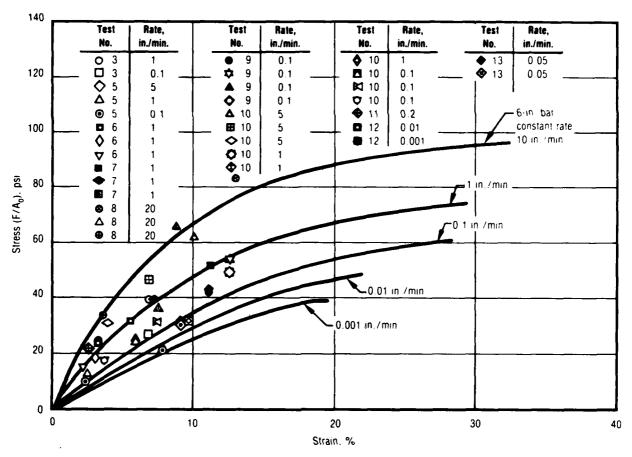


Figure 39. Initial Ramp for UTP-19,360B-400/1777 70°F Tests Compared to the 6-in. Bar Constant Rate Data

perturbation was neglected. Some data taken from the literature⁽²⁾ have been reproduced for evaluation of the stress-strain behavior across the biaxial field (see Figures 44 through 47).

The shear samples (Test No. 17) were 1 x 1 x 3-in. blocks of propellant that were bonded to the test fixture (shown in Figure 48) after being machined. The pull rods were attached to the offset plates so that the load was transmitted through the center of the sample as shown. Since strain was limited to 5% for the shear relaxation test, the sample was assumed to be in simple shear. The shear strain (7) was calculated as the tangent of

Reference 2 - Jones, J., "Solid Propellant Structural Integrity Investigations: Dynamic Response and Failure Mechanisms in Solid Propellants," RPL-TDR-64-32, Vol. I, Lockheed Propulsion Co., February 1964.

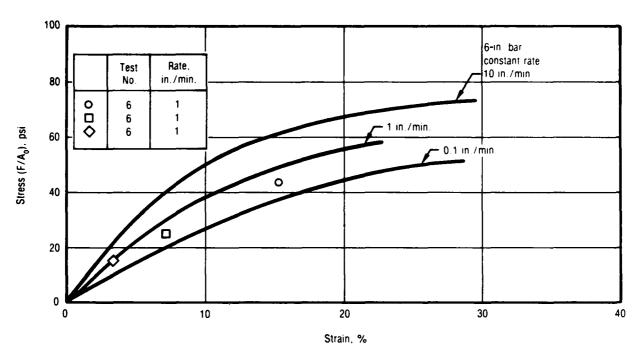


Figure 40. Initial Ramp for UTP-19,360B-400/1777 123°F Tests Compared to the 6-in. Bar Constant Rate Data

the displacement angle or  $\Delta L/G.L.$  The shear stress (7) was calculated as force/area (area = 3 sq. in.).

The data modification to insert peak and minimum stress points previously discussed were utilized for the biaxial and nonisothermal tests.

#### 3.2.1 Biaxial Constant Rate Test No. 14

The biaxial constant rate tests to failure were run with the  $1/4 \times 1-1/4 \times 6-in$ . specimens of UTP-3001 and UTP-19,360B. The 40, 70, and  $120^{\circ}F$  tests were run at crosshead rates of 2, 0.2, and 0.2 in./min. The test equipment, with specimens in place, is shown in Figure 49. A typical load-time curve is shown in Figure 50 for UTP-19,360B at  $71^{\circ}F$  and 2 in./min. crosshead rate. Because of the fixtures and more difficulty in adjusting linkage than with the 6-in. bar specimens, the three samples did not start loading simultaneously. Sample 2 was adjusted to an effective zero and is shown in Figure 51. Tabular data are given in Table 24.

TABLE 22. STRESS-STRAIN COMPARISON FOR UTP-3001 TESTS (SHEET 1 OF 2)

8	Remarks	Mean	+4.2%		Box 2	Box 2	Mean	over box													
T8718	After Machin- ing	11	15	5	-	<i>ا</i> به	n ve	9	7	7	∞ (	<b>∞</b>	6	6	σ	σ;	2 5	7 t	i 5		
01 TESTS	Box No.	1	-	-	~	ο ο	v ~	· ~	8	7	8	~	8	7	7	~	N C	, v	ı ~		m
FOR UTP-3001	Differ- ence,	13	7	-	-16	<b>6</b> ر	را 4.0	~	1	0	œ	-	5	6	ထ		Ն ե	5 =	: 7		οţ
MPARISON 1 OF 2)	Stress, psi Test No. 1	<b>~</b> 36	27.5	16.5	22	37	22.5	15	27	20	77	5	~ 87	~ 55	86	68.5	35 66 76	47	54		17 CF
S-STRAIN CC (SHEET	Stress, psi	99°0†	27.28	16.70	47.8	33.62	22.40	15.3	30.1	50.0	25.8	15.2	96.1	59.8	106	76.3	30°.7	52.0	23.6		42.69
22. Stres	Strain,	2.40	2.47	2.36	5.94	3.54	6.29	4.02	7.71	1.00	84.0	0.35	η6·9	3.78	12.55	7.53	3.29	7.83	3.72		6.91
ABLE	Rate, in./min.	5	-	 	-			<b>-</b>	-	<b>-</b>	<del>-</del>	-	2	Ŋ	-	<b></b> 1	- c		0.1		- 0
	Temp- erature, F	Ambient	Ambient	Ambient	Ambient	Ambient	120	120	120	<b>9</b>	<b>9</b>	0#	Ambient					-•	<b>&gt;</b>	Ambient	Ambient
	Test No.	5			9								10								m
₽									1	111	ļ										

TABLE 22. STRESS-STRAIN COMPARISON FOR UTP-3001 TESTS (SHEET 2 OF 2)

T8718

Test No.	Temp- erature, R	Rate in./min.	Strain	Stress, psi	Stress, psi Test No. 1	Differ- ence,	Box No.	Days After Machin-	Remarks	rks
9	Ambient Ambient	1.0	6.63	41.8 98.9	41	1.9	<b>=</b> =	2 2		
13	Ambient Ambient	.05	10.81	64.3 56.2	~ 56 ~ 53	14.7 5.9	ন ন	22		
12	Ambient Ambient	0.01	10.40 9.82	40.5 29.9	46 27	-12	<b>a a</b>	ოო		
=======================================	Ambient	8	3.08	41.8	~ 36	41-	-	m		
Notes:	1. Box 1 us	1 used up	16 days aft	er machini	ed up 16 days after machining; test 5 mean = 4.2% over constant rates tests.	ean = 4.2\$	over	constant ra	ates t	ests.

Box 1 used up 16 days after machining; test 5 mean = 4.2% over constant rates tests. Box 2 used up in 13 days after machining; box 2 mean = 1.1% over box 1 constant rate but it changed from -30 to +10% during the 13 days. . .

TABLE 23. STRESS-STRAIN FOR UTP-19,360B TESTS (SHEET 1 OF 2)

								2	61.101
Test No.	Temp- erature, F	Rate, in./min.	Strain,	Stress, psi	Stress, psi Test No. 1	Differ- ence,	Box No.	Days After Machin- ing	Remarks
2	Ambient Ambient	5	2.49	21.8	~24 17.5	-9 -26		11 51	Mean -8.0≸ below
,	Ambient	0.1	2.36	10.54	9.5	=	<del>-</del>	5	test 1
٥	Ambient Ambient	<del>-</del> -	5.50 3.08	32.0 18.05	31.5 20.5	2 <del>7</del> -	∾ —	- 2	Box 2 mean
	Ambient 120		2.24 15.29	14.76	16	8 <del>7</del> 7		n o n	-2.7% below box 1
	120 120 120 120 120 120 120 120 120 120		2.03 2.03 2.05 2.05 3.05 3.05 3.05 3.05 3.05 3.05 3.05 3	25.0 46.13 24.26	29.6 50.5 24.5	5 6 8 7		0 ~ ~ & •	
5	40 Ambient	- r	1.35 6.94	14.0	16 50 5	-12 -7	~ ~	<b>დ</b> თ	
		W000	3.78 2.55 3.29 1.06 7.38	30.8 26.7 36.7 24.3 31.2	~33 54.5 39 27 27	7692255		8 8 8 2 2 2 C	
	Ambient		10.11	62.0	<b>~</b> 62	jo	- 0	5	
m	Ambient Ambient	0.1	7.00	39.57 27.2	37.5 25.5	9	m m		Box 3 mean +7.6\$ above box 1
<b>6</b> 0	Ambient Ambient Ambient	20 20 20	13.04 8.46 3.61	85.1 66.7 32.9	~ 85 ~ 68 ~ 40	182	<b>-</b> m m m	- u m	

TABLE 23. STRESS-STRAIN FOR UTP-19,360B TESTS (SHEET 2 OF 2)

BOOKS CONTRACTOR SANGERS CONTRACTOR SANGERS

					(SHEET 2 OF 2)	<b>.</b>			T8719
Test No.	Temp- erature, F	Rate, in./min.	Strain,	Stress, psi	Stress, psi Test No. 1	Differ- ence,	Box No.	Days After Machin- ing	Remarks
6	Ambient	0000	11.07 12.53 5.95 5.95	4.0 4.3 4.8 4.8 4.0	37.7 41.2 22 22	01 5 13 01	ю r	0 13 4 4	
7	Ambient Ambient Ambient		11.28 7.09 3.27	51.8 38.5 23.6	51.5 37.5 21.5	0 20	. നനന	र र र	
13	Ambient Ambient	0.05	9.21	32.1 30.9	~31 ~30.5	<b>#</b> F	ოო	16 20	
51	Ambient Ambient	0.01	9.68	31.8	28.5 20	12 0	ოო	26 41	
=	Ambient	8	3.03	23.2	<b>₹</b>	01	•	m	

are high the absolute differences are not that great. There is an indication of drift from low to high during testing with box 2. All of these data appear to be the same family with a few exceptions. While the percentages Note:

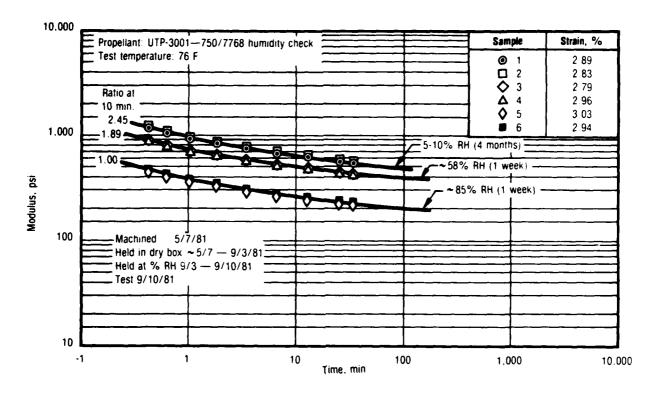


Figure 41. 1/2-in. Bar Stress Relaxation Data at 3% Nominal Strain 28773

### 3.2.2 Biaxial Straining-Cooling Test No. 15

Biaxial specimens of UTP-3001 and UTP-19,360B propellants were simultaneously strained and cooled from 115 to  $40^{\circ}$ F at a crosshead rate of approximately 3 x  $10^{-5}$  in./min. over a 40-hr period. The results for UTP-3001 are shown in Figure 52. The stress-time traces for all three samples appear to start together but spread out as the test progresses. The tabular data are given in Table 25.

### 3.2.3 Biaxial Stress Relaxation Test No. 16

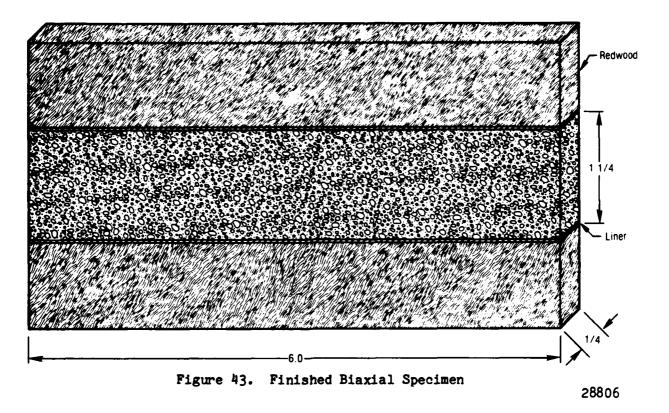
Biaxial stress relaxation tests were run with the  $1/4 \times 1-1/4 \times 6-in$ . specimens of UTP-3001 and UTP-19,360B propellants at a nominal 3% strain and temperatures of 40, 70, and 120 F. The  $40^{\circ}$ F data for UTP-19,360B are shown in Figure 53 as typical with good reproducibility. The loading ramp rate was 0.2 in./min. Data are given in Table 26.

Test	Test Description	Damage Cycle/Test	Strain Cycle
14	Biaxial constant rate	Biaxial samples of UTP-3001 and UTP-19,360B were ramp loaded to failure at rates of 2, 0.2, and 0.02 in./min. at temperatures of 41, 70, and 120°F.	· /
15	Biaxial straining- cooling	Biaxial samples of UTP-3001 and UTP-19,360B were simultaneously strain and cooled from 120 to 40°F over a 40 hr. period.	· T
16	Biaxial relaxation	Biaxial samples of UTP-3001 and UTP-19,360B were run in stress relaxation tests at 40, 70, and 120°F.	,
17	Shear relaxation	Shear samples of UTP-3001 and UTP-19,360B were run in stress relaxation tests at 70°F.	See above
18	6-in, bar straining- cooling	6-in. bars of UTP-3001 and UTP-19,360B were simultaneously strain and cooled from 120 to 40 °F at three slow rates.	, T
19	Biaxial Quinlan complex history	Biaxial samples of UTP-3001 and UTP-19,360B were cycled for the Quinlan complex history test at 70°F.	· [MV/M]
20	6-in. bar cyclic test	6-in: bars of UTP-3001 and UTP-19.360B were run in cyclic strain tests at 0.1 in./min, and 70°F.	* MMM/MMM/MM/M
21	Biaxial thermal similitude	Biaxial samples of UTP-3001 and UTP-19,360B were run in ramp- relaxation-ramp tests with simultaneous cooling or heating (i.e., for reverse ramp)	See above plus last half thermal cycled
	Note: Nominally three	e samples were run for each test and condition.	Legend: T = temperature t = time ε = strain

Figure 42. Biaxial and Nonisothermal Phase III Testing

# 3.2.4 Shear Relaxation Test No. 17

Shear relaxation tests were run on 1 x 1 x 3-in. samples of UTP-3001 and UTP-19,360B propellants. The samples were post bonded to steel plates and run one at a time by loading them at 0.2 in./min and ambient temperature with offset fixtures so the load was transmitted through the center line of the sample. The 3 samples for each propellant were hand reduced and digitized for computer storage and printout. A typical example is shown for UTP-3001 in Figure 54 with data given in Table 27. Peak stresses and strain were very close as was the 1 hour relaxation stress on each propellant even though the samples were run separately.



# 3.2.5 Straining-Cooling Multiple Rates Test No. 18

The rate effect on the straining-cooling response was determined on UTP-3001 and UTP-19,360B propellants. The 1/2 x 1/2 x 6-in. bar sample was used so that testing could be completed in the shortest time possible. The rate effect for the uniaxial specimens was then applied to the biaxial test No. 15. Cooling was from 110°F to 40°F at the crosshead rates of 0.002, 0.0002, and 0.0004 in./min. Typical data are shown for the 0.002 in./min. rate for UTP-3001 in Figure 55. Good reproducibility is shown within the set of 3 samples. Data are given in Table 28.

## 3.2.6 Biaxial Quinlan Complex History Test No. 19

The  $1/4 \times 1-1/4 \times 6-in$ . biaxial samples of UTP-3001 and UTP-19,360B were subjected to the complex cycling and relaxation history indicated in Figure 42. When the tests were run the linkage, misalignment, etc. was such that the samples were not loaded an equivalent amount. The first sample to be loaded had the correct strain determined but strain on the other samples had to be adjusted to the time the stress ramp started on each. Each sample was reduced separately

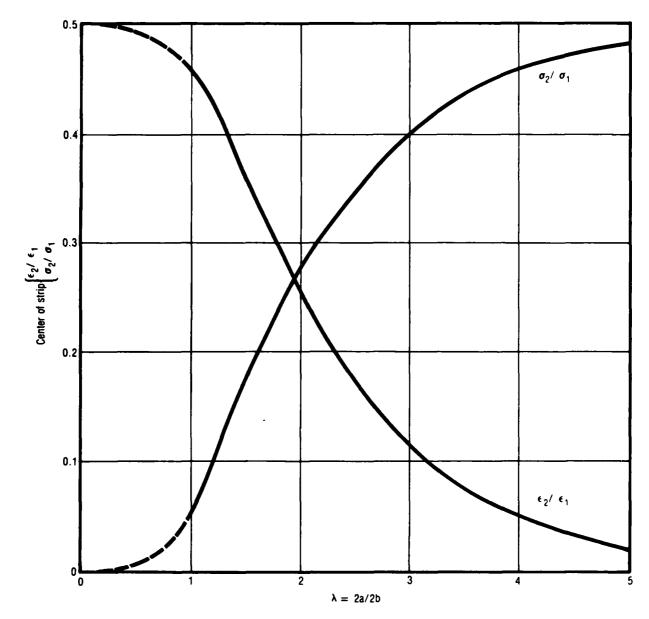


Figure 44. Principal Stress and Strain Ratios at the Center of Biaxial Strips for Varying Height-to-Width Ratios for a Poisson's Ratio of 1/2 28807

and data were modified to pick up the peak and minimum stress points. Sample 1 for UTP-3001 is shown in Figures 56 through 58 where the complex test has been divided into segments on an expanded time scale to show test details. The first cycle in Figure 56 showed no load and the second cycle showed very little. By contrast sample 3 (not included here) had a first peak stress-strain of 37 psi, 2.26% and second peak of 76 psi, 4.97%. During the unload part of the cycle

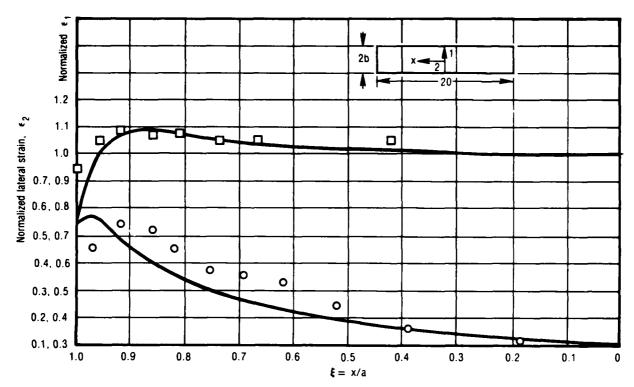


Figure 45. Normalized Axial and Lateral Strains Along the Midplane Biaxial Strip Specimen

after stress reached zero, strain decay was estimated from other tests where strain was measured by cathetometer. Data are given in Table 29.

#### 3.2.7 Cyclic Testing Test No. 20

The complex cyclic testing of UTP-3001 and UTP-19360B propellants was done with the 1/2 x 1/2 x 6-in. bar samples. These tests were run at 70% with cycling from 8 to 4% then 12 to 8% followed by 8 to 4% strain at a crosshead rate of 0.1 in./min. The strain levels were set so that the sample would not reach zero stress on the unload cycles. By doing that a correct evaluation of strain was obtained during the tests. Data for the UTP-19,360B receptlant are shown as typical in Figure 59 with digitized results given in 7. 12 30. This propellant broke without completing the nominal 30 cycles but UTP-3001 went all the way through the test. The nominal 10 cycles per segment was dependent upon when the time scale matched available personnel. It was 11 cycles for the first 2 segments for UTP-19360B as shown in Figure 59.

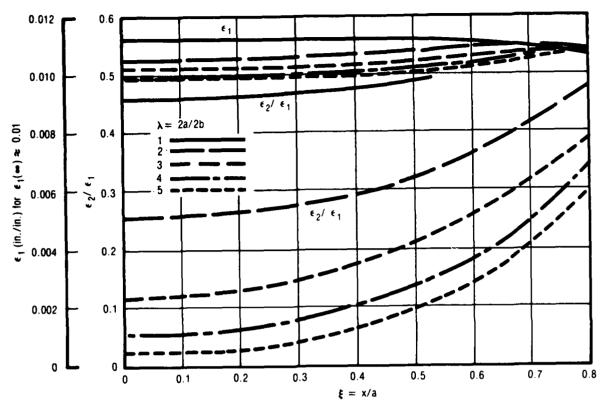


Figure 46. Strain Variations Along Midline of Strip Specimen (N = 0) for Poisson's Ratio of 1/2

# 3.2.8 Biaxial Ramp-Relax-Ramp Test No. 21

The ramp-relax-ramp tests were run on UTP-3001 and UTP-19,360B propellants with the  $1/4 \times 1$ - $1/4 \times 6$ -in. biaxial specimens. The first test was ramped at 0.0005 in./min. to 6% strain and simultaneously cooled from 120 to  $70^{\circ}F$ . It was held at 6% strain nearly 23 hours then ramp to failure while cooling towards  $40^{\circ}F$ . The data for UTP-3001 are shown in Figure 60 with data points given in Table 31 as shown in Figure 60. The cooling cycle did not end at the peak strain consequently the relaxation of stress was not the normal type behavior. The continued cooling increased the propellant stress capability so that the normal relaxation behavior did not start until the propellant temperature stabilized.

This test was repeated starting at 110°F and taken to 6% strain with a peak stress of 70 psi compared to 30 psi for the above test. The longer ramp time

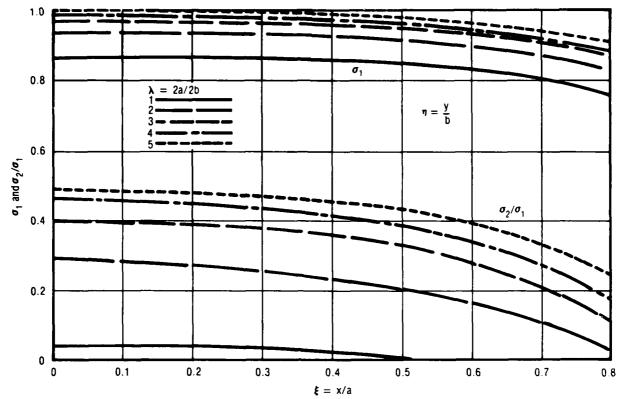


Figure 47. Stress Variations Along Midline of Strip Specimen (N = 0) for Poisson's Ratio of 1/2

allowed the cooling to reach 40°F at the peak stress. The samples were allowed to relax overnight and then unloaded to 3% while warming the samples to room temperature. Figure 60 was considered to be sufficient to represent the test.

#### 3.2.9 Propellant Aging Effects During Phase III Testing

The biaxial testing in Phase III was scheduled to minimize the dry box storage time after sample machining. The purpose was to reduce the within carton or box variability encountered during the uniaxial testing in Phase II. The majority of the testing was accomplished within 1 week of machining the samples. Those that went over a week did not seem to be influenced excessively.

The data obtained for initial ramp loadings (i.e., undamaged behavior) on the tests run are given in Table 32 for UTP-3001 and Table 33 for UTP-19,360B. The biaxial constant rate data for UTP-3001 are plotted in Figure 61. These

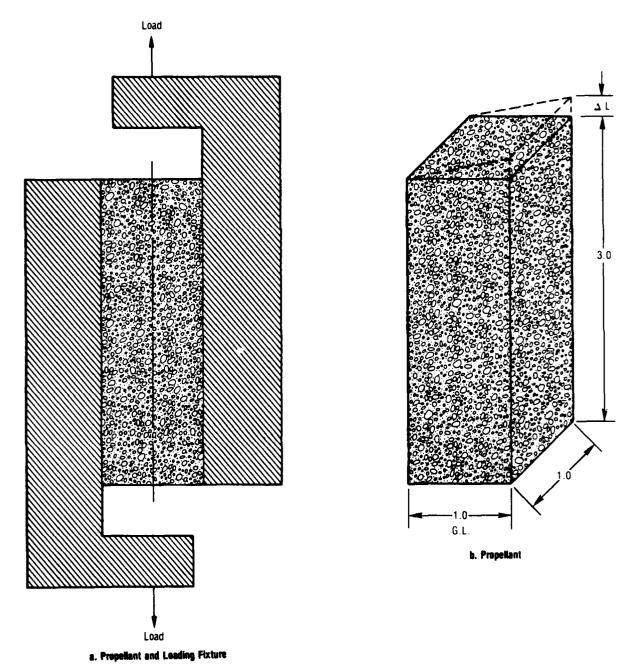


Figure 48. Shear Sample and Test Attachment

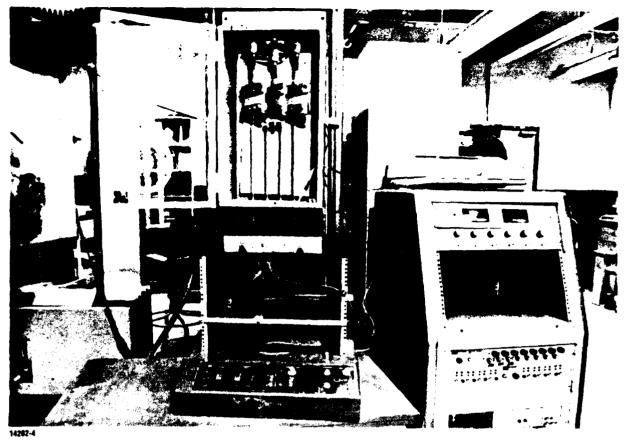


Figure 49. Biaxial Test Setup and Instrumentation

results were used to obtain comparison data for other tests at different strain levels in Table 32. The stress at 8% strain versus temperature is shown in Figure 62 for UTP-3001. The tests at different temperatures were taken from different redwood boxes of propellant and there appears to be some between box and sample differences. The data shifted for rate effects to the 0.2 in./min. in the lower plot indicates that extrapolation of the temperature stress plot would be unreliable. This eliminated any reasonable direct comparison with the slower rate straining - cooling tests as noted in Table 32. The comparisons did show reasonable agreement with some samples from each of the redwood boxes used. The straining-cooling data are expected to be of the same general family.

(Text continued on page 168.)

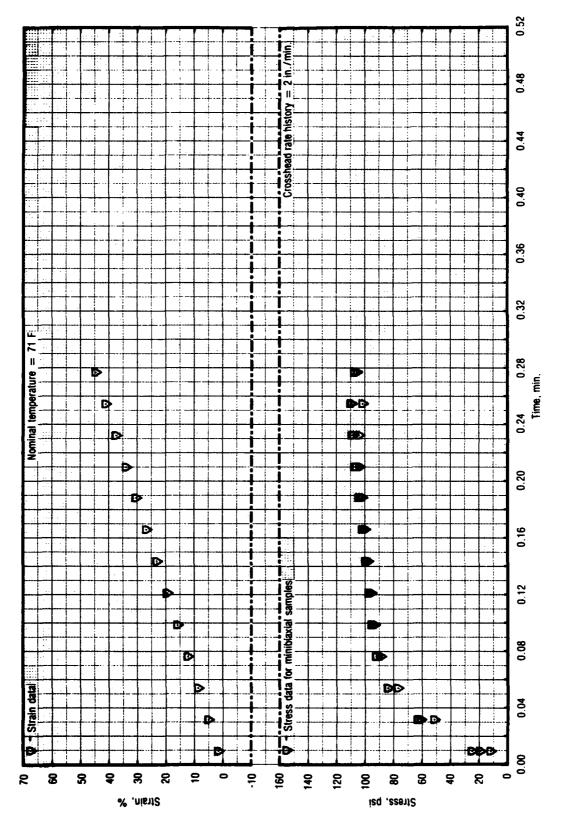


Figure 50. Test No. 14 - Stress for UTP-19,360B-400/1777

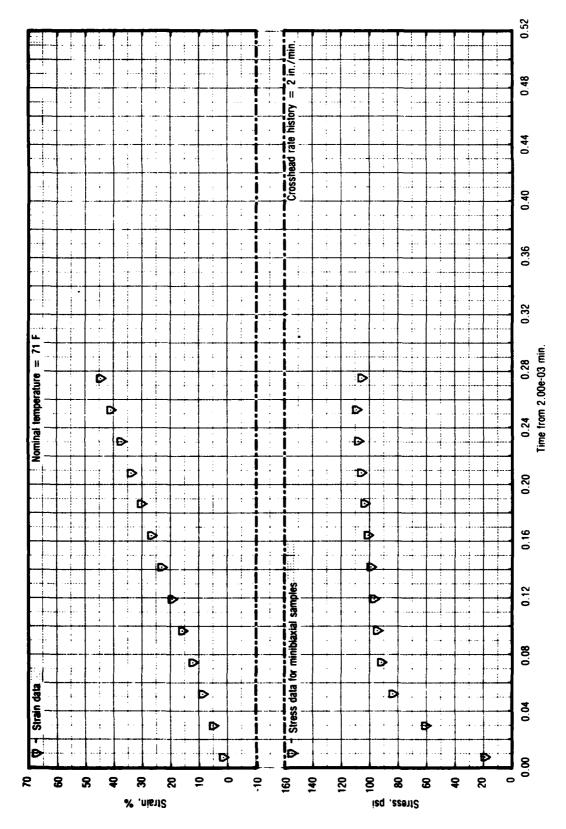


Figure 51. Test No. 14 - Stress for UTP-19,360B-400/1777

STRAINING AND COOLING DATE: 12/10/81 OPERATUR: JWD	RELATIONSHIPS;  = Force/Area = Sample Extension/Length NOMINAL VALUES;  Test Temp = 71 F Gae Length = 1.25 in Nom. Strain = 50 % XHD Rate = 2 in/min	20.016 0.177	1,512	\$\text{Ample 1} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text{Ample 2} & \text
STRESS WHILE STRAINI) DA	# 0X	19.991 19.980 0.101 0.170 -0.393 0.170	1.537 1.494	######################################
TABLE 24. MINIBIAXIAL S PROPELLANT: UTP 19360B 400/1777 REQUESTOR: CGriten Francis Wor:	DEFINITIONS: Time B Stress (DSI) B Strein (X) T(Gir) B Test Air Temperature (F) T(prop) B Test Propellant Temperature (F)	CALIBRATION DATA; Cal Wt = 10.0 lbs Lead Cal (lbs/volts) Offset (volts) Pot Cal (in/volts) Temp Cal (f)	AREAS (sq in):	STRESS DATA (ps.):  SET 1

ver Protesta Istockska Istockska Istockova Parasta

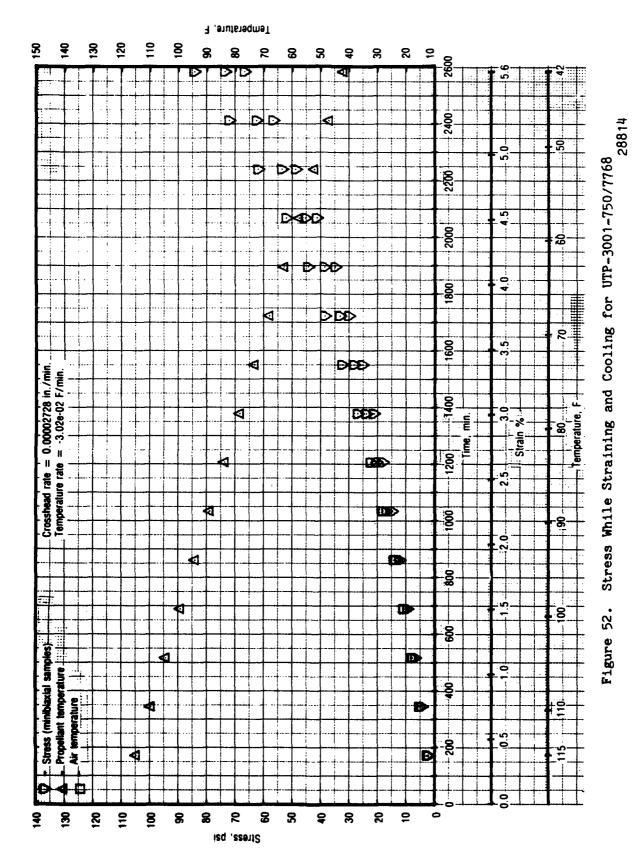
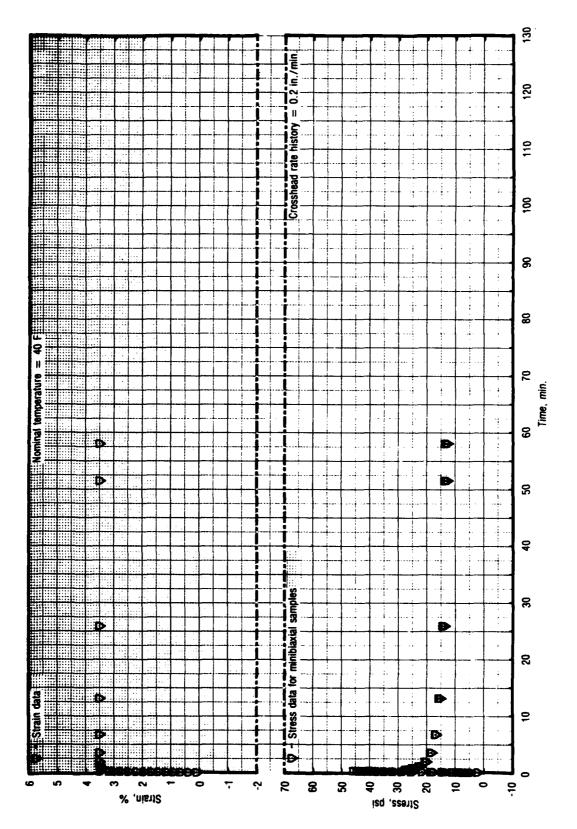


TABLE 25. MINIBIAXIAL STRESS WHILE STRAINING AND COOLING

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PROPELLANT: UTP 3061 750/7768 REQUESTOR: Carlton Francis BOR: 2742-400-0000		23	OPERATOR: 1962	12/22/81		
ITIONS: IAME From	(u)	K.	KFLATIONSHIPS:  g = Force/Area  c = Sample Extension/Length	Area Extension	/Length	
A L	(F)		VALUE	2 =.	<b>4</b> 0 F	
			Kede Fend XIO Rate XIO Rate		in  2728 in/min	c
CALLT	-	SAMPLE	Ņ			
Coad Cal (16s/in) and Cal (in) and Cal (in)	120.0	1.52	1.84			
st Test:	6.88 30.32 38.0	6.76	6.92 0.92			
REAS (sq in):	0.885	0.834	0.879		: : : : : : : : : : : : : : : : : : : :	
DATA (psi): 172.3000 114 344.5000 104 516.9000 104	84 0 38 0 78 1 75	04-30 04-74 17-4	SAMPLE 23.	2007	DW0-4W	00 00000 00000
8647, 1000 94 043, 6000 94 205, 9000 83 378, 1000 778		048-19 000000	2000 2004 2000 2000 2000	48000 48000 4000 7000 7000 7000 7000	2007 2004 2004 2004 2004 2004 2004 2004	0
1922 7000 68 1894 9000 62 2067 2000 57 2239 4000 57	04-19	30 .07 34 .87 41 .37 48 .89	33.73 38.66 53.66 53.77	24.0 24.0 24.0 24.0 20.0 20.0 20.0 20.0	סכטוהיסו	NWW4
2584.0000 42	-	56.85 66.83	73.74	84.49		5.28



Test No. 16 - Stress While Step Straining for UTP-19,360B-400/177 Figure 53.

TABLE 26. MINIBIAXIAL STRESS WHILE STEP STRAINING

accell described, recolecte accepted, experient accepted, accepted, accepted, employees, accepted, employees,

-81	Extension/Lenath = 40 E = 1.25 in = 3 Z		:	39 Avg S1 96 5.34 0 39 5.34	.53 13.05 1 .61 16.84 1 .64 20.49 1 .53 24.15 1	4.7.7.4.4.4.4.4.4.4.4.4.4.4.4.4.4.4.4.4	44 46 44 00 00 00 00 00 00 00 00 00 00 00 00	63 226 81 63 224 69 65 22 47	8.67 11 16.83 5.77 15.81 5.52 14.22 13.26 0.261 3.56 0.261
DATE: 12-28 (IFRATOR: BC	KFLATIONSHIPS;  E = Force/Apr  E = Sample E  NUMINAL VALUES;  Test Temp  Gage Length  Nom. Strain XHD Rate	SAMPLE 3 29.773 30.009 0.251 0.071	1.478	SAMPLE 2.17 58 6.47 79 10.58	22 14.41 75 18.16 15 21.68 68 25.23	200 200 200 200 200 200 200 200 200 200	440 455 655 440 454 456 456 456 456 456 456 456 456 456	0.53 0.53 0.53 0.53 0.53 0.53 0.53 0.53	18.31 16.81 15.53 15.76 17.82 13.82 13.82 13.82 15.73 13.73 13.73 13.73
	in) e (E)	29 . 816 0 . 058 - 0 . 385 - 42 . 1	1.478	rain 33	43 43	でいる でいる で	000	Biring Biring	พ.ศ.ศ.ศ.ศ.ศ.ศ.ศ.ศ.ศ.ศ.ศ.ศ.ศ.ศ.ศ.ศ.ศ.ศ.ศ
100/1777	Start of Test (m. 1) emperature (F)	Э н н		T(prop) T(air) 42.1 43.0 42.0 41.9	86.48	NWM4	7 41.	40 V W	44444 44000 44444 140000
PELLANT: UTP 19360B 4 UESTOR: Carlton 2242-400-000	INITIONS: IIMO From of Estress (ps of Estrain (X) I (air) = Iest Air I (prop) = Iest Prope	BRATION DATA: Cal Mt = 10.0 lbs Load Cal (lbs/volts Offset (volts) Pot Cal (in/volts)	AREAS (sq in):	DATA (p.				20.00	20 3.51132 21 6.71208 22 13.71208 23 25.91282 24 51.51364

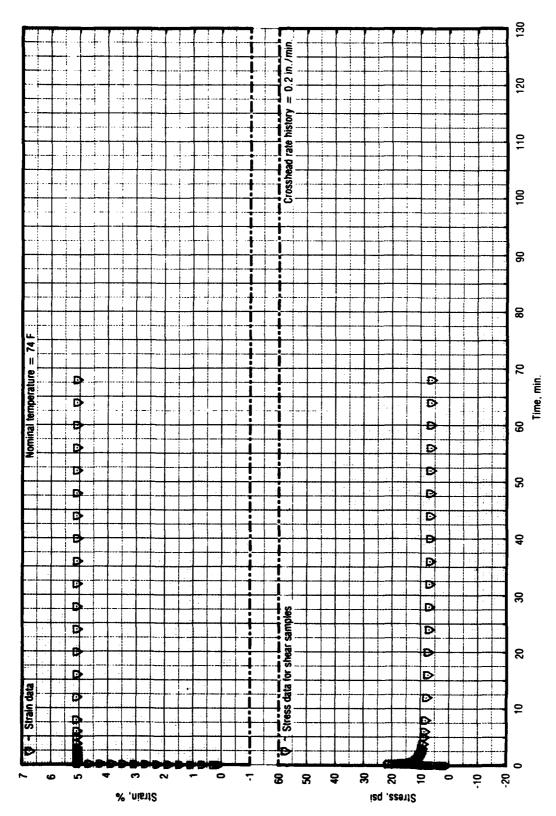


Figure 54. Test No. 17 - Stress While Step Straining for UTP-3001-750/7768

Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Table   Tabl	TAB	WHILE STEP OF 2)	•	
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0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	" Time From Start of Test (min) " Stress (ps.) " Strain (%) " Test Air Temperature (F) " Test Propellant Temperature (F)		TIONSHI I SPORT CONT CAL XADAR TA	Signature of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the state of the s
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## STEP STRAINING SHEAR STRESS WHILE (SHEET 2 OF 2) TABLE 27.

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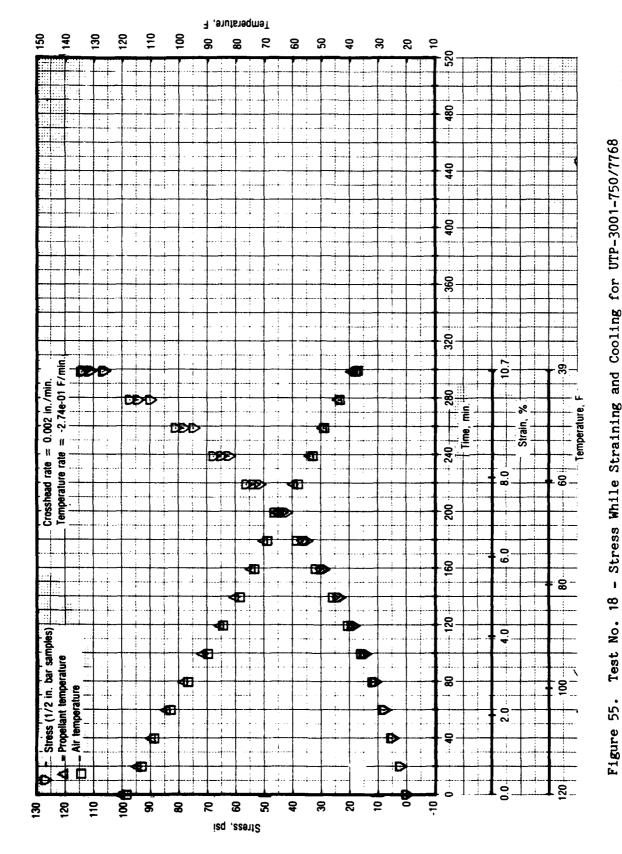
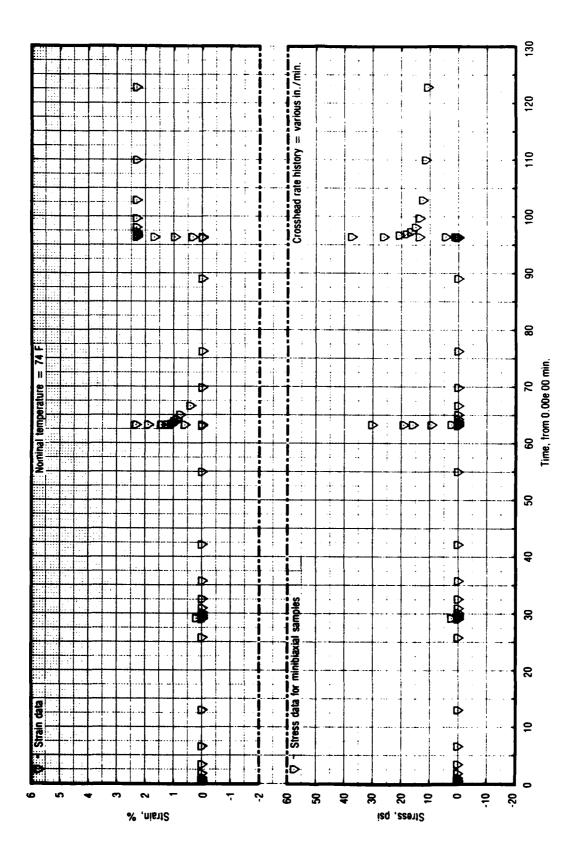


TABLE 28. 1/2-IN. BAR STRESS WHILE STRAINING AND COOLING

DATE: 12/2/81 OPERATOR: JWD	RELATIONSHIPS;  # Force/Area  # Sample Extension/Length NOMINAL VALUES;  Test Temp  Gage Length # 5.97 in Nom. Strain # 10.2 in/min XHD Rate # .002 in/min	DOM:	2	0	### ### ### ### ### ### ### ### ### ##
		1 ( 40 W···	40	0.23	ณ 4446เพิ่มสู่เพล่น 2004 5 - เกิบกลุ่มจับสู่เพล่น 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200 - 200
	(F)	SAMPLE 2 2 2 1 0 1 0 0 0 0 0 0 0 0 0 0 0 0 0 0	- 40 ስ4 ተዚህ	0.228	44 0 GREAT STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND STAND S
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1 750/7768 6-0000	From Start (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost) (Cost	= 10.0 lb elts) lts) =	= 10.0 lt glts) lts) =		
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PROPELLANT: REQUESTOR: WOR:	DEFINITIONS: Time T(air) T(prop)	CALIBRATION Pretest:   gad Cal   est cal fot Cal Temp	Post Test: Load Cal Offset Pot Cal	AREAS (sq in	2 044276 4444400000000000000000000000000000000



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Figure 56. Test No. 19, Part 1 - Stress for UTP-3001-750/7768

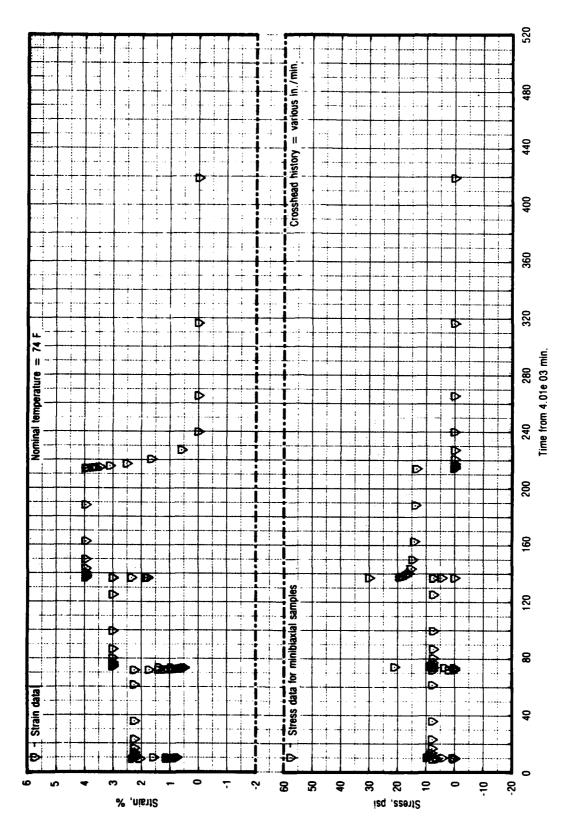


Figure 57. Test No. 19, Part 2 - Stress for UTP-3001-750/7768

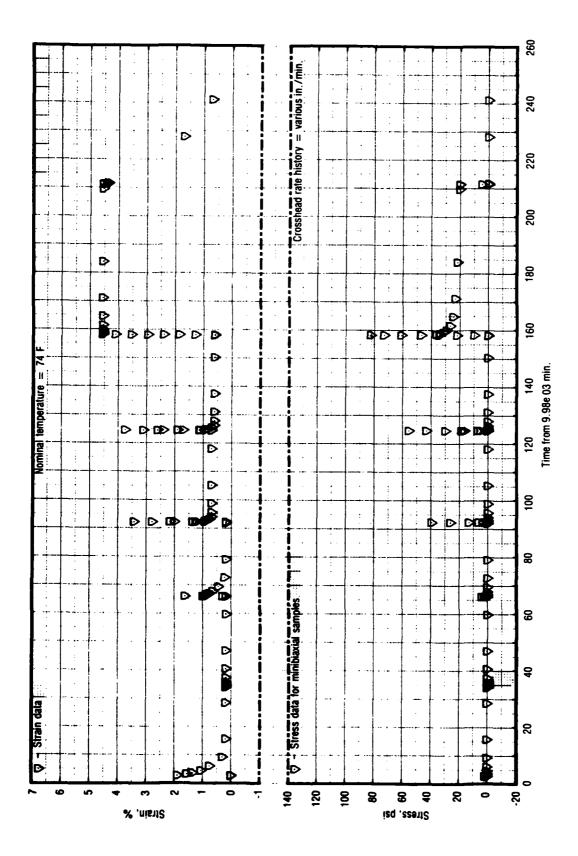


Figure 58. Test No. 19, Part 3 - Stress for UTP-3001-750/7768

TABLE 29. TEST NO. 19 - MINIBIAXIAL STRESS, SAMPLE (SHEET 1 OF 6)

DATE: 1/18/82 S1 UPERATOR: IMD	KILATIONSHIPS.  c = Force/Area  c = Sample Extension/Length  NUMINAL VALUES.  Test Temp = 74 F  Gage Length = 1.25 in  Nom. Strain = Various X  XHD Wate = Various in/Min		
(Shiel of 6)	n) (F)	SAMPLE 24.768 0.083 -0.396	**************************************
	of Test (Mi ture (E) Temperature		1(air) 799.4 799.4 759.4 76.9
50/7768 30c is	Start (2) (Empera	115) ts) ==	T (000000000000000000000000000000000000
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PROPELLANT: REQUESTOR: WOR:	DEFINITIONS.	CALIBRALION Cal II Load Cal Offset Pot Cal	STRESS 054 117 517 520 523 523 523 523 523 523 523 523 523 523

TABLE 29. TEST NO. 19 - MINIBIAXIAL STRESS, SAMPLE (SHEET 2 OF 6)

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TABLE 29. TEST NO. 19 - MINIBIAXIAL STRESS, SAMPLE (SHEET 3 OF 6)

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TABLE 29. TEST NO. 19 - MINIBIAXIAL STRESS, SAMPLE (SHEET 4 OF 6)

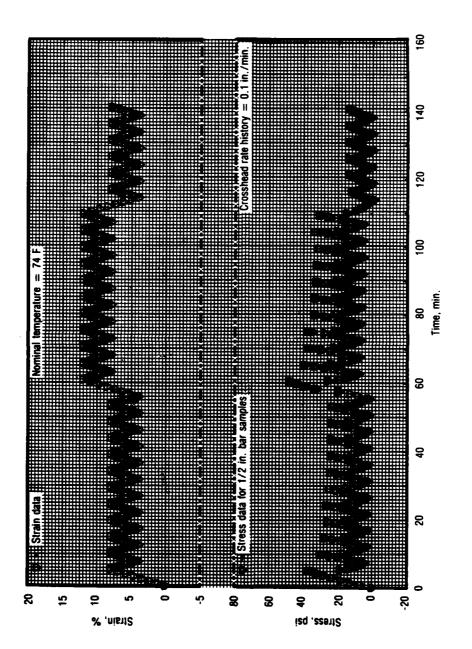
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TABLE 29. TEST NO. 19 - MINIBIAXIAL STRESS, SAMPLE (SHEET 6 OF 6)

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Test No. 20 - Stress While Cycling for UTP-19,360B-400/1777 for Complete Test Figure 59.

TABLE 30. 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 1 OF 15)

19360B 400/1777 1/14/8 on, Francis -400-4000	From Start of Test (min)  s (psi) n (%) hir Temperature (F) Propellant Temperature (F) Gage Length Now, Strain Now, Strain	SAMPLE 3 /uolts) 29.776 29.679 29.966 ts) -0.012 0.036	0.252 0.252 0.252	(volts):	-0.176 0.137 0.047 0.0 -0.351 0.187 0.097 0.1	-0.700 -0.875 -0.875 -0.316 0.328 0.328 0.353 0.353 0.353 0.353	-1.216 -1.131 -1.045 0.199 0.100 0.199	-0.871 -0.784 -0.784 -0.696 -0.696 -0.696 -0.696 -0.696 -0.696	123 0.028 0.028 0.028 0.008 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.108 0.	11.040. 0 000. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0 100. 0
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TABLE 30. 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 2 OF 15)

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TABLE 30. 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 5 OF 15)

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TABLE 30. 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 6 OF 15)

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TABLE 30. 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 7 OF 15)

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TABLE 30. 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 8 OF 15)

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TABLE 30. 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 9 OF 15)

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TABLE 30. 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 10 OF 15)

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TABLE 30. 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 11 OF 15)

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TABLE 30. 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 12 OF 15)

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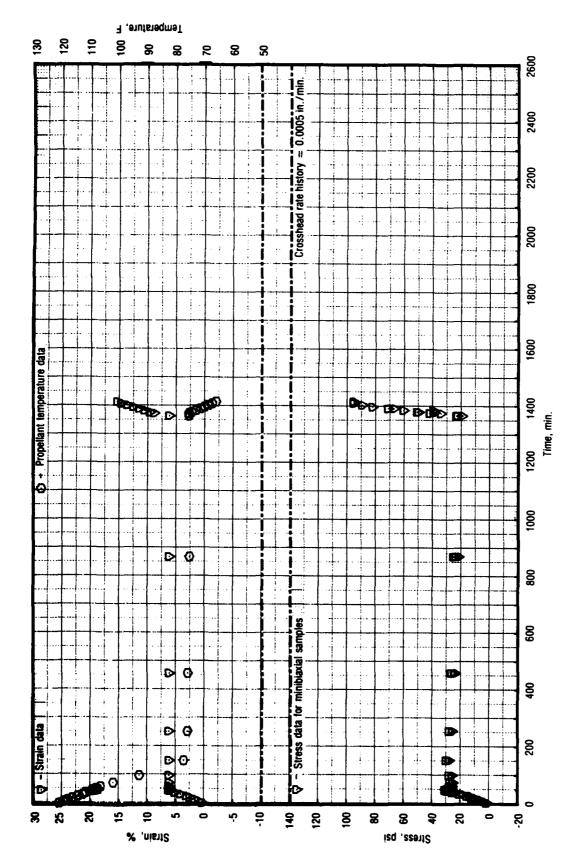
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TABLE 30. 1/2-IN. BAR STRESS WHILE CYCLING (SHEET 15 OF 15)

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- Stress While Complex Straining and Cooling for UTP-3001-750/7768 Test No. 21 Figure 60.

TABLE 31. MINIBIAXIAL STRESS WHILE COMPLEX STRAINING AND COOLING (SHEET 1 OF 2)

PROPELLANT:	UTP 3881 75 Carl ten Fra 2742-406-00	0/7768 ncis 00				DATE: 1/5,	5/82		
DEFINITIONS:	Time Fr Stress	Start of 5i)	Test (min	•		RFLATIONSHIP	S: e/Area le Extensio	on/Length	
T(air) T(prop)	1472 1484 1684	Temperatu Flant Te	re (F) Aperature	(F)		NUMINAL VALUE Test Temp Gage Leng	(C) #	0,hold,4	0F F
							.0 0.≡		
CALIBRATION I Pretest: Ligad Cal Ligad Cal Pot Cal	DATA: Wt = (18, 401t (18, 401t) (10, 401ts) (10, 401ts) (10, 401ts) (F)	10.0 lbs		29.713 0.051 -0.389 119.8	SAMPLE 29.633 0.116	29.910 0,105			
Post Test: Load Cal Offset Pot Cal	Cal Wt = (1bs/volt (volts) (in/volts) (F)	5). 0 1bs		29.756 0.199 0.199 65.4	29.505	29.874			
AREAS (sq in)	-			1.480	1.453	1.480			
DATA	1): TIME 8717 3782	10	- M	ac Mo	1		100	15	
41J-0L	5.8784 2.1287 8.3784 6.288	8111 847 447	1117 117 100 100 100 100 100 100 100 100	u	• • • •	17.00 10.00 10.00 10.00		<b>†</b> • • •	
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C. 410	9.5424 9.7535 0.3544 0.3553	0.00	<b>6</b> 0 m //	ומנהמני	4 1	• :	* '	• • • • •	
92.65	51, 15616 52, 75655 55, 95708 62, 35736	4 4	109.7 107.5 108.7	• • • • •	• •		1	24.52 27.76 26.84	0.889 0.777 0.717 0.717
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TABLE 32. STRESS-STRAIN COMPARISON FOR UTP-3001 TESTS PHASE III

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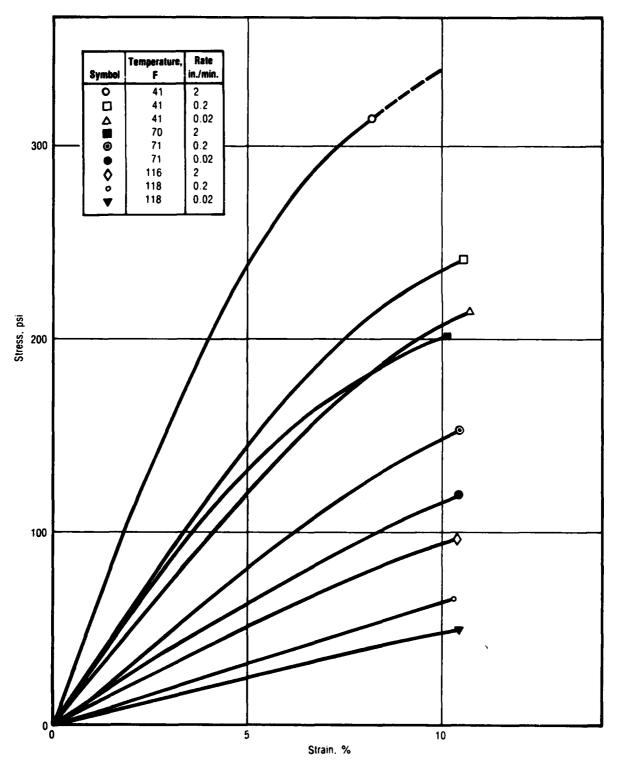
Test	Temper- ature,	Rate,	Strain,	Stress.	Stress Test	Differ- ence,	Box	Days	
No.	Ce.	in./min.	*	psi	No. 15	*	No.	Machining	Remarks
	11	2	6.85	290			7	8	Average
	#1	7	9.65	352			8	~	Sample No. 3
	<b>1</b>	0.5	10.26	239			~	m	
	41	0.02	5.29	123			N	m	Average
	41	0.02	10.10	209			~	ı m	No. 3
	2	2	10.14	203			-	ı W	•
	7	0.2	₩8.6	147			_	י וכ	
	71	0.02	9.86	115			_	, rv	
	116	~	10.38	97.0			N	۰ ۵	
	118	0.2	5.03	28.0			~	~	Average
	118	0.5	10.09	64.3			۱ م	۰ ۸	No. 1
•	118	0.02	10.17	48.6			N C	ο (2)	No.3
$\Xi$	99.5	0.000027	1.50	10.4			m	<b>#</b>	Straining-
									cooling
	017	0.2	3.13	107	95	12.6	m	10	
	73	0.2	3.22	50.8	53	-4.2	(**	<b>=</b>	
	117	0.2	3.30	20.5	21	-2.4	· (*)	ī.	
<del>8</del>	17	0.5	5.08	21.3	₹ 23.2	-8.2	1	=	No. 1 shear
	7.	0.5	5.87	23.6	≈ 26.5	-14.2	ı	=	No. 2 shear
	1.	0.2	6.52	25.0	≈ 29.2	-14.4	ı	=	No. 3 shear
$\Xi$	95.2	0.0002	2.53	9.57			-	7	Straining-
									cooling
	9.76	₩000.0	2.29	10.4				m	E =
	190.4	0.002	1.95	7.97			_	'n	14 14
	14	_	2.26	37.0	~ 42	-11.9	#	81	No. 3
$\widehat{\Xi}$	118.1	0.0005	1.71	7.42			<b>4</b>	Ŋ	Straining-
									cooling

(1) Insufficient data to extrapolate to an equivalent temperature for stress adjustment. (2) Comparison of shear to biaxial data is only approximate. Notes:

STRESS-STRAIN CC-A-ARISON FOR UTP-19,360B TESTS PHASE III TABLE 33.

41	rest No.	Tempera- ture, F	Rate, in./min.	Strain,	Stress, psi	Stress Test #15	Difference,	Box No.	Days After Machining	Remarks
11   2   10.01   111   11   2   8   8   8   8   8   8   8   8   8	14	41	2	10.01	112			2	8	Average
41       0.2       10.22       80.6       41       0.2       10.22       80.6       4.1       4.2       8.63       8.14       4.4       8.63       8.14       7.1       2       8.63       8.14       7.1       2       8.63       8.14       8.14       8.14       8.14       8.14       8.14       8.14       8.14       8.14       8.14       8.14       8.14       8.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14       9.14		<b>1</b>	2	10.01	111			^	œ	
41         0.02         10.15         63.0         Average           71         2         8.63         81.4         Average           70         0.2         10.20         65.6         Average           71         0.02         10.03         50.4         Average           120         2         10.13         63.9         Average           119         0.2         10.24         50.4         Average           119         0.2         9.87         49.4         Average           118         0.02         10.14         40.3         Average           118         0.02         10.14         40.3         Average           119         0.2         10.14         40.3         Average           119         0.02         10.14         40.0         10.0         2         7           119         0.02         3.50         44.0         10.0         3         12         Straining           10         0.2         3.09         19.6         16.7         17.4         3         7         Average           10         0.2         5.33         14.5         ×13.0         5.8         - <t< td=""><td></td><td>t <del>1</del></td><td>0.2</td><td>10.22</td><td>80.6</td><td></td><td></td><td>۱۸</td><td>ο ας</td><td></td></t<>		t <del>1</del>	0.2	10.22	80.6			۱۸	ο ας	
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		10.3	0.0005	1.03	0.34	~ 4·70	34.9	<b></b>	<b>⇒</b>	Straining-

Notes: (1) Straining-cooling tests were compared as small strain and accompanying temperature to avoid the strain-thermal shift complication.

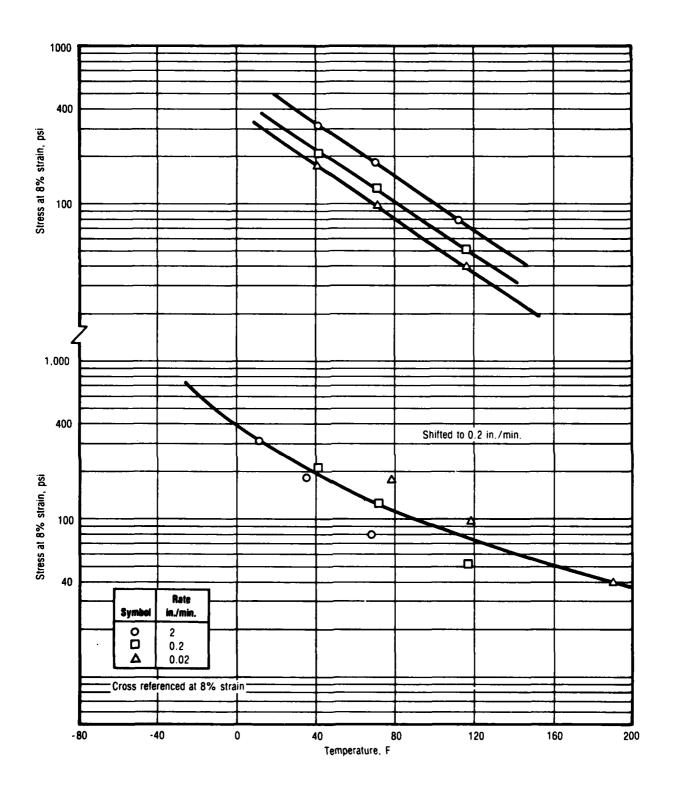


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Figure 61. Biaxial Constant Rate Data for UTP-3001

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Figure 62. Test No. 14 - Biaxial Constant Rate Data for UTP-3001 28829

The UTP-19,360B biaxial constant rate data plotted in Figure 63 were used to obtain the stress comparison data (Table 33) at different strain levels. The stress-temperature plot in Figure 64 gave a good temperature extrapolation which was used to obtain comparison stress values for the slower rate straining-cooling tests. The stress values selected for comparison were early in the test at an elevated temperature and small strain. This was chosen to specifically avoid the complication of evaluating the combined thermal-mechanical interaction shift factor (A_F). While one of the straining-cooling tests showed a difference of 35%, the absolute delta stress was small.

# Uniaxial, Biaxial and Shear Comparison

Comparisons of the different samples were made in order to show that the propellant, used in each of the tests, was of the same family. This was done at ambient temperature for selected rates and was limited to strain rates that were close to each other. This minimized the time-temperature equivalence shifts to small changes for neglectable data input errors. The adjustments made for strain levels are given in Table 34.

The comparisons between uniaxial and biaxial in the table are close to the theoretical ratio of 75 to 80%. The shear to uniaxial ratios of 0.28 and 0.39 bracket the nominal theoretical value of 1/3. For the comparisons made in Table 34, the UTP-3001 and UTP-19,360B propellants have to be considered part of the same family. Any minor differences can be attributed to a between carton effect.

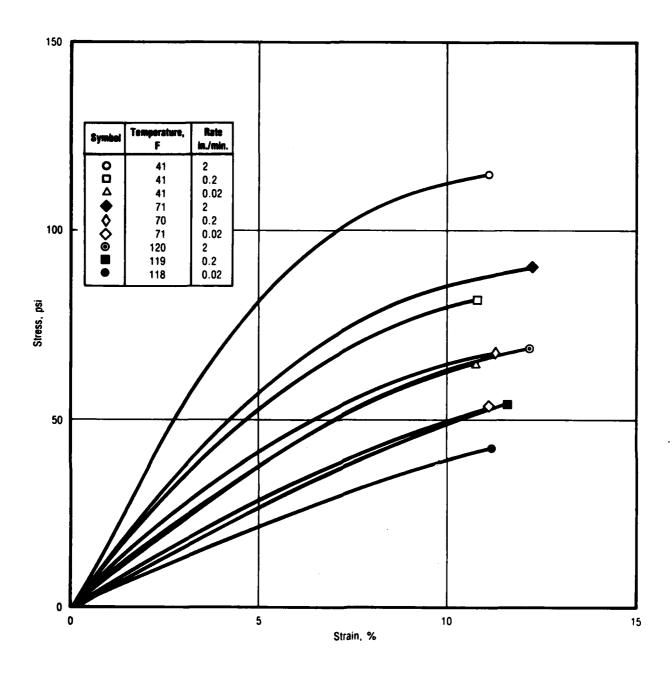


Figure 63. Biaxial Constant Rate Data for UTP-19,360B 28831

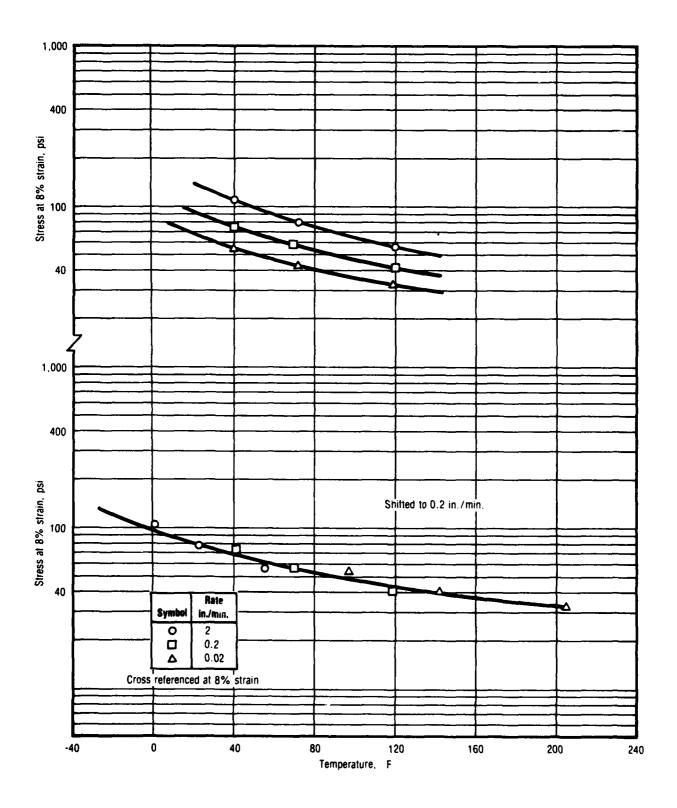


Figure 64. Test No. 14 - Biaxial Constant Rate Data for UTP-19,360B 28832

TABLE 34. COMPARISON OF UNIAXIAL, BIAXIAL, AND SHEAR TEST DATA FROM PHASE III

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	Temper-			UTP-3001	3001	UTP-19,360B	360B
Sample	ature, F	Rate, in./min.	Strain Rate, in./in./min.	Strain,	Stress, psi	Strain,	Stress, psi
Uniaxial Adjusted to	70	-	0.1667	10.0 5.8	128 86.8	10.0 5.6	57.7 38.9
Biaxial	70	0.2	0.160	10.0	148	10.0	65.0
Shear Adjusted to	02	0.5	0.2 0.160	5. 8.	23.3 24.0	5.6	14.3
Stress Ratios for:	for:		UTP 3001		UTP 19360B	808	
Uniaxial/biaxial	al		128/148 = 0.86		57.7/65.0 = 0.89	- 0.89	
Shear/uniaxial			24.0/86.8 = 0.28		15.0/38.9 = 0.39	: 0.39	

#### 4.0 THEORETICAL DEVELOPMENT

#### 4.1 INTRODUCTION AND PRELIMINARY STUDY

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Generally speaking, solid propellants may be considered as lightly crosslinked long-chain polymers, highly filled with coarse solid particles. They respond viscoelastically to the action of external stimuli, but certain aspects of their behavior cannot be reproduced by the classical linear or nonlinear theories of fading-memory materials. Thus, in recent years, much work has been concerned with the development of appropriate models to predict the mechanical response of solid propellants. A current trend is to express the observed response in terms of some measure of "damage" at the continuum level where damage is described as the difference between the observed response and that predicted by a fading-memory theory, like Linear Viscoelasticity. There is now sufficient experimental evidence to show that damage (References 13, 17, 20, 24, and 28) per se is a microscopic phenomenon associated with the initiation and growth of flaws, debonding between matrix and solid filler particles, and molecular chain scission. Although it is largely irreversible, damage is partially recoverable shortly after removal of the loading system. This recovery from damage is termed healing. It is clear that, depending on the propellant and service requirements, it may also have to be accounted for in a constitutive theory for solid propellants.

In the present program, two approaches to characterizing damage have been followed. In the first one, damage is treated as the algebraic difference between the measured stress and that predicted by Linear Viscoelasticity, so that:

$$g_{c}(t) = \sigma(t) - \sigma_{\ell}(t)$$
 (1)

in which  $\sigma_{\ell}$  and  $\sigma_{c}$  are the linear-viscoelastic and correction terms, respectively, with  $\sigma_{r}$  the measured stress. In the second approach, the difference between measured and fading-memory type stresses is handled through a stress-correction function in the following form.

The softening function (C) is made to depend on the past maximum strain or stress and  $\sigma_{\hat{\Gamma}}(t)$  represents an appropriate function of the fading-memory type stresses.

Broadly speaking, the current versions of models by M. Gurtin and M. Quinlan are of the type presented in equation (1) above, while those of R. Schapery, W. Hufferd and Swanson are of the form given by equation (2).

The following presents some experimental evidence on the type of nonlinearities exhibited by solid propellants, and briefly discusses the pioneering work of Mullins and Tobin in treating the large hysteresis observed in tire rubbers. Next, the theory of Linear Viscoelasticity is applied to predict the response of UTP-19,360B and UTP-3001 under various strain histories. The ensuing results are meant as a basis for comparing the propellant response as predicted by each of the candidate constitutive laws. This comparison should be most meaningful because each of the theories considered evolved from a set of modifications to Linear Viscoelasticity. Subsequently, the nonlinear theory of Farris (Reference 5) is presented. This theory was employed during the first phase of the program to predict the response of TP-H1011 and to compare the results with those of the other five constitutive laws. Finally, a detailed description is given of each of these five stress-strain relations. This includes the original concept of the models, their current versions, comparisons of predicted and measured stresses for a variety of strain histories, and some pertinent guidelines for characterizing solid propellant according to each theory.

## 4.1.1 Experimental Background

The complex behavior of solid propellants, as well as some attempts at developing usable stress-strain laws for these meterials, are well documented in References 3 through 31. It is shown there that a given deformation process causes a change in the response properties of solid propellants, for instance a drop in the relaxation modulus. As stated before, this deviation from some

expected response is what has been called damage. It is evidenced as phenomenological macroscopic changes that are caused by undefined, but real, irreversible or partially reversible microscopic changes. Polymer bond breakage, vacuole formation in the polymer matrix, dewetting between the polymer matrix and solid filler particles, microcracking, etc., are among the possible microscopic causes of observed permanent-memory effects in propellants.

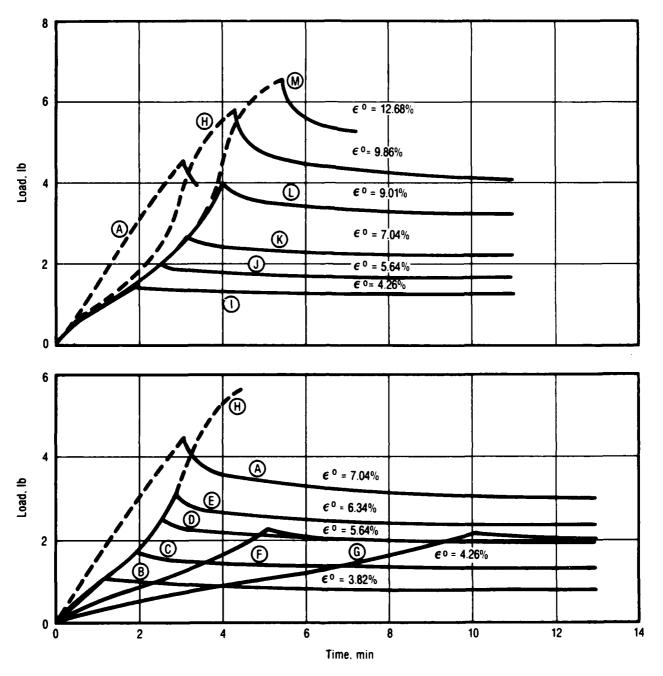
Studies on uniaxial solid propellant samples have indicated that these materials exhibit large hysteresis even at small strains. These studies have also revealed that the state of damage in solid propellants is determined primarily by the maximum strain or stress undergone during the loading histories.

The typical nonlinear hysteresis and permanent-memory effects exhibited by solid propellants are illustrated in Figure 65. A series of finite-duration, variable-strain-level ramp pulses were used to obtain the propellant response subsequent to a given damage history (References 12 and 13). All ramps had the same initial moderate rise rate, with two exceptions to be noted later, and all ramps had the same very slow decline rate.

Observations of the load on the specimen after returning to its original length (zero strain) showed that it took approximately 30 min for the stress to relax to zero.

A series of tests were run on a  $1/4 \times 1/4 \times 4$ -in. tab-end sample. The virgin specimen was initially strained to a level of 7.04% and allowed to relax to achieve a rest-state condition. The first part of the testing is shown in the lower half of Figure 65 (curves A-H) and the last part in the upper half (curves H-M).

Curve A shows the load response to this first pulse. The specimen was then subjected to four successive ramp strain pulses ranging from  $\epsilon^0$  = 3.82% to  $\epsilon^0$  = 6.34%. There was a rest period allowed between each pulse. The results are shown in Figure 65 as curves B through E.



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Figure 65. Relaxation after Damage

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Two aspects of the propellant's behavior are worth noting. First, during the constant strain rate portion of the ramp, each successive load-time curve is essentially identical. This indicates that the "new material" has the same nonlinear rate-dependency under repeated strain conditions as long as the strain

levels are below the previous maximum strain of  $\epsilon^0 = 7.04\%$ . Second, the relaxation portions of the curves are essentially homologous, indicating that a viscoelastic relaxation process is taking place.

Curves F and G present the results of two additional tests at two successively lower strain rates where the sample was loaded to 5.64% strain each time. A strong rate dependency is observed during the rise portion of the ramp; however, curves F and G rapidly rejoin curve D indicating that the material is behaving in a viscoelastic fading-memory fashion.

The specimen was next subjected to a ramp strain pulse reaching a higher strain level ( $\epsilon^{\circ}$  = 9.86%) than the maximum 7.04% previously experienced (Figure 65, curve H). The first part of curve H repeats the loading ramp portion of curves B-E to indicate the same "new material" rest-state. Note that the load-time curve returns to the initial or virgin constant strain rate curve once the previous maximum strain (7.04%) has been passed.

Subsequently, the specimen was strained with the ramp pulse to four different strain levels less than 9.86% ( $\epsilon^{\circ}$  = 4.26%, 5.64%, 7.04%, and 9.01%), as shown in curves I through L. The results show that a new rate-dependency has developed (compare the rising portions of I through L with the rising portion of H). Thus, another "new-material" rest-state has been produced as a result of the second maximum strain level of 9.86%. Lastly, the specimen was strained to another new maximum of  $\epsilon^{\circ}$  = 12.68% as shown in curve M. It again appears that it returned to the virgin undamaged curve once the 9.86% strain level was exceeded.

The above experimental evidence suggests that the form of the constitutive equation should remain unchanged with respect to the material's current reststate. This condition should remain as long as the damage is unchanged (i.e., the  $\epsilon_{\max}$  is unchanged during its subsequent strain histories).

Figure 66 shows a replot (curves N and O) of some of the results just discussed. After an initial maximum strain (7.04%) the specimen was allowed to

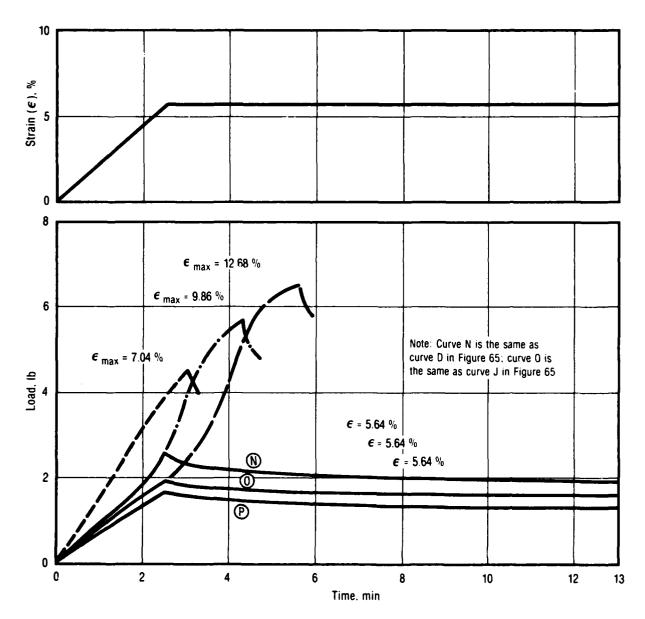
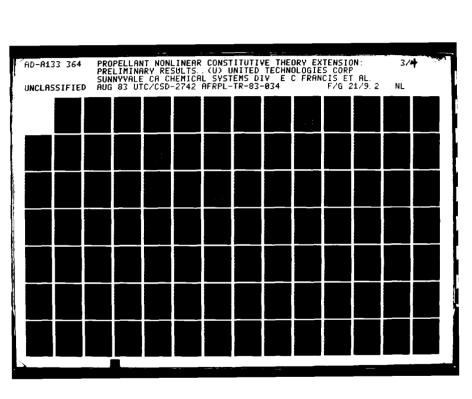
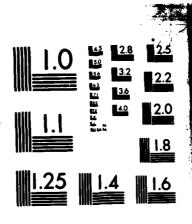


Figure 66. Relaxation after Damage

return to a rest-state and then strained to a value of  $\epsilon^{\rm O}$  = 5.64%, with the result shown as curve P. These three identical strain history tests of three different material states indicate that the higher the state of damage (primarily  $\epsilon_{\rm max}$ ), the softer the material response upon subsequent testing.

In addition, other experimental studies have pointed out the importance of healing effects, load duration, and initial strain rate (References 12, 13, 14,





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and 15). Finally, it is important to note that the behavior of solid propellant, depicted in Figures 65 and 66, is similar to that exhibited by rubber. The nonlinear uniaxial stress response of rubber, with and without carbon black filler and in the absence of time effects, was characterized quite well by Mullins and Tobin (Reference 27) with equation (3).

$$\epsilon = \epsilon_{\mathbf{u}} \mathbf{F}$$
 (3)

where:

- $\epsilon$  = engineering strain. The Mullins-Tobin model is not limited to small strains.
- $\epsilon_{\rm u}$  =  $\epsilon_{\rm u}(\sigma)$  = strain as a function of engineering stress for the polymer without filler and without damage. The characteristic shape of this function is shown in Figure 67.
  - $F = F(\sigma_{max}, N) = damage or softening function which depends on the maximum stress experienced by the rubber and the number N of loading and unloading cycles. F is not very sensitive to N, but depends strongly on any hard filler particles that may be present.$

A large amount of rubber data can be predicted by means of this equation when the samples are not allowed to rest between cycles. Recovery or healing occurs as a function of the rest time. Therefore, healing would have to be considered in an accurate characterization of rubber.

Introducing the inverse of  $\epsilon_{\rm u}$  =  $\epsilon_{\rm u}(\sigma)$ , equation (3) may be put in the form:

$$\epsilon = f(\epsilon/F)$$
 (4)

which shows that F (where F<1) is a strain-magnification factor. The ratio  $\epsilon/F$  is interpreted by Mullins and Tobin to be the average strain in the rubber phase of a hard particle-filled rubber. Without damage in a highly-filled rubber, F<<1. As the rubber is cycled between the strains  $\epsilon=0$  and  $\epsilon=\epsilon_{\max}$ , the ratio  $\epsilon/F$  at any strain decreases, and therefore the stress decreases. The shape of the

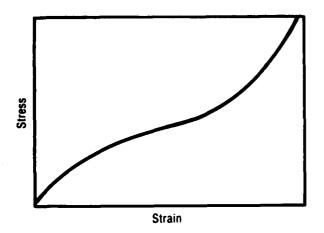


Figure 67. Stress-Strain
Curve for Rubber
22074

4.2 SELECTED THEORIES 22074

4.2.1 Linear Viscoelastic Constitutive Equation

4.2.1.1 Linear Viscoelastic Model

The one-dimensional stress-strain law for a thermorheologically simple linear viscoelastic solid may be expressed as:

$$\sigma(t) = \int_{0}^{t} E \left(S_{t} - S_{\tau}\right) \frac{d\xi}{d\tau}(\tau) d\tau$$
 (5)

stress-strain curve is still as shown in Figure 67. It is similar to that for solid propellant after first-time loading. This fact and the ability of

the model presented in equation (3) to reproduce a large amount of rubber data

explains the great influence of the Mullins-Tobin approach on the devel-

opment of nonlinear constitutive

theories for solid propellants.

where:

o = stress

 $\epsilon$  = strain

E(t) = relaxation modulus (PRONY series representation
 using a matrix solution for curve fitting data;
 CSD Data Analysis Procedure No. 7.3)

 $S_t - S_\tau = temperature-shifted time, given by:$ 

$$S_{t} - S_{\tau} = \int_{\tau}^{t} \frac{d\tau}{A_{T} \left[T(\tau)\right]}$$
 (6)

and

a_T = time-temperature shift function, taken in the form:

$$A_{T} = \left(\frac{T_{R} - T_{a}}{T - T_{a}}\right)^{m} \tag{7}$$

in which  $T_R$  is the shift reference temperature, and both  $T_a$  and m are material parameters. The material parameters were obtained using CSD Data Analysis Procedure No. 7.4, which is a curve fit routine using Powell's algorithm.

The linear viscoelastic model was used to predict the response of UTP-19,360B and UTP-3001 under several strain histories. The corresponding results are included here as a basis against which to compare the stress predictions obtained using the nonlinear stress-strain laws considered in the program.

#### 4.2.1.2 Stress Predictions

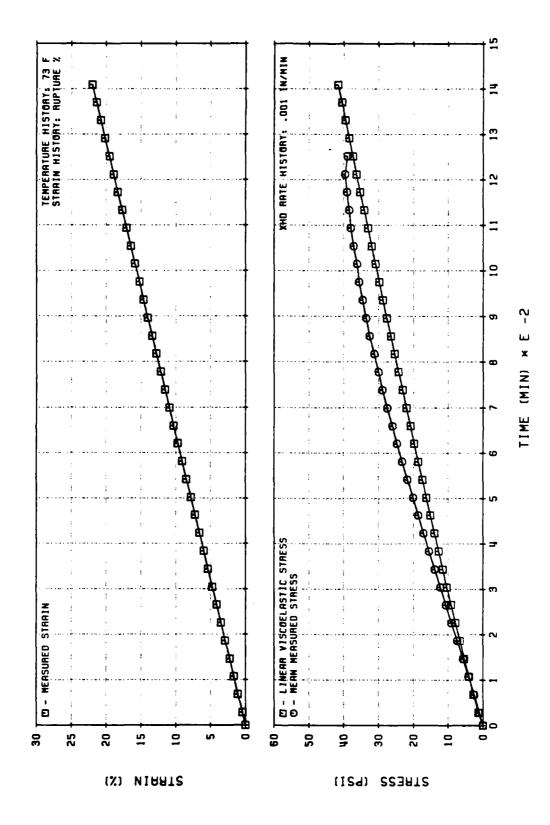
The measured response is compared against that predicted by linear viscoelasticity for UTP-19,260B in the following order (Figures 68 through 78).

The results for the lowest and highest constant-rate tests (Test No. 1) appear in Figures 68 and 69. Those for the dual-rate tests (high-to-low and low-to-high, Test No. 3) are shown in Figures 70 and 71. Figure 72 contains the comparisons for a saw-tooth strain history (Test No. 5) with increasing strain peaks and rest periods between cycles. The results corresponding to short- and long-duration similitude tests are presented in Figures 73 and 74. The predicted and measured responses for the three-step relaxation test (Test No. 13) are included in Figure 75. In addition, the time-temperature superposition principle is put to use with constant rate tests (Test No. 1) at 70, 40 and 123°F, as shown in Figures 76 to 78, respectively. Stress predictions for some of the same tests on UTP-3001 are shown in Figures 79 through 86.

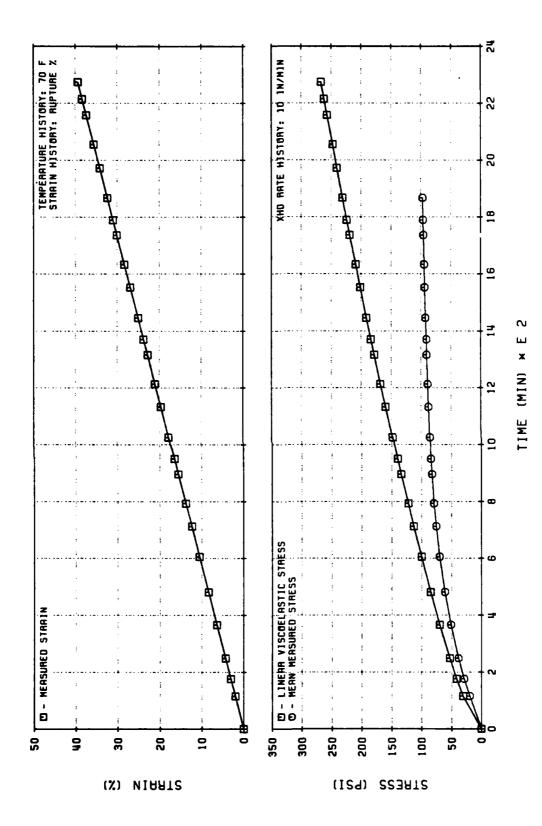
## 4.2.2 R. Farris Nonlinear Theory for Solid Propellants

The work of R. Farris (Reference 5) was a major attempt at predicting the nonlinear response of solid propellants in rocket motor analyses.

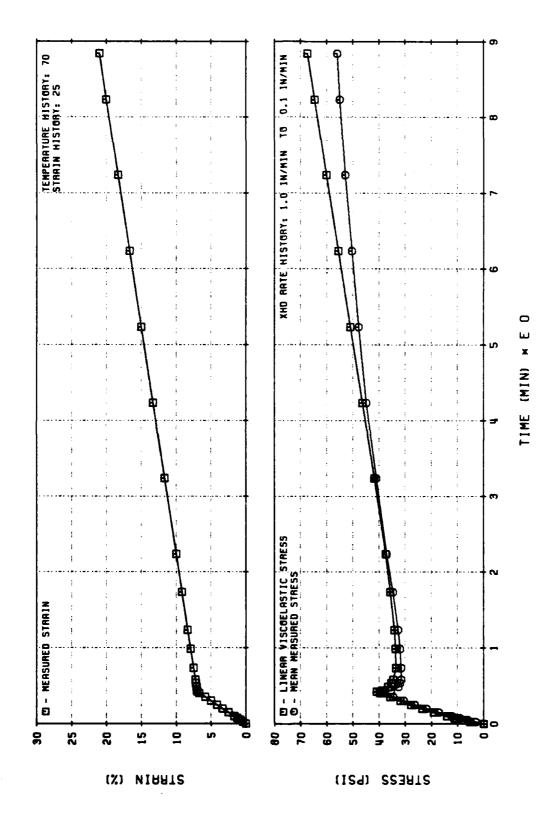
Experience with Farris' stress-strain law at Chemical Systems showed that this theory could not predict the response of solid propellants under strain (Text continued on page 200.)



Linear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 Constant Rate Test History (Code No. 1) Figure 68.

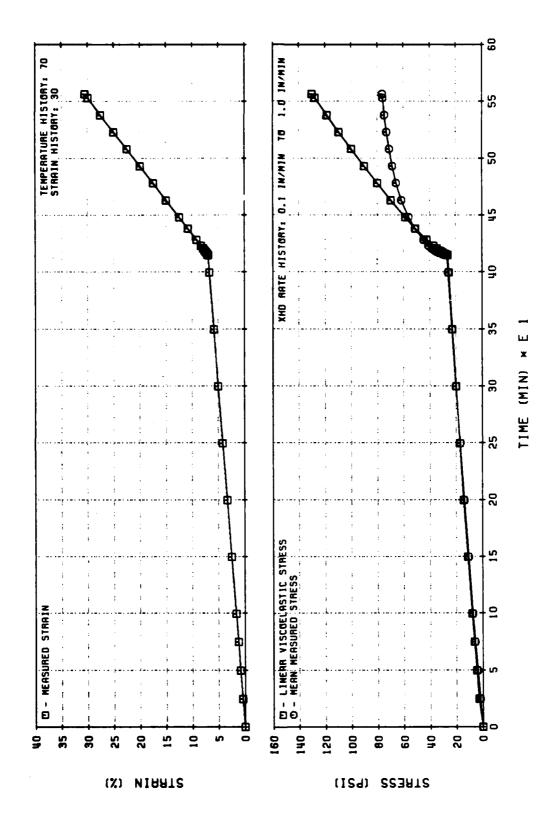


Linear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 Constant Rate Test History (Code No. 1) Figure 69.



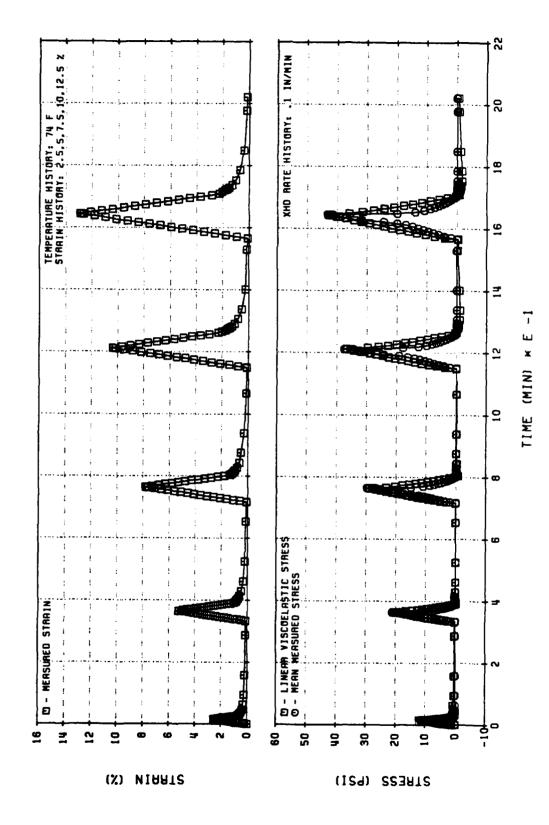
Linear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 Two Rate Test History (Code No. 3) Figure 70.

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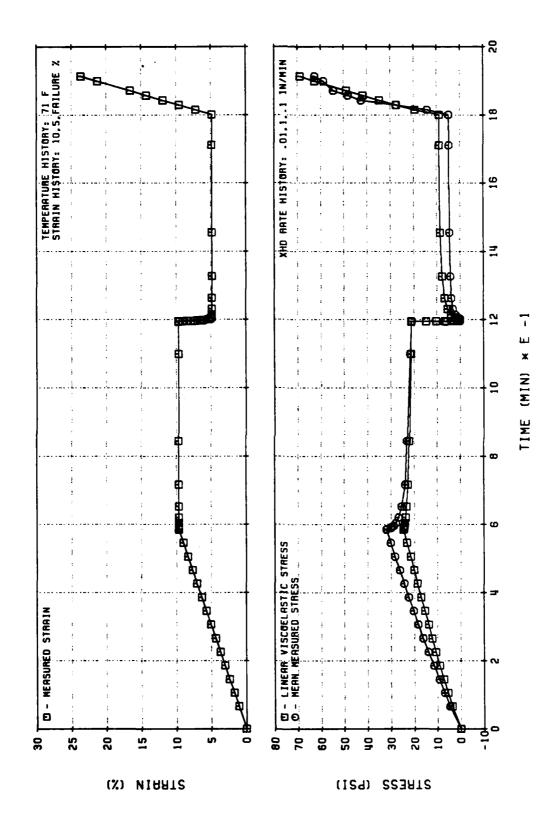


Linear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 Two Rate Test History (Code No. 3)

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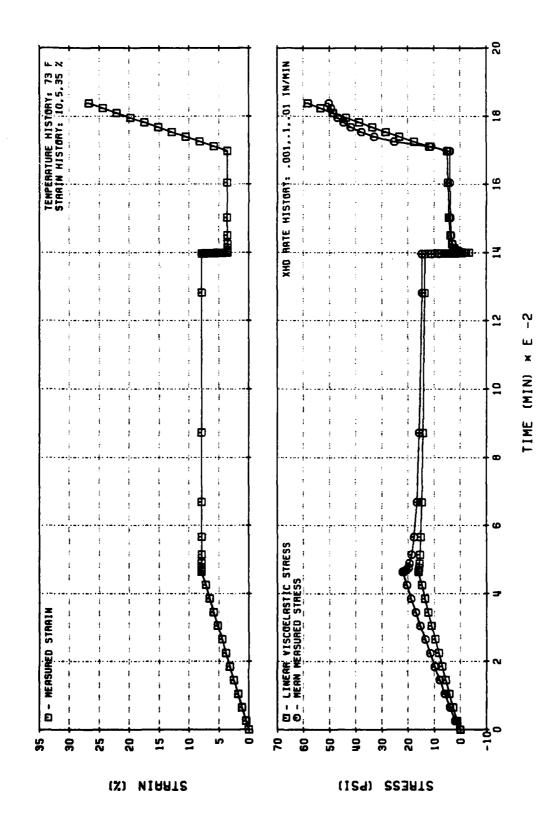


Linear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 Multiple Loading Test History (Code No. 5) Figure 72.

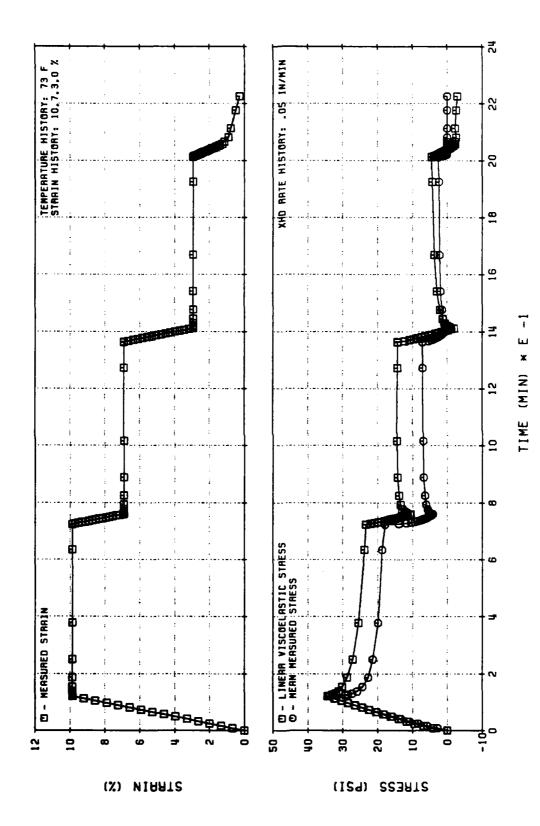


Linear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 Similitude Test History (Code No. 12)

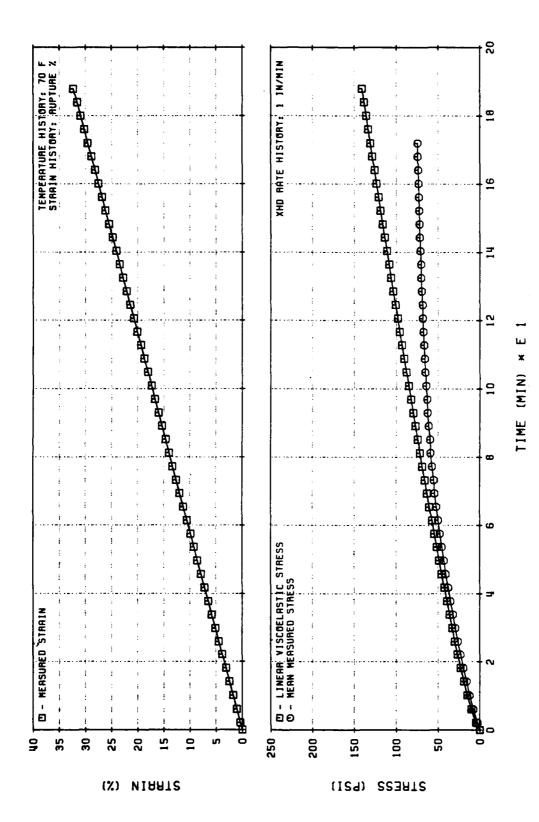
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Linear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 Similitude Test History (Code No. 12) Figure 74.

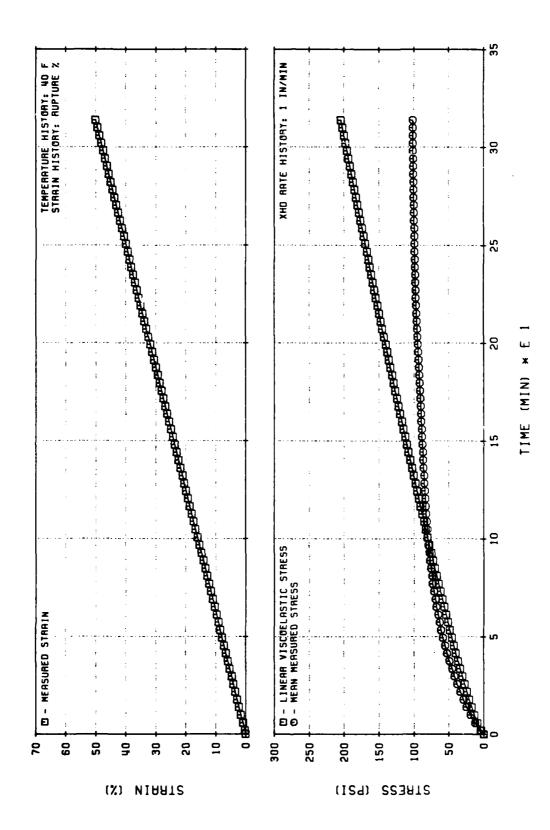


Linear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 Three Step Relaxation Test History (Code No. 13) Figure 75.



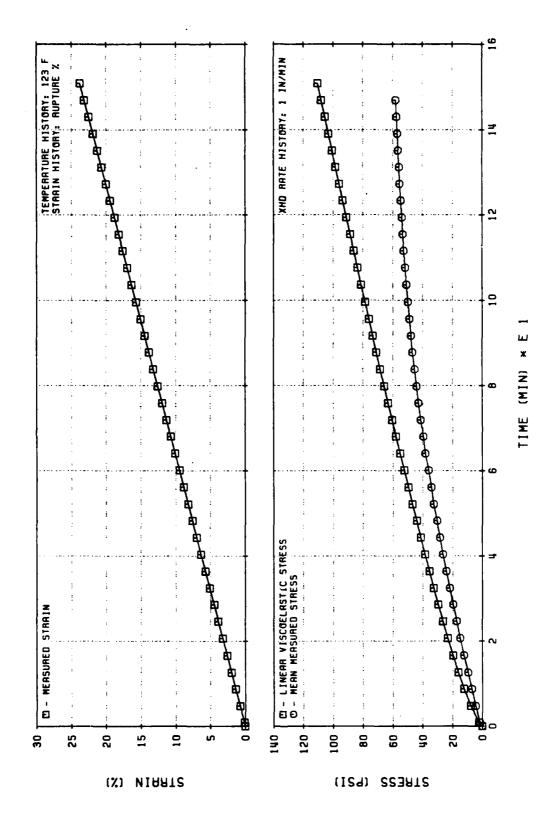
Linear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 Constant Rate Test History (Code No. 1) Figure 76.

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Linear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 Constant Rate Test History (Code No. 1) Figure 77.

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Linaar Viscoelastic Stress Predictions for UTP-19,360B-400/1777 Constant Rate Test History (Code No. 1) Figure 78.

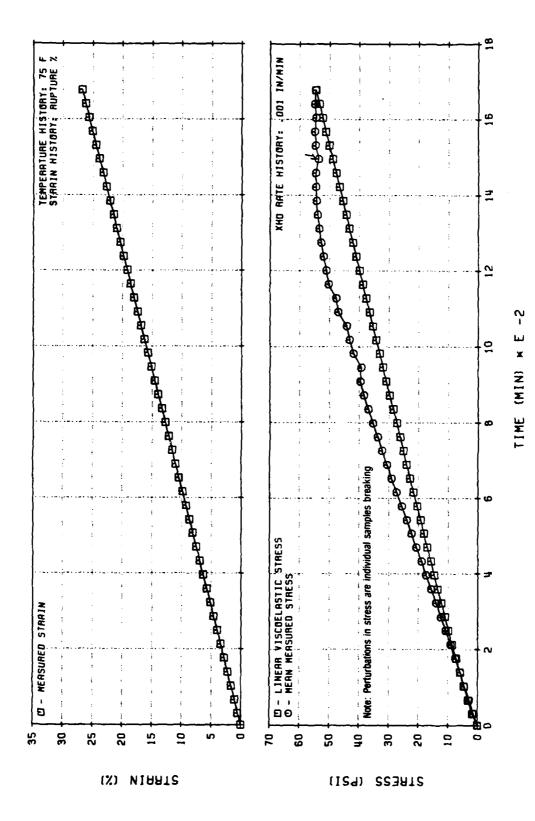
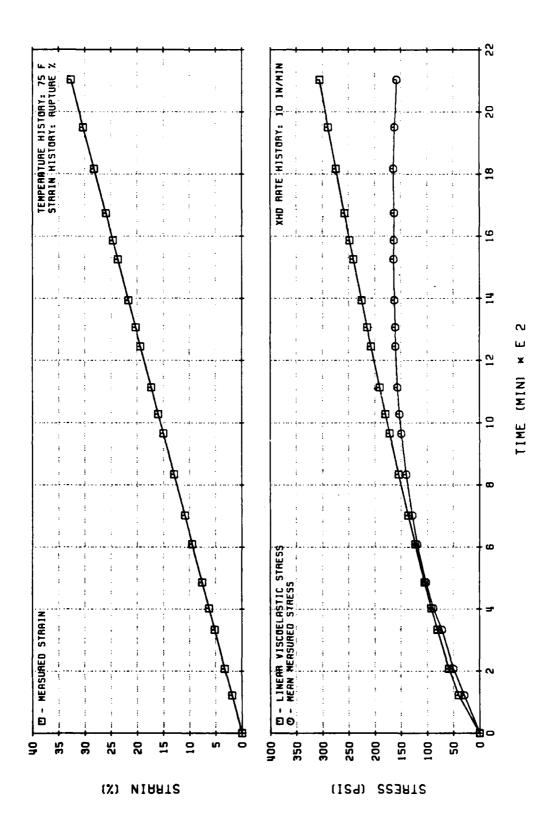
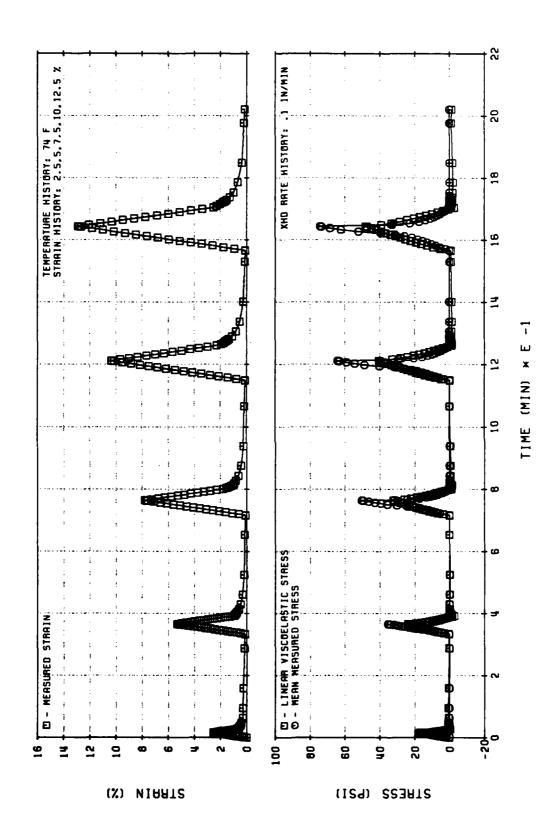


Figure 79. Linear Viscoelastic Stress Predictions for UTP-3001-750/7768 Constant Rate Test History (Code No. 1)



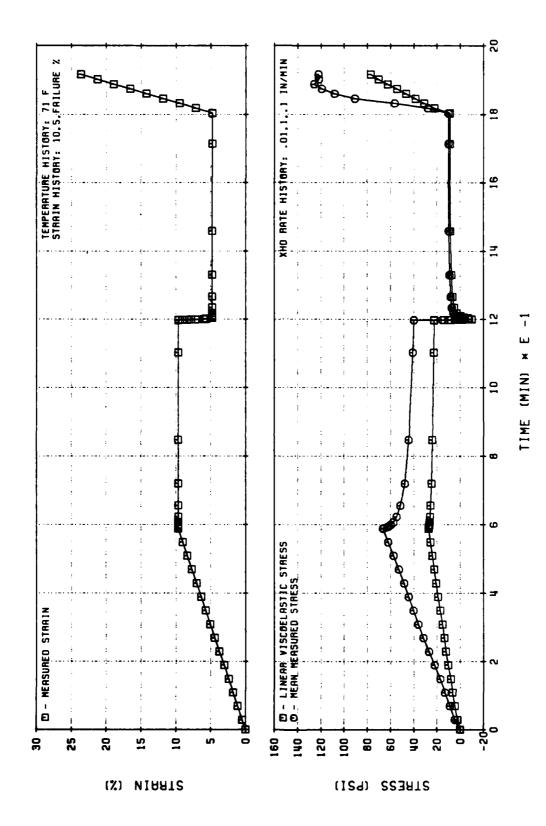
Linear Viscoelastic Stress Predictions for UTP-3001-750/7768 Constant Rate Test History (Code No. 1) Figure 80.

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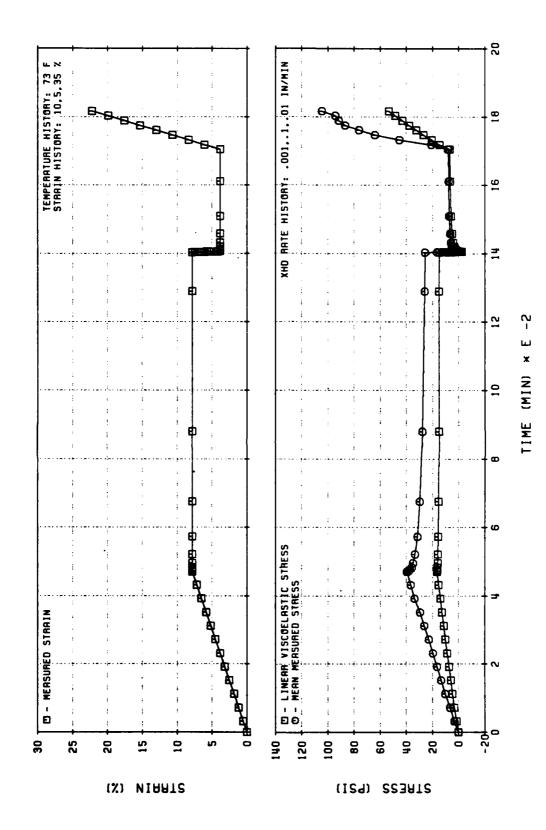


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Linear Viscoelastic Stress Predictions for UTP-3001-750/7768 Multiple Loading Test History (Code No. 5) Figure 81.



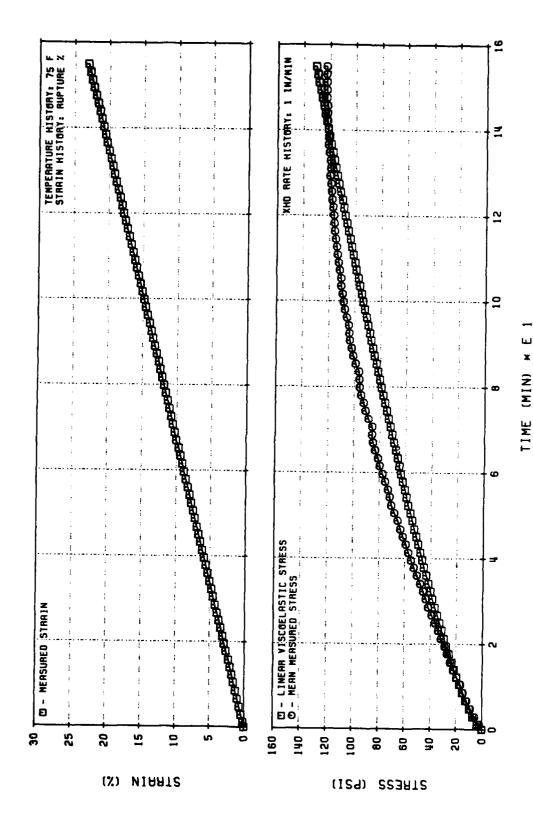
Linear Viscoelastic Stress Predictions for UTP-3001-750/7768 Similitude Test History (Code No. 12)



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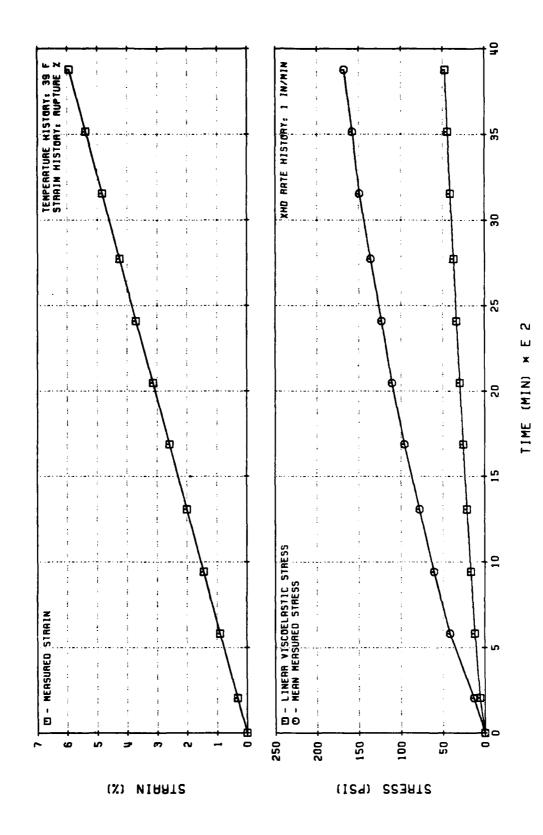
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Linear Viscoelastic Stress Predictions for UTP-3001-750/7768 Similitude Test History (Code No. 12) Figure 83.



Linear Viscoelastic Stress Predictions for UTP-3001-750/7768 Constant Rate Test History (Code No. 1) Figure 84.

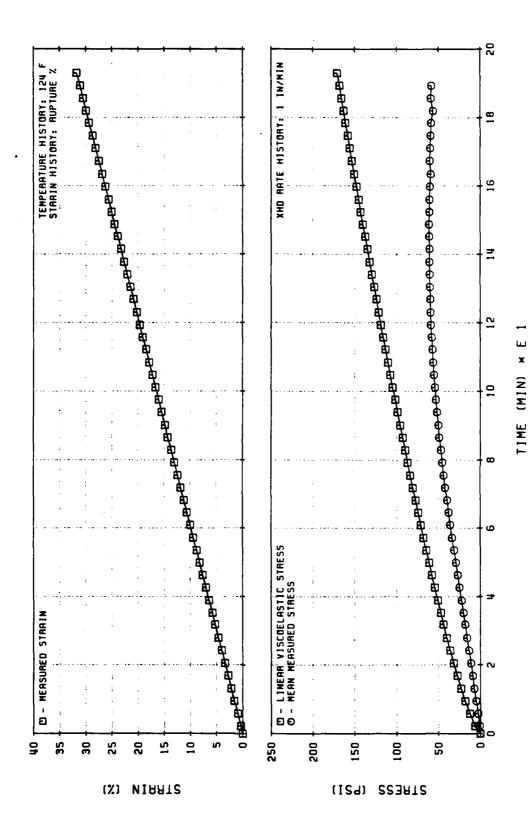
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Linear Viscoelastic Stress Predictions for UTP-3001-750/7768 Constant Rate Test History (Code No. 1) Figure 85.



Linear Viscoelastic Stress Prediction for UTP-3001-750/7768 Constant Rate Test History (Code No. 1) Figure 86.

histories that were not included in the set used for material characterization. Although the predictions were generally acceptable for loading histories of the types included in such sets.

The nonlinear theory of Farris was considered in the first phase of the program and it was compared to the other five approaches originally proposed.

Farris postulated a model, based upon previous work on rubber elasticity, to account for the permanent-memory effects exhibited by many solid propellants under constant rate loading. This model presumes the existence of inhomogeneities in the local strain field between filler particles, a distribution of polymer chain lengths between filler particles, and a uniform failure strain for each polymer chain. This model has been successful in predicting the nonlinear permanent memory response of solid propellants before dewetting. The model predicts the same response in compression as in tension. This prediction is in agreement with experimental observations, although the molecular mechanisms contributing to the permanent memory response in compression are clearly different from those in tension.

Once dewetting occurs, the model is modified to account for vacuole formation and different results in compression and tension are expected. Farris presented the constitutive equation as the sum of an essentially time-independent bulk stress  $\sigma_B$  and a time-dependent deviatoric stress  $\sigma_{ij}$ , so that in general

$$\sigma_{ij}(t) = \sigma_B \delta_{ij} + \sigma_{ij}^d(t)$$
 (8)

where  $\delta_{ij}$  is the Kronecker delta equal to unity if i=j and zero otherwise. The form developed for the deviatoric stress is

$$\sigma_{ij}^{d}(t) = e^{-BI}d/I_{\gamma} \begin{cases} A_{1} e_{ij}^{d}(t) + A_{2} \left(\frac{I_{\gamma}}{\|I_{\gamma}\|_{p_{2}}}\right)^{m_{2}} e_{ij}^{d}(t) & (9) \\ t + \int A_{3}(t - \xi) e_{ij}^{d}(\xi) d\xi + \left(\frac{I_{\gamma}}{\|I_{\gamma}\|_{p_{4}}}\right)^{m_{4}} \int_{0}^{t} A_{4}(t - \xi) e_{ij}^{d}(\xi) d\xi \end{cases}$$

where

 $I_d$  = volume dilatation =  $e_{11} + e_{22} + e_{33}$  for small strains

$$I_{\gamma} = \text{octahedral shear strain} = \frac{1}{3} \left[ (e_{11} - e_{22})^2 + (e_{22} - e_{33})^2 + (e_{33} - e_{11})^2 \right]^{1/2}$$

d e_{1,1} = deviatoric strain tensor

B,  $A_1$ ,  $A_2$ ,  $A_3$ ,  $A_4$ ,  $M_2$ ,  $M_4$ ,  $P_2$ ,  $P_4$ , = constants

and

$$\|\mathbf{I}_{\gamma}\|_{\mathbf{p}_{\underline{i}}} = \left[\int_{0}^{t} |\mathbf{I}_{\gamma}(\xi)|^{\mathbf{p}_{\underline{i}}} d\xi\right]^{1/\mathbf{p}_{\underline{i}}} \tag{11}$$

Equation (9) has been applied to reasonably complex deformation histories using unpressurized and pressurized uniaxial and biaxial test specimens. The agreement was not as good as would have been desirable, but it was still better than Linear Viscoelasticity. Time-temperature superposition was included in equation (9) by introducing a time-temperature shift factor, at, and redefining the  $L_{\rm D}$  norm of equation 140 in the form:

$$\| \mathbf{I}_{\gamma} \|_{\mathbf{p}_{\underline{\mathbf{i}}}} = \left( \int_{0}^{t} \frac{|\mathbf{I}_{\gamma}(\xi)|^{p_{\underline{\mathbf{i}}}}}{a_{\underline{\mathbf{T}}}(\xi)} d\xi \right)^{\mathbf{I}/p_{\underline{\mathbf{i}}}}$$
(12)

Experimental data for simultaneous cooling and straining have been fit using equation (9) with the introduction of a time-temperature shift function  $a_T$  through equation (11). The justification for introducing an  $a_T$  in the above manner is not immediately obvious or adequately explained in the available literature.

To present the response to interrupted and cyclic constant strain rate tests, equation (9) was modified by setting  $P_{ij} = \infty$  and  $A_{ij} = -A_{ij}$  so that:

$$\sigma_{ij}^{d}(t) = e^{-BI}d^{I\gamma} \left\{ A_{1} e_{ij}^{d}(t) + A_{2} \left( \frac{I\gamma}{\|I\gamma\|_{P_{2}}} \right)^{m_{2}} e_{ij}^{d}(t) \right.$$

$$\left. + \left[ 1 - \left( \frac{I\gamma}{\|I\gamma\|_{\infty}} \right)^{m_{4}} \right] \int_{0}^{t} A_{3}(t - \xi) e_{ij}^{d}(\xi) d\xi$$

where:

$$\|I_{\gamma}\|_{\infty} = \max \left|I_{\gamma}(\xi)\right|, \ 0 < \xi < t$$

The multiplier for the hereditary integral in equation (12) vanishes whenever the current value of  $I_{\gamma}$  is at its largest, and is non-zero for all other values. This representation allows for viscoelastic (fading memory) response on unloading.

The bulk stress,  $\sigma_{\rm B}$ , in equation (8) was taken to be essentially time-independent, although there is coupling between distortion and dilatation as indicated in the exponential multiplier in equations (9) and (13).

The first attempt to represent the bulk stress took the form of a series

$$\frac{\sigma_{\mathbf{k}\mathbf{k}}}{3} = \sum_{\mathbf{j}=0}^{\mathbf{N}} \mathbf{A}_{\mathbf{j}} \mathbf{I}_{\mathbf{d}}^{\mathbf{j}} \mathbf{I}_{\gamma}^{\mathbf{j}} ; \mathbf{A}_{\infty} = 0$$
 (14)

This equation adequately predicts the bulk stress as long as it does not vary greatly. However, when a hydrostatic pressure is superimposed, very poor results are obtained. In an attempt to overcome this difficulty, Farris modeled the compressibility of the gas voids caused by vacuole dilatation by treating them as spherical voids contained in an elastic medium. Assuming that (1) the voids themselves offer no resistance, (2) that all void dilatation is caused by distortion of the surrounding elastic material, (3) the void content at zero pressure may be represented as a power law in terms of the octahedral shear strain  $I_{\gamma}$ , and (4) that the bulk behavior varies linearly with hydrostatic pressure (P), the model yields:

$$I_{d} = C_{1} P + C_{2} I_{\gamma}^{n} e^{\left(\frac{-3P}{4G}\right)}$$

$$(15)$$

for the dilatation. The  $C_1$ ,  $C_2$  and n are constants and G is the shear modulus of the elastic matrix material.

# 4.2.3 R. Schapery's Nonlinear Stress-Strain Law

# 4.2.3.1 Original Model

The constitutive theory advanced in References 17 and 18 for viscoelastic materials with microcracking was taken by Dr. R. Schapery as the starting point for predicting the response of solid propellants under general loading conditions. The one-dimensional version of this law takes the following simple form:

$$\sigma = \frac{\mathbf{A_F}}{\lambda} \sigma_{\ell} \tag{16}$$

where  $\sigma_{\ell}$  is the linear viscoelastic stress for a thermorheologically simple material:

$$\sigma_{\ell} = \int_{0}^{\infty} E(\xi - \xi') \frac{d\epsilon_{\sigma}}{d\tau} d\tau , \qquad (17)$$

with

 $\epsilon_{\rm O} \equiv \epsilon$  -  $\alpha \, ({\rm T}$  -  ${\rm T}_{\rm O})$  = strain due to mechanically applied stress

$$\xi = \int_{0}^{t} dt'/A_{T} \left[T(t')\right]$$

 $\xi' \equiv \xi(\tau)$ 

 $E(\xi)$  = linear viscoelastic relaxation modulus

 $T_O$  = temperature at t = 0

 $A_F = A_F$  (T) = temperature-dependent material function

 $\lambda = \lambda (S_{\ell})$ : softening function in which the damage parameter:

$$S_{\ell} = \int_{0}^{\xi} \left( \frac{\sigma_{\ell} A_{F}}{f} \right)^{q} d\hat{\xi}$$
 (18)

depends only on the strain and temperature histories, and:

$$f = \begin{cases} 1 & \text{for } 0 \le \epsilon < \epsilon_1 \\ (\epsilon/\epsilon_1)^{\beta} & \text{for } \epsilon_1 \le \epsilon < \epsilon_2 \\ (\epsilon_2/\epsilon_1)^{\beta} & \text{for } \epsilon \ge \epsilon_2 \end{cases}$$

for constant threshold strains  $\epsilon_1$  and  $\epsilon_2$ , with  $\beta>0$ .

The function  $F=F(\epsilon_\sigma)$  and the positive, constant exponent, q, originate with the equation for microcrack speed,

$$\frac{dA}{d\hat{\xi}} = M(K_{I}/f)^{q} \tag{19}$$

where M is a positive constant and:

$$d\hat{\xi} = dt/A_c \tag{20}$$

in which  $A_c = A_c(T)$  is the shift-factor for microcrack growth rate.

The functional form of the softening function,  $\lambda = \lambda(S_{\ell})$ , depends on the type of behavior that need be reproduced. The following special case was used:

$$\lambda = \left[1 + cS_{\ell}\right]^{p/q}$$

where c and p are positive constants. Note that when  $S_{\ell}=0$ , or c = 0, a linear viscoelastic stress-strain equation is recovered from Equation (16).

Taking  $A_F = 1$ , several sets of numerical values for the constitutive parameters corresponding to TP-H1011 were tried without success. This theory was also

used to predict the response of UTP-19,360 and UTP-3001. Having failed to perform better than Linear Viscoelasticity in many cases, it has undergone several changes since.

### 4.2.3.2 Current Model

The essential form of the modified uniaxial stress-strain relation is given by:

$$\sigma = f(\epsilon^{\circ}, \epsilon^{\circ}m, S)$$
 (21)

where:

 $\sigma$  = engineering stress

 $\epsilon_{\mathbf{r}}^{\,0}$  = psuedo strain

$$\epsilon_{\mathbf{r}}^{0} = \frac{1}{E_{\mathbf{R}}} \int_{0}^{\mathbf{t}} \mathbf{E} (\mathbf{t} - \tau) \frac{d\epsilon}{d\tau} d\tau$$
 (22)

 $\epsilon_{\rm m}^{\rm O}$  = maximum value of  $|\epsilon^{\rm O}|$  up to the current time

S = damage parameter

$$S = \left( \int_{0}^{t} \left| \epsilon \circ \right|^{q} dt \right)^{1/q}$$
 (23)

ER = arbitrarily selected reference modulus,

E(t) = linear viscoelastic relaxation modulus,

$$= E_e + E_2 t^{-n} = E_2 (E_\tau + t^{-n}), \tag{24}$$

 $E_{\tau} = E_{e}/E_{2}$ 

q = positive constant.

and

The functional form of f in equation (21) depends on the material considered. Studies on solid propellant to date indicate it may be taken as follows for some solid propellants.

$$f = Y_1 Y_2 Y_3 P_{15} sign (\epsilon^0)$$
 (25)

in which

$$sign (\epsilon^{\circ}) = \begin{cases} 1 & \text{if } \epsilon^{\circ} > 0 \\ 0 & \text{if } \epsilon^{\circ} = 0 \\ -1 & \text{if } \epsilon^{\circ} < 0 \end{cases}$$

and  $P_{15}$  is used to normalize function  $Y_3$  to unity, at a reference point. The  $Y_4$ 's are the following functions of damage and pseudo strain:

$$Y_1 = Y_1 (S) = \begin{cases} 1 + A_1S + A_2S^2 + A_3S^3 & \text{for } S \leq S_0 \\ A_4 S^{A_5} & \text{for } S > S_0 \end{cases}$$
 (26)

$$Y_2 = A_2 S^{0.63-Sx} (\epsilon_m^0)^{(0.463 - M_x - L_x)} |\epsilon^0|^{L_x}$$
 (27)

$$Y_3 = C_0 + C_1x + C_2x^2 + C_3x^3 + C_4x^4 + C_5x^5$$
 (28)

where:

$$x = x_r \left| \frac{\epsilon^0}{\epsilon_m^0} \right| \lambda \tag{29}$$

in which  $\mathbf{x}_{\mathbf{r}}$  is the only root of the equation:

$$\max(S_r) = Y_3 (x_r) \tag{30}$$

with  $\max$  ( $S_r$ ) representing the  $\max$  maximum value of  $S_r$  up to the current time, and:

$$S_{r} = \frac{S^{3}x \left| \epsilon_{m}^{0} \right|^{M_{X}}}{P_{15}}$$
 (31)

while  $\lambda$  is a factor that accounts for relatively small higher order effects possibly due to rehealing and particle interaction.

The constants entering the definitions of  $Y_1$ ,  $Y_2$ , and  $Y_3$  depend on the material. For UTP-19,360B they are:

$$S_0 = 42$$
  
 $S_x = 0.637$   
 $M_x = -0.387$   
 $L_x = 0.85$  (32)

and the factor  $\lambda$  is given by:

$$\lambda = K_{\mathbf{x}} C_{\mathbf{cm}}^{-C_{\mathbf{x}}} \tag{33}$$

The resulting form of equation (21) for UTP-19,360B is thus:

$$\sigma = P_{15} \text{ A6 Y}_1 \text{ Y}_3 \quad |\epsilon^{\circ}|^{L_X} \text{ sign } (\epsilon^{\circ})$$
 (34)

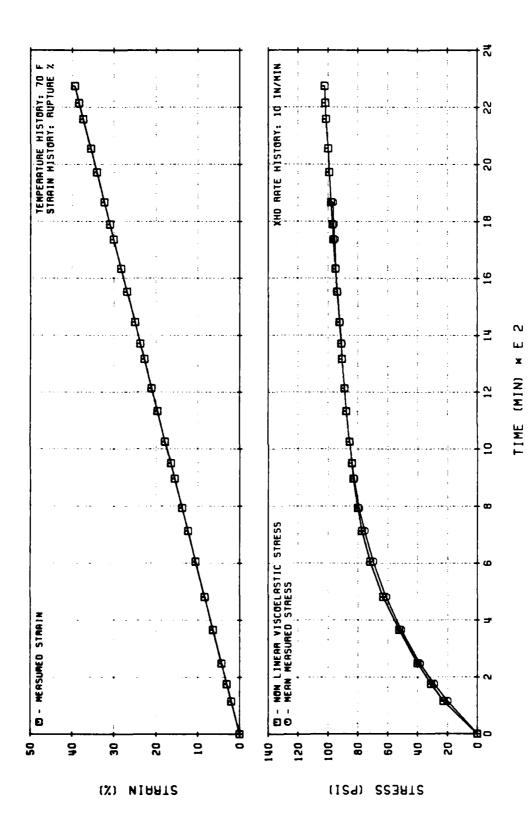
Clearly, if  $L_X = 1$ , equation (34) may be written as:

$$\sigma = A_{\rm F} \int_{0}^{t} E (t - \tau) \frac{d\epsilon}{d\tau} d\tau$$
 (35)

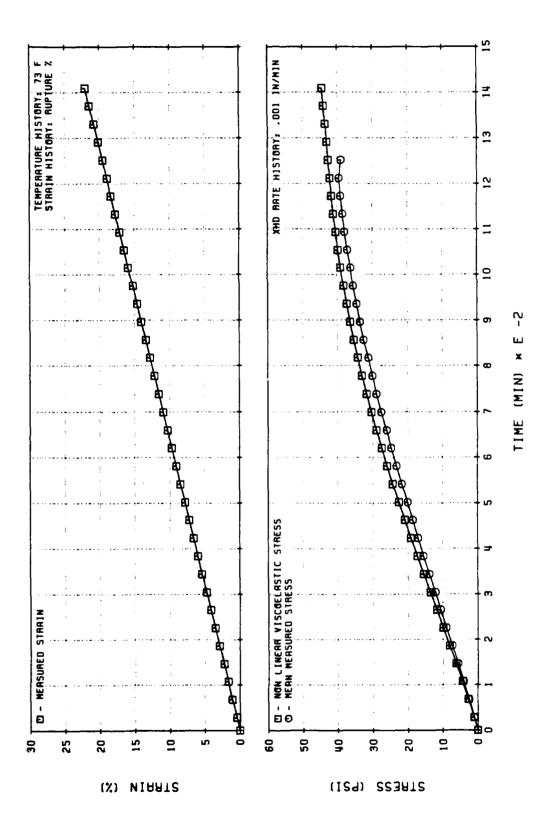
in which  $a_F = a_F$  ( $\epsilon^{\circ}$ ,  $\epsilon^{\circ}_m$ , S) plays the role of a softening function, reminiscent of the Mullins-Tobin approach.

## 4.2.3.3 Stress Predictions

The current version of the nonlinear model developed by R. Schapery may be used to predict the response of solid propellants with a rather remarkable degree of accuracy, as may be seen in Figures 87 to 94, which are sample cases of the isothermal tests considered in the program. The first two plots (Figures 87 and 88) correspond to the highest and lowest constant rate tests (Text continued on page 216.)



Dr. Schapery's Nonlinear Viscoelastic Stress Predictions for UTP-19,360B 400/1777 Constant Rate Test Data (Code No. 1) Figure 87.



Dr. Schapery's Nonlinear Viscoelastic Stress Predictions for UTP-19,360B 400/1777 Constant Rate Test Data (Code No. 1) Figure 88.

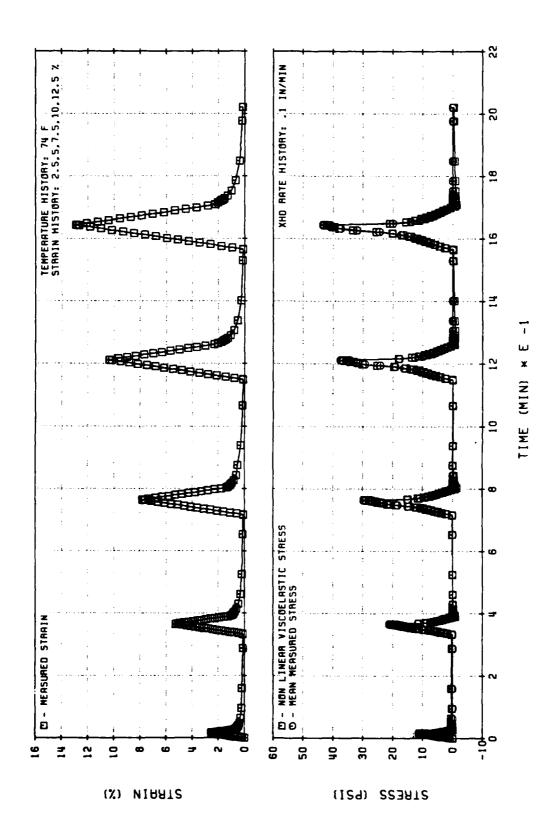


Figure 89. Dr. Schapery's Nonlinear Viscoelastic Stress Predictions for UTP-19,360B 400/1777 Multiple Loading Test History Data (Code No. 5)

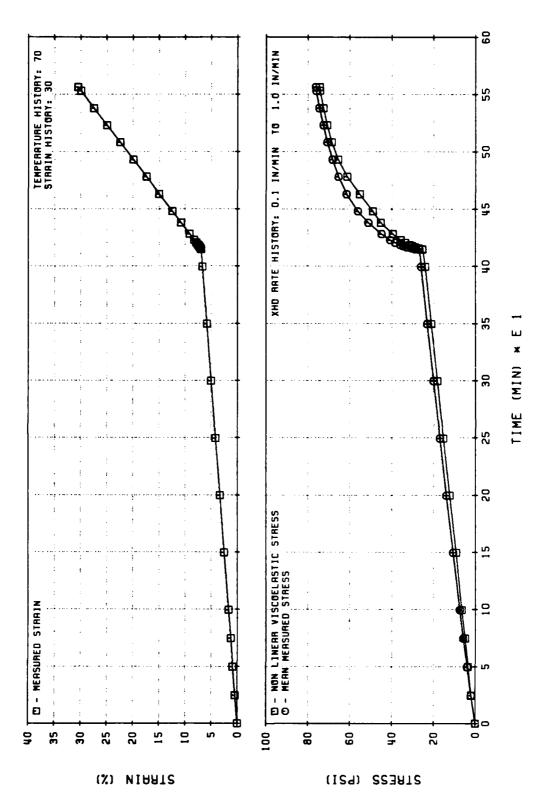


Figure 90. Dr. Schapery's Nonlinear Viscoelastic Stress Predictions for UTP-19,360B 400/1777 Two Rate Test Data (Code No. 3)

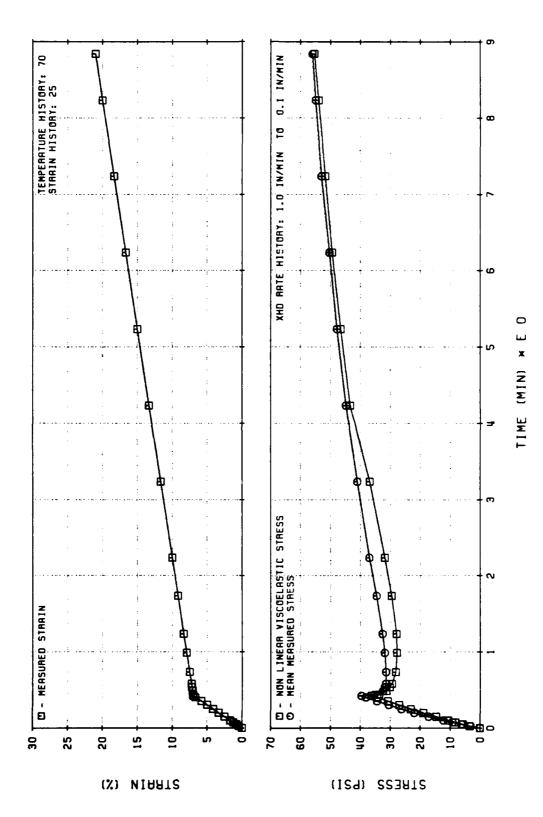
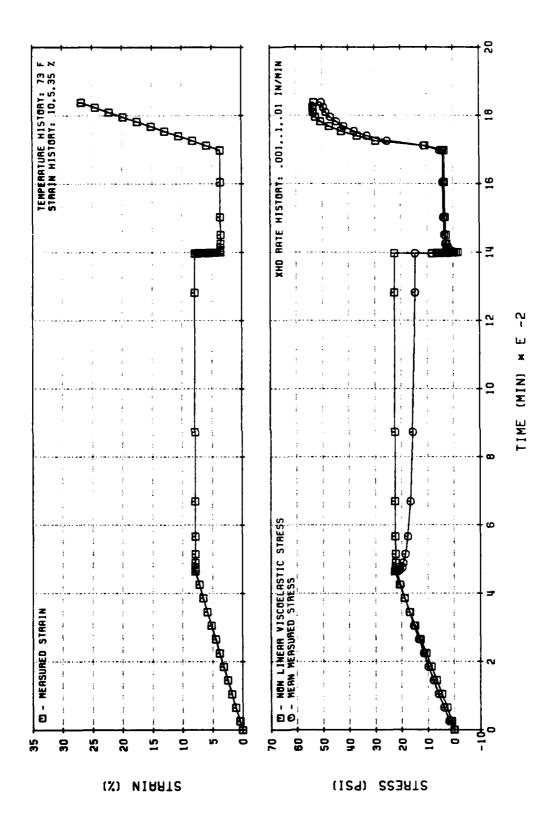
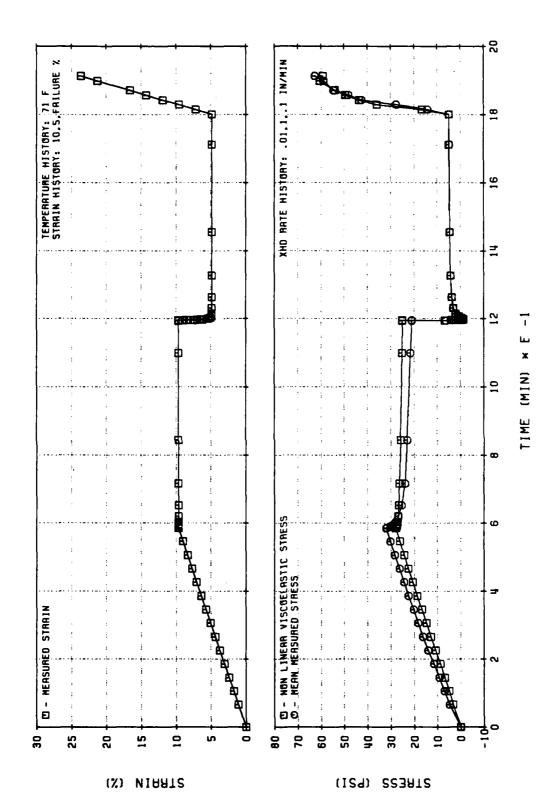


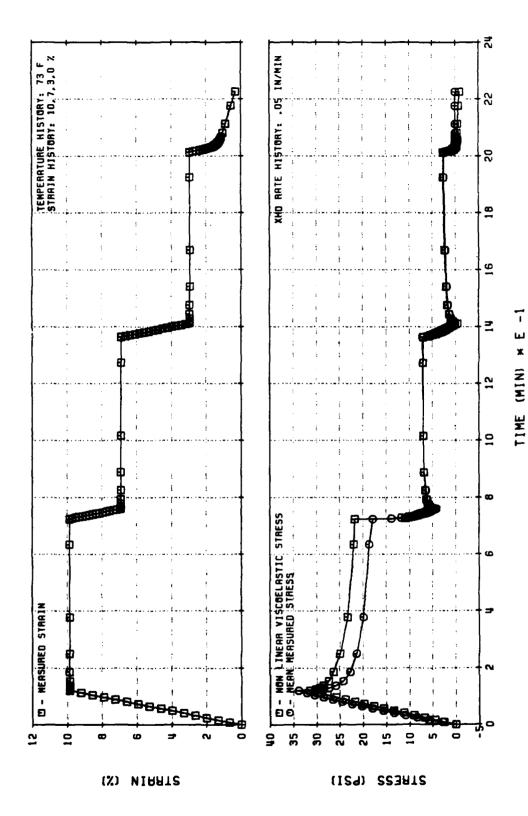
Figure 91. Dr. Schapery's Nonlinear Viscoelastic Stress Predictions for UTP-19,360B 400/1777 Two Rate Test Data (Code No. 3)



Dr. Schapery's Nonlinear Viscoelastic Stress Predictions for UTP-19,360B 400/1777 Similitude Test History Data (Code No. 12) Figure 92.



Dr. Schapery's Nonlinear Viscoelastic Stress Predictions for UTP-19,360B 400/1777 Similitude Test History (Code No. 12) Figure 93.



Dr. Schapery's Nonlinear Viscoelastic Stress Predictions for UTP-19,360B 400/1777 Three Step Relaxation Test History (Code No. 13) Figure 94.

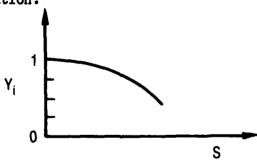
(Test No. 1) available in the data for which the difference between theory and test is greatest. Figure 89 shows the saw tooth test (Test No. 5) at constant rate with increasing strain peaks. Figures 90 and 91 pertain to the dual-rate tests (Test No. 3) Results for the short- and long-duration similitude tests (Test No. 12) are given in Figures 92 and 93, and Figure 94 includes a three-step relaxation test.

Finally, it is important to mention that a complete characterization of UTP-19,360B was also carried out using  $L_{\rm X}$  = 1 (the value leading to equation (35)), and the ensuing response predictions were very close to those obtained with  $L_{\rm X}$  = 0.85; only the low-to-high dual rate test of Figure 100 was predicted somewhat better with  $L_{\rm X}$  = 0.85.

# 4.2.3.4 Material Characterization

In evaluating the material constants and property functions, the following observations may be valuable:

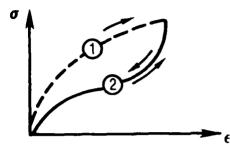
(1)  $Y_1$  appears to be a decreasing and concave down function of damage, as presented in the following figure, its variation being brought about by vacuole formation.



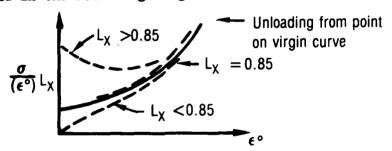
- (2) The function  $S_r$ , which provides a certain measure of damage, increases as a direct result of a reduction in the number of polymer chains supporting the internal stresses; the larger  $S_r$ , the higher the stress on each chain.
- (3) For UTP-19,360B, the state of damage is essentially constant during unloading and reloading, and the shape of the so-called backbone curve resembles the stress-strain curve for rubber, which is of the form:

$$Y_3 \mid \epsilon \circ \mid 0.85$$
 (36)

as shown in the sketch below, in which the steepness increased with increasing  $S_{\mbox{\scriptsize r}}$ .



- (1) Loading portion
- (2) Unloading and reloading portion (backbone curve)
- (4) The selection of  $L_{\rm X}$  can be made by plotting unloading data in the form suggested in the following diagram.



Noting that the quantity:

$$\frac{\sigma}{(\epsilon^{\circ})^{L_{X}}}$$

resembles a secant modulus, and that for most tests of UTP-19,360B  $L_{\rm X}$  = 0.85 produced a finite limiting value as  $\epsilon^{\rm O}$  approached zero, it is suggested that  $L_{\rm X}$  be found in this fashion for other propellants.

(5) For constant-rate tests, one has:

$$\epsilon^{\circ} = \epsilon_{m}^{\circ}$$
 $\lambda = 1$ 

and thus, from equation (29):

Also:

$$S_r = \max(S_r)$$

from which:

$$Y_3 = S_r$$

Equation (34) then reduces to:

$$\sigma = 1.861 \text{ Y}_1 \text{ S}^{0.637} (\epsilon_{\text{m}}^{0})^{0.463}$$
 (37)

with  $Y_1 = Y_1$  (S) given by equation (26). For very small damage:

$$Y_1$$
 (S) = 1

so that equation (36) becomes:

$$\sigma \approx 1.861 \text{ s}^{0.637} (\epsilon_{\text{m}}^{0})^{0.463}$$
 (38)

in which the stress increases with damage, probably because of molecular chain stiffening due to an increase in stress per chain.

With the foregoing observations in mind, determining the material properties can be accomplished as follows:

- (1) The exponent, n, appearing in the relaxation function, is obtained from relaxation-modulus data.
- (2) The normalized coefficient,  $E_r$ , entering the relaxation modulus, is determined to make unloading curve 2 in the figure above pass through the origin.

- (3) The exponent, q, present in the definition of the damage parameter, is evaluated using equation (37) and two constant-rate tests at small values of damage.
- (4) The function Y₁ is obtained by curve-fitting equation (36) to constant-rate tests over all strains out to failure.
- (5) Experience to date indicates that the function  $Y_2$  is independent of S and  $\epsilon_m$ , and therefore equation (30) may be used instead of the more general form of equation (37).
- (6) The backbone curve Y₃ is determined using unloading and reloading data like those available in a cyclic test whose first peak strain is the largest.
- (7) Finally the correction factor,  $\lambda$ , can be ascertained from a relaxation test at a large strain level.

### 4.2.3.5 Multiaxial Generalization

A micromechanics model has been developed which predicts the form of equation (30), and it is presently being used to develop a multiaxial form of the theory.

#### 4.2.4 M. Gurtin's Theories for Nonlinear Viscoelastic Materials

Four essentially different approaches have been followed by M. Gurtin in trying to predict the response of solid propellants that exhibit damage. The stress-softening theory appears to be the most accurate of the four laws as will be pointed out.

#### 4.2.4.1 Original Model

The one-dimensional stress-strain law for materials undergoing internal damage was based on the hypothesis that the state of damage at any time is completely characterized by the maximum strain,  $\epsilon_{\rm m}$ , that the material has experienced.

$$\epsilon_{\mathbf{m}}(\mathbf{t}) = \max \epsilon(\mathbf{s})$$
 (39)

The stress,  $\sigma$ , is given by a constitutive equation of the form:

$$\sigma(t) = g[\epsilon(t), \epsilon_m(t)]$$
 (40)

and it depends only on the current values of strain and damage. Such an equation is, of course, rate-independent.

In this theory, if the maximum strain occurs at the present time, then:

$$\epsilon_{\rm m}$$
 (t) =  $\epsilon$ (t), (41)

and equation (39) reduces to:

$$\sigma = G(\epsilon_m) = g(\epsilon_m, \epsilon_m)$$
 (42)

The stress-strain curve:

$$\sigma = G(\epsilon_{\mathbf{m}}) \tag{43}$$

is called the virgin curve and is traced out in an experiment with monotonically increasing strain.

Using the virgin curve, equation (39) may be rewritten in the form:

$$\sigma = F(\xi, \epsilon_{m}) G(\epsilon_{m})$$
 (44)

with:

$$\xi = \frac{\epsilon}{\epsilon_{\rm m}} \tag{45}$$

the relative strain, and:

$$F(\xi, \epsilon_{m}) = \frac{g(\epsilon_{m}, \epsilon_{m})}{G(\epsilon_{m})}$$
(46)

The function  $F(\xi, \epsilon_m)$  is called the damage curve at the damage level  $\epsilon_m$ , and is such that:

$$F(1, \epsilon_{\mathbf{m}}) = 1 \tag{47}$$

In some situations of interest  $F(\xi, \epsilon_m)$  is independent of  $\epsilon_m$ :

$$F(\xi, \epsilon_{m}) = F(\xi) \tag{48}$$

when this is so,  $F(\xi)$  is referred to as the master damage curve, and equation (43) reduces to:

$$\sigma = F(\xi)G(\epsilon_{\mathbf{m}}) \tag{49}$$

As pointed out previously, this is a rate-independent theory, and as such, cannot be used for loading rates that differ much from that used to determine the damage function. This situation was remedied by changing the stress-strain law to the one described next.

### 4.2.4.2 Nonlinear Model Based on Stress Softening

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To develop a simple theory of stress softening which allows for rate effects and which returns to Mullins' original idea of using the past stress maximum as the damage parameter, two fundamental ingredients are considered. The first is the virgin stress, S, which represents the stress the material would experience in the absence of softening. This stress is assumed governed by a constitutive equation of the type encountered in linear viscoelasticity. The second ingredient is a damage function, F, which gives the true stress,  $\sigma$ , when the virgin stress, S, and its past maximum  $S_m$ , are known.

The one-dimensional form of the constitutive law for a classical linear viscoelastic material is given by:

$$\sigma(t) = \int G(t - \tau) \dot{\epsilon}(\tau) d\tau$$
 (50)

in which  $\sigma$  (t) is the stress;  $\epsilon$  (t), the strain; and G(t), the relaxation function. It is further assumed that  $\epsilon$  (t)=0, prior to t = 0.

The generalization of equation (50) is begun by defining the quantity:

$$S(t) = \int_{-\tau}^{t} G(t - \tau) \dot{\epsilon}(\tau) d\tau$$
 (51)

which is called the virgin stress and which represents the stress that would be present in the absence of softening. It is assumed that the extent of softening is governed by a constitutive equation giving the true stress,  $\sigma(t)$ , when S(t) and its past maximum are known.

$$S_{m}(t) = \max S(\tau)$$

$$0 < \tau < st$$
(52)

Without loss of generality, this constitutive equation is written in the form:

$$\sigma = S_m F\left(\frac{S}{S_m}, S_m\right)$$
 (53)

and it is assumed that the damage function, F, satisfies the following conditions:

$$F(1, S_m) = 1$$
 (54)  
 $F(x, S_m) < x$  for  $x < 1$ 

These restrictions imply that:

$$\sigma(t) \leq S(t), \tag{55}$$

also that:

$$\sigma_{\mathbf{m}}(\mathbf{t}) = S_{\mathbf{m}}(\mathbf{t}), \tag{56}$$

and, that the following conditions are equivalent:

i) 
$$\sigma(t) = S(t)$$
  
ii)  $S(t) = S_m(t)$   
iii)  $\sigma(t) = \sigma_m(t)$  (57)

where  $\sigma_{\rm m}$  is the past stress maximum, defined analogically to  $S_{\rm m}$ . The inequality equation (54) asserts that the material actually softens, while equation (56) indicates that this softening occurs when and only when  $S(t) < S_{\rm m}(t)$  (or equivalently  $\sigma(t) < \sigma_{\rm m}(t)$ ). The results of equation (55) and (56) show that one may equally well use the true stress,  $\sigma(t)$ , as the damage parameter.

$$S_{m}(t) = S(\xi) \tag{58}$$

thus using equation (57) in (52), and recalling that equation (53):

$$\sigma(\xi) = S(\xi) F(1, S_m) \equiv S(\xi)$$
 (59)

which, by virtue of (57) and the definition of  $S_m$ , implies that:

$$\sigma(\xi) = S(\xi) = S_m(t) > S(\lambda) > \sigma(\lambda); \quad 0 < \lambda < t S$$
 (60)

proving equations (54) and (55) and the first two equations of equation (57). To establish the third relation in equation (57), note that if:

$$S_m(t) = S(t)$$

which implies that:

$$S_m(t) = \sigma(t)$$

because of equation (58); then, since

$$S_{\mathbf{m}} = \sigma_{\mathbf{m}} \tag{61}$$

one would have that:

$$\sigma_{m}(t) = \sigma(t)$$
.

Conversely, if

$$\sigma(t) = \sigma_{m}(t),$$

then:

$$S(t) > \sigma(t) = \sigma_m(t) = S_m(t)$$

so that:

$$S(t) = S_m(t)$$
.

Returning to the constitutive equation (52), it is interesting to consider the special case in which the damage function depends only on  ${\rm S/S}_{\rm m}$ :

$$F\left(\frac{S}{S_m}, S_m\right) = F\left(\frac{S}{S_m}\right);$$
 (62)

which is a master damage curve of the type considered in the rate-independent model discussed previously.

When the virgin stress obeys an elastic stress-strain relation:

$$S = E \epsilon \tag{63}$$

then:

$$S_{\mathbf{m}} = \mathbf{E}\epsilon_{\mathbf{m}} \tag{64}$$

in which  $\epsilon_{\rm m}$  is the past strain-maximum, and equation (52) yields:

$$\sigma = \mathbb{E}\epsilon_{\mathbf{m}} \ \mathbb{F}\left(\frac{\epsilon}{\epsilon \ \mathbf{m}} \ , \ \mathbb{E}\epsilon_{\mathbf{m}}\right) \tag{65}$$

so that, defining:

$$F^{*}\left(\frac{\epsilon}{\epsilon_{m}}, \epsilon_{m}\right) \equiv F\left(\frac{\epsilon}{\epsilon_{m}}, E\epsilon_{m}\right)$$
 (66)

leads to the starting assumption of Gurtin and Francis (Reference 26):

$$\sigma = \mathbb{F}^{\mathfrak{g}}\left(\frac{\epsilon}{\epsilon_{\mathfrak{m}}}, \ \epsilon_{\mathfrak{m}}\right) \mathbb{E}\epsilon_{\mathfrak{m}} \tag{67}$$

presented earlier as the rate-independent model.

Although implicit in equation (52) is the assumption that the functional form of F would be the same for unloading conditions as for reloading, it was found experimentally that a different damage function is needed for each of these processes. Actually, there is more than one way of obtaining the same damage function. For TP-H1011, for instance, the following procedure was employed.

Considering the strain history shown in Figure 95, on the loading portion we have:

$$S(t) = \epsilon \int_{0}^{\epsilon} G(\tau) d\tau \quad \text{for } t \leq T$$
 (68)

hence, S(t) increases monotonically and, by equation (55):

$$Sm(t) = S(t) = \sigma(t)$$
 (69)

and, upon unloading, the past maximum of S is the true stress:

$$\sigma_{\rm m} = \sigma(T), \qquad T < t < 2T$$
 (70)

Further, by equation (50):

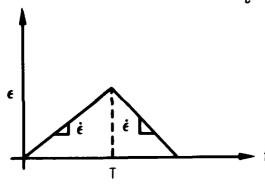
$$S(t) = |\dot{\epsilon}| \left\{ \int_{0}^{T} G(t - \tau) d\tau - \int_{T}^{t} G(t - \tau) d\tau \right\}$$
 (71)

or, equivalently:

$$S(T + t) = G(t) - \sigma(t)$$
 (72)

with:

$$G(t) = |\dot{\epsilon}| \int_{t}^{t} G(\tau) d\tau$$
(73)



Letting  $\sigma_1(t)$  and  $\sigma_2(t)$  denote the true stress during loading and unloading, respectively, with t in  $\sigma_2(t)$  measured from the time T at which unloading begins, equations (52) and (67) yield the simple formula:

Figure 95. Strain History Used To Characterize the Damage Function

$$\frac{\sigma_2(t)}{\sigma_m} = F\left[\frac{G(t) - \sigma_1(t)}{\sigma_m}, \sigma_m\right] (74)$$

Thus, summarizing, the stress-softening approach to damage is described through the following constitutive equation:

$$\sigma(t) = S_m F\left(\frac{S}{S_m}, S_m\right)$$
 (75)

where

$$S(t) = \int_{0}^{t} G(t - \tau) \frac{d\epsilon}{d\tau} (\tau) d\tau$$

and in which the damage function, F, may be determined from saw-tooth tests with increasing peak strains and with sufficiently long rest periods between cycles. For conditions of reloading, F is given by:

$$\frac{\sigma_2}{\sigma_m} = F \left[ \frac{\sigma_1 \ (t)}{\sigma_m} \ , \ \sigma_m \ \right] \tag{76}$$

while for unloading the following form is employed:

$$\frac{\sigma_2}{\sigma_m} = F \left[ \frac{G(t) - \sigma_1(t)}{\sigma_m}, \sigma_m \right]$$
 (77)

Typical curves for unloading and reloading damage functions of TP-H1011 are included as Figure 96 and 97, respectively.

Finally, we point out that the use of this approach to predict the response of TP-H1011 yielded results that were far more accurate than those obtained with any of the other theories in their original form.

#### 4.2.4.3 Nonlinear Models Based on Maximum Strain.

A series of constitutive relations based on the past maximum strain have been proposed by M. Gurtin. The precursor of these relations took the form:

$$\sigma(t) = \int_{0}^{t} G(t - \tau) \frac{de}{d\tau} (\tau) d\tau \qquad (78)$$

where G represents the relaxation modulus, and the function e was expressed as a product of the reduced damage function, F, and the virgin-response function, g, in the following way:

$$e = F\left(\frac{\epsilon}{\epsilon_m}, \epsilon_m\right) g(\epsilon_m)$$
 (79)

with:

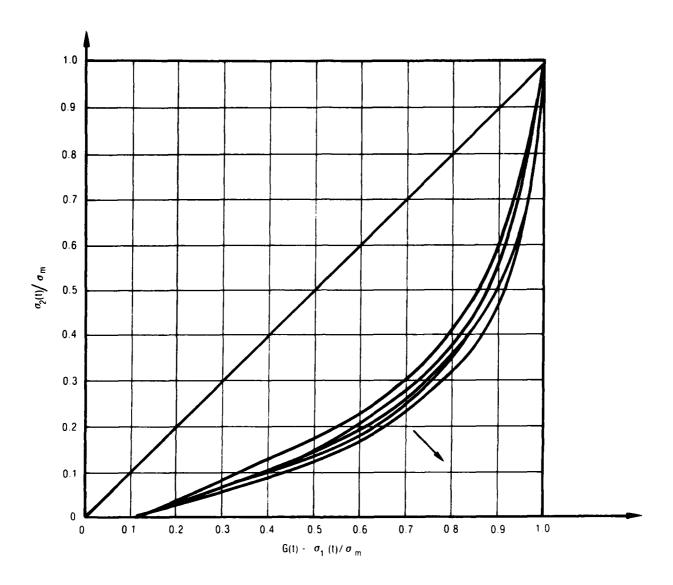


Figure 96. Damage Function for Unloading (TP-H1011)
28896

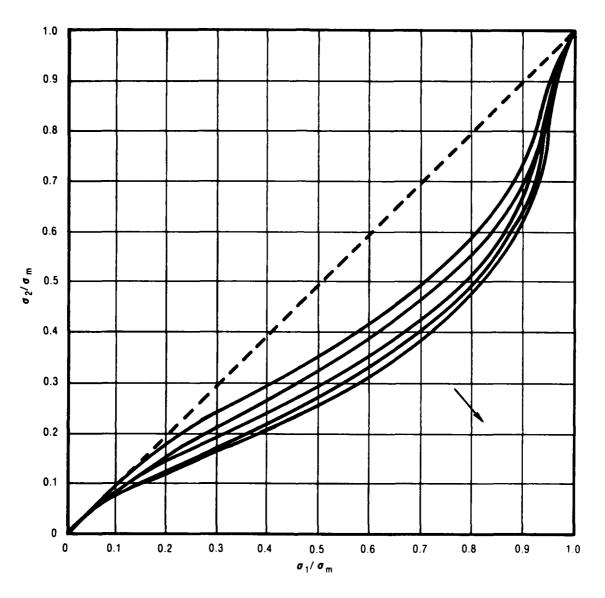


Figure 97. Damage Function for Reloading (TP-H1011) 28892

$$\mathbf{F}(\mathbf{1}, \epsilon_{\mathbf{m}}) = \mathbf{1} \tag{80}$$

and

$$\epsilon_{\mathbf{m}} = \max \epsilon(\tau)$$

$$0 < \tau < \mathbf{t}$$
(81)

so that, during virgin response, since:

$$\epsilon = \epsilon_{\mathbf{m}}$$
 (82)

one had:

$$\sigma(t) = \int_{0}^{t} G(t - \tau) \frac{dg(\epsilon)}{d\epsilon} \frac{d\epsilon(\tau)}{d\tau} d\tau$$
 (83)

and, by taking:

$$g(\epsilon) = \sum_{k=1}^{k=K} A_k \epsilon^k$$
 (84)

the best values for the  $a_k$ 's could be determined using least squares and the data of all constant-rate tests.

To characterize the reduced damage function, F, involved the determination of a creep function, J, solution of:

$$\int_{0}^{t} G(t-\tau) \frac{dJ(\tau)}{d\tau} d\tau = 1$$
(85)

and, such that:

$$e(t) = \int_{0}^{t} J(t - \tau) \frac{d\sigma(\tau)}{d\tau} d\tau$$
 (86)

Thus, taking the reduced damage function, F, in the form:

$$F(x, y) = F_1(x) F_2(x,y)$$
 (87)

with

$$M - 1$$

$$F_1(x) = x^M + \sum_{m=1}^{M-1} d_m(x^m - x^M)$$
(88)

$$F_2(x,y) = 1 + \sum_{p=1}^{p} b_p y^p [x - x^Q + \sum_{q=2}^{Q-1} c_q (q - Q)]$$
 (89)

and equating equations (71) and (75), the coefficients entering F were to be determined using least squares and all the saw-tooth data with increasing strain peaks.

When this constitutive law was applied to UTP-19,360B data, it was deemed necessary to change the form of the function e, because of the large errors observed in the predicted response.

The last of a sequence of modifications yielded the following stress-strain law:

$$\sigma(t) = \int_{0}^{t} G(t - \tau) \left\{ K\left(\epsilon_{m}, \dot{\epsilon}_{m} \frac{d}{d\tau} \left[ F\left(\frac{\epsilon}{\epsilon_{m}}, \epsilon_{m}\right) \epsilon_{m} \right] \right\} d\tau \qquad (90)$$

where, as before, G was the relaxation modulus, and:

$$\mathbf{F} (1, \epsilon_{\mathbf{m}}) = 1 \tag{91}$$

with:

$$\epsilon_{\mathbf{m}} = \max \ \epsilon(\tau)$$

$$0 < \tau < \mathbf{t}$$
(92)

In the present case, the virgin response was given by:

$$\sigma(t) = \int_{0}^{t} G(t - \tau) K (\epsilon_{m}, \dot{\epsilon}_{m}) \frac{d\epsilon_{m}}{d\tau} d\tau$$
 (93)

while, the damage response, for which  $\epsilon_{\mathrm{m}}$  remains constant, took the form:

$$\sigma(t) = K (\epsilon_m, 0) \int_{0}^{t} G(t - \tau) \frac{\partial F}{\partial x} (x, y) d\tau$$
 (94)

in which:

$$K(\epsilon, \dot{\epsilon}) = 1 + A_1 (\epsilon - \epsilon_0) + A_2 (\epsilon - \epsilon_0)^2 + A_3 (\epsilon - \epsilon_0)^3 + \dot{\epsilon} (B_1 \epsilon + B_2 \epsilon^2 + B_3 \epsilon^3)$$
(95)

$$F(x,y) = \alpha(x) \left[ 1 + (D_5 y + D_7 y^2) \left\{ x - x^3 + D_6 (x^2 - x^3) \right\} \right]$$
 (96)

$$\alpha(x) \equiv x^5 + \sum_{m=1}^{4} D_m (x^m - x^5)$$
 (97)

with:

$$x = \epsilon/\epsilon_{\rm m} \tag{98}$$

and:

$$\mathbf{y} = \epsilon_{\mathbf{m}} \tag{99}$$

A set of stress predictions obtained for UTP-19,360B, with the resulting version of the theory, is included in Figures 98 through 104. Figure 98 is for the high-to-low dual rate test, Figures 99 through 101 are segments of the long-duration similitude test, and Figure 102 through 104 are segments of the three step relaxation test.

Since the dependence of the function K on the strain rate was felt to be artificial, the treatment of damage was revised in the manner explained next.

### 4.2.4.4 Current Model

The latest version of M. Gurtin's nonlinear stress-strain law is based on a strain-dependent relaxation function and has the form:

$$\sigma(t) = \int_{0}^{t} G\left[\epsilon(\tau), \tau\right] \dot{\epsilon}(t - \tau) d\tau$$
 (100)

where:

$$G(\epsilon, t) = G_r(t) + G_c(\epsilon, t)$$
 (101)

G = relaxation modulus

 $G_c$  = correction modulus, defined as:

$$G_{c}(\epsilon, t) = \sum_{n=1}^{N} A_{n}(\epsilon)(e)^{-t/\tau_{n}}$$
 (102)

$$A_{n}(\epsilon) = \sum_{p=1}^{P} A_{np} \epsilon^{p}$$
(103)

For this material, the virgin curve (  $\epsilon = \epsilon_{\rm m} = \epsilon$ ) is:

$$\sigma = \sigma_{\mathbf{r}} + \sigma_{\mathbf{c}} \tag{104}$$

with  $\sigma_r$ , the linear viscoelastic stress:

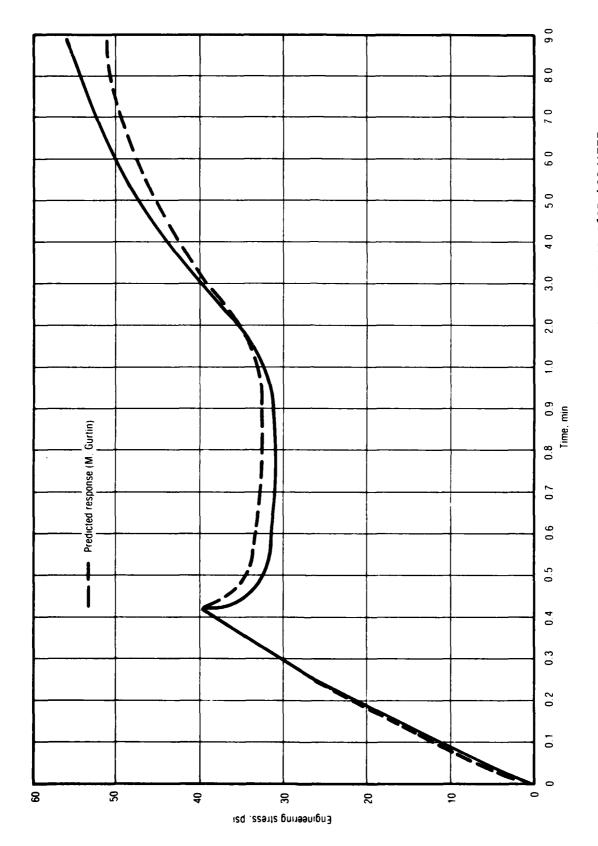
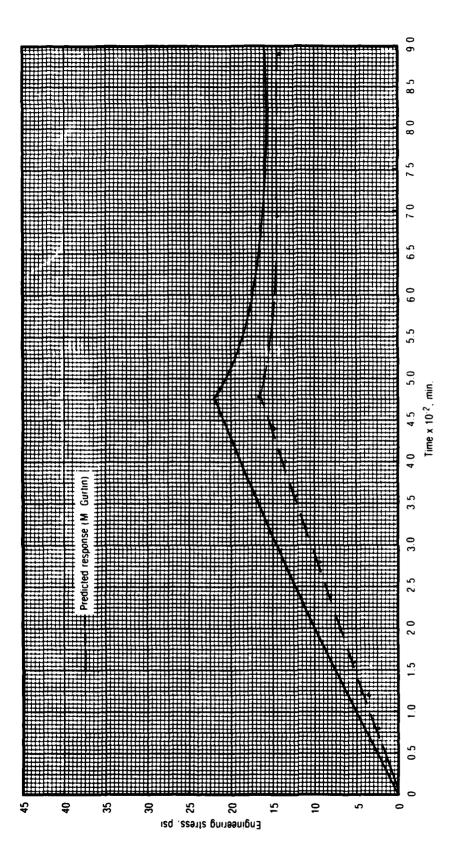


Figure 98. Two-Rate Loading (1 in./min to 0.1 in./min) of UTP-19,360B-400/1777



Relaxation-Unload-Reload of 6-in. Bar of UTP-19,360B-400/1777

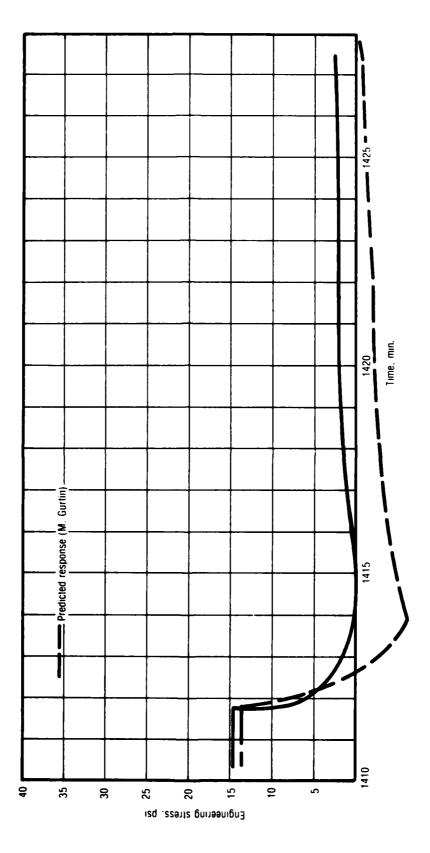


Figure 100. Relaxation-Unload-Reload of 6-in. Bar of UTP-19,360B-400/1777

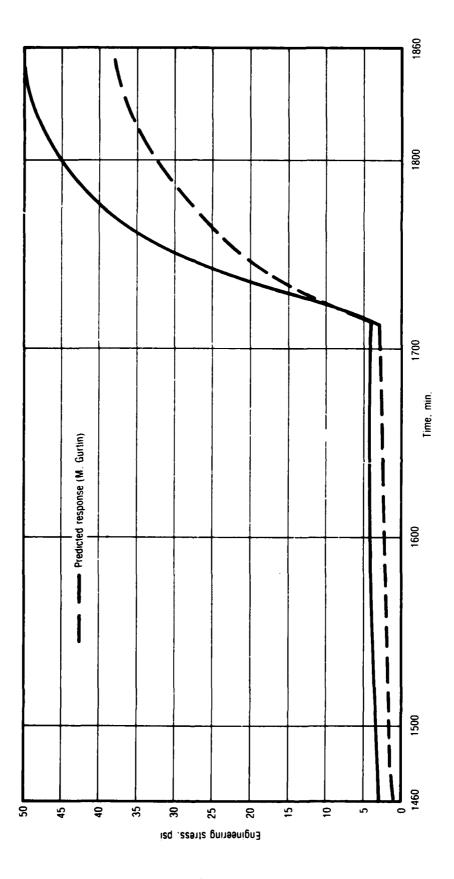


Figure 101. Relaxation-Unload-Reload of 6-in. Bar of UTP-19,360B-400/1777

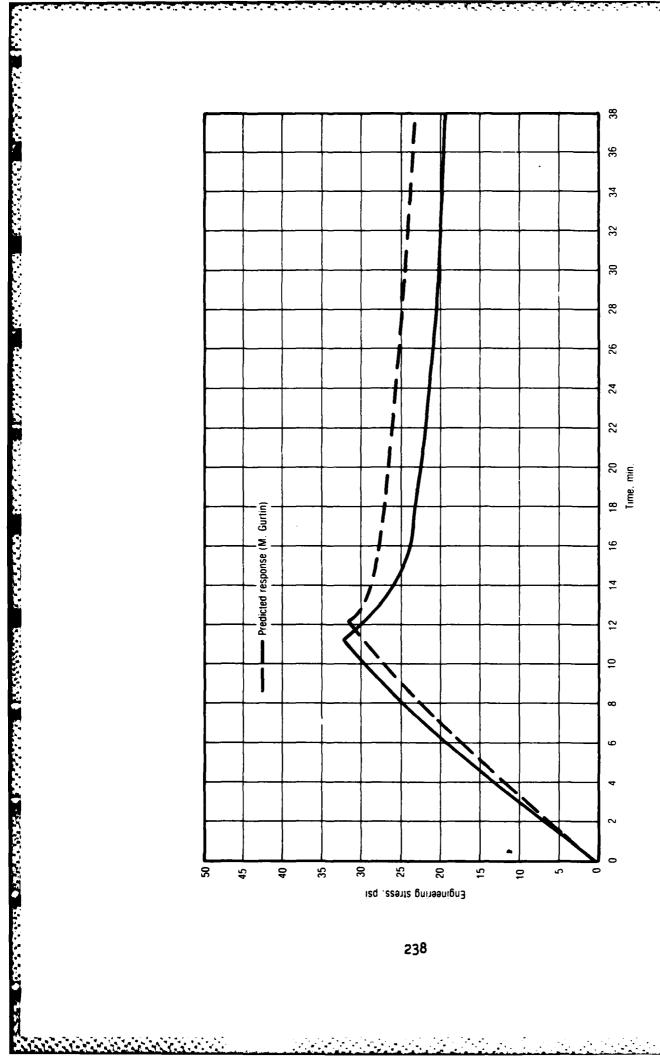


Figure 102. Three-Step Relaxation of 6-in. Bar of UTP-19,360B-400/1777

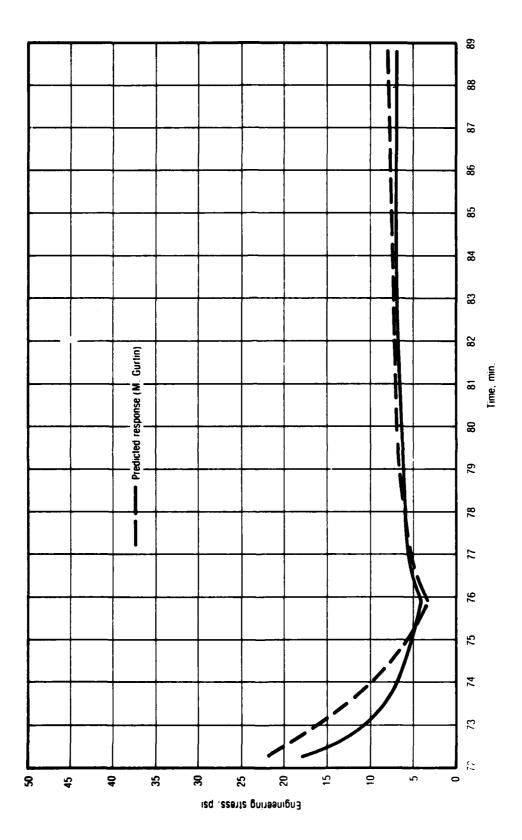
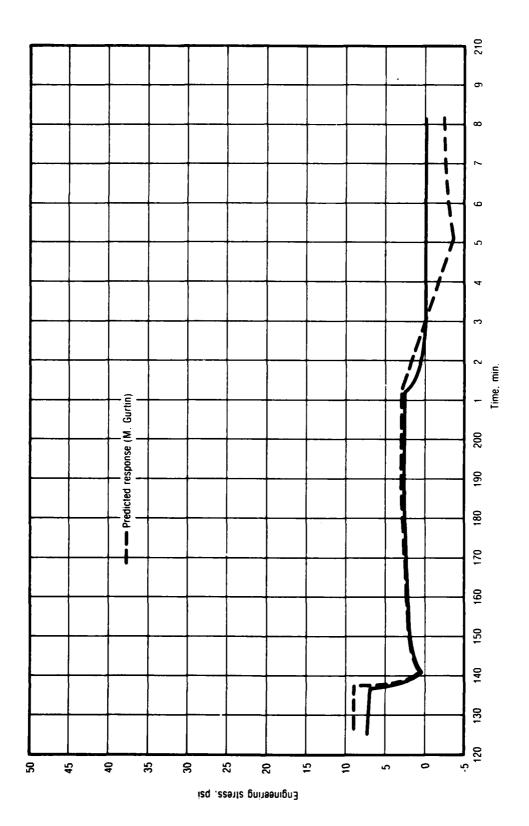


Figure 103. Three-Step Relaxation of 6-in. Bar of UTP-19,360B-400/1777



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Three-Step Relaxation of 6-in. Bar of UTP-19,360B-400/1777 Figure 104.

$$\sigma_{\mathbf{r}}(\mathbf{t}) = \int_{0}^{\mathbf{t}} G_{\tau}(\tau) \stackrel{\cdot}{\epsilon} (\mathbf{t} - \tau) d\tau$$
 (105)

and the correction stress,  $\sigma_{\rm C}$ , given by:

$$\sigma_{\mathbf{c}}(\mathbf{t}) = \int_{0}^{\mathbf{t}} G_{\mathbf{c}} \left[ \epsilon(\tau), \tau \right] \dot{\epsilon}(\mathbf{t} - \tau) d\tau$$
 (106)

Hence, to characterize the virgin response, only constant rate tests need be employed. In this instance,  $\sigma$  and  $\sigma_r$  are known, so that from equation (88),  $\sigma_c$  may be computed, and equated to equation (90) using the fact that  $\dot{\epsilon}$  is a constant; i.e.:

$$\sigma(t) - \sigma_r(t) = \sigma_c(t) \equiv \int_{\Omega} G[\epsilon(\lambda), \lambda] d\lambda$$
 (107)

which, upon recalling equations (91) and (92) becomes:

$$\sigma(t) - \sigma_r(t) = \sum_{n=1}^{N} \psi_n(t)$$
 (108)

where:

$$\psi_{n}(t) = \sum_{p=1}^{P} A_{np} \dot{\epsilon} \int_{0}^{t} \epsilon^{p(\lambda)} e^{-\lambda/\tau_{n}} d\lambda$$
 (109)

Furthermore, since for a constant-rate test:

$$\epsilon(\lambda) = \dot{\epsilon}\lambda$$
 (110)

it follows that:

$$\Psi_{n} (t) = \sum_{p=1}^{p} A_{np} \dot{\epsilon}^{p+1} \int_{0}^{t} \frac{p - \lambda \tau_{n}}{\lambda e} d\lambda$$
 (111)

and after integrating by parts:

$$\Psi_{n}(t) = \sum_{p=1}^{p} A_{np} \dot{\epsilon}^{p+1} \tau_{n}^{p+1} f_{p}^{\cdot}(t/\tau_{n})$$
(112)

with

$$f_0(x) = 1 - e^{-x}$$
 (113)  
 $f_p(x) = -x^R e^{-x} + pf_{p-1}(x)$ ;  $p = 1, ...p$ 

Clearly, equations (89) and (92) to (94) may be used to determine the coefficients  $a_{np}$  appearing in the definition of the correction modulus. The procedure suggested by M. Gurtin to accomplish this is as follows:

1. Take N tests with constant rates  $\epsilon_1$ ,  $\epsilon_2$ , . . . ,  $\epsilon_N$ ; and set:

$$\tau_{\underline{i}} = \frac{1}{\dot{\epsilon}_{\underline{i}}}; \ \underline{i} = 1, ..., N \tag{114}$$

- 2. Select the degree, P, of the series expansion of the correction modulus, as it appears in equation (87).
- 3. Use the  $\epsilon_1$  test and the approximation:

$$\sigma_{C}(t) = \Psi_{1}(\epsilon_{1}, t) \tag{115}$$

to find the a_{1p}.

4. Use the  $\dot{\epsilon}_2$  test and the approximation:

$$\sigma_{c}(t) - \Psi_{1}(\dot{\epsilon}_{2}, t) = \Psi_{2}(\dot{\epsilon}_{1}, t) \tag{116}$$

to find the a2p.

5. Use the  $\epsilon_3$  test and the approximation:

$$\sigma_{c}(t) - \Psi_{1} \stackrel{(\epsilon_{3}, t)}{\leftarrow} - \Psi_{2} \stackrel{(\epsilon_{3}, t)}{\leftarrow} = \Psi_{3} \stackrel{(\epsilon_{3}, t)}{\leftarrow} (117)$$

to find the  $a_{3P}$ , and so on for the  $a_{4P}$  . . .  $a_{NP}$ .

6. Iterate this procedure if necessary; that is, define:

$$\overline{\Psi}(\dot{\epsilon}, t) = \sum_{n=1}^{N} \Psi_n(\dot{\epsilon}, t)$$
 (118)

so that  $\Psi$  is known. For each ramp test, define:

$$\overline{\sigma}_{\mathbf{c}} = \sigma_{\mathbf{c}} - \overline{\Psi} \tag{119}$$

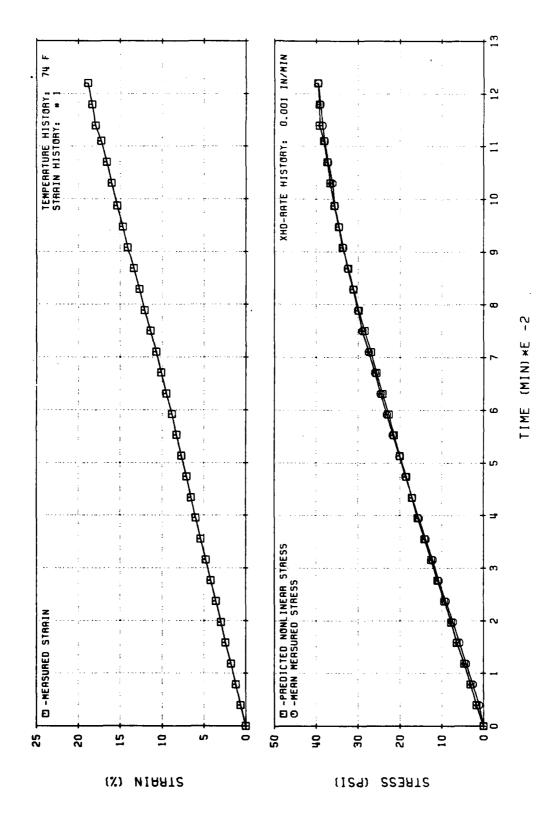
and repeat the above procedure using  $\sigma_c$  to find constants  $\overline{a_{nP}}$ . The new values of the  $a_{nP}$  are:

$$(\mathbf{A}_{np})_{new} = \mathbf{A}_{np} + \overline{\mathbf{A}}_{np} \tag{120}$$

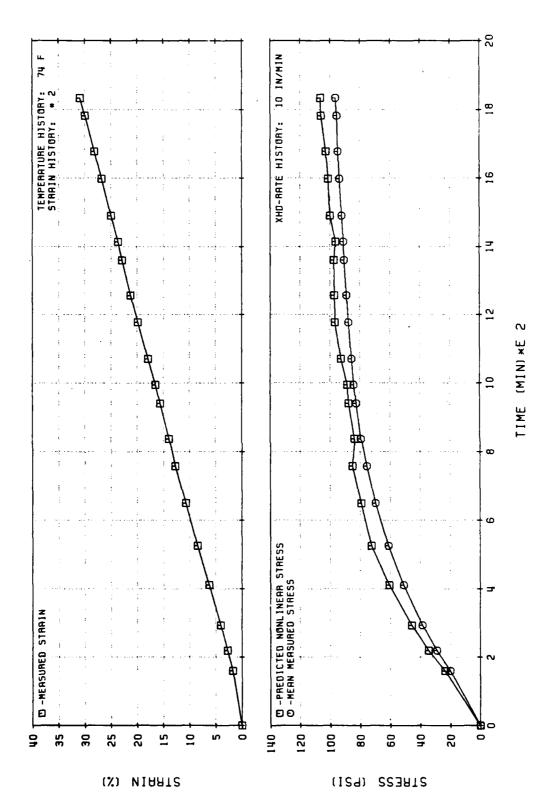
## 7. Repeat the process if necessary.

It is important to point out here that numerical difficulties may be encountered in applying this technique to characterizing the virgin response of solid propellants. In fact, some convergence problems were faced in connection with the UTP-19,360B data. Moreover, characterization of the damaged response calls for a large number of cyclic tests over a wide range of rates. This increases the convergence difficulties. The model was employed with the constant-rate tests only for this reason. Figures 105 to 108 show the results of the stress predictions obtained with the current version of the model. The first two plots correspond, respectively, to the lowest and highest rates available in the data base at ambient temperature.

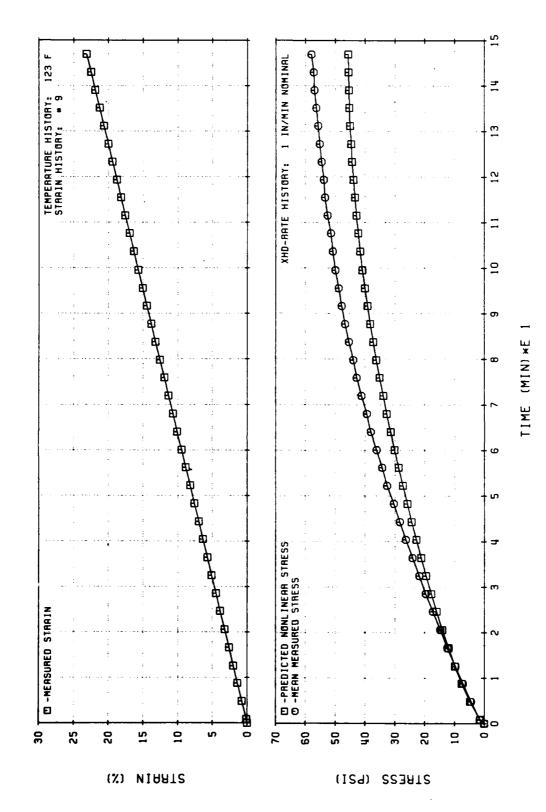
So far in the program, none of the models developed by Gurtin has taken into account the effects of temperature on propellant response. However, the



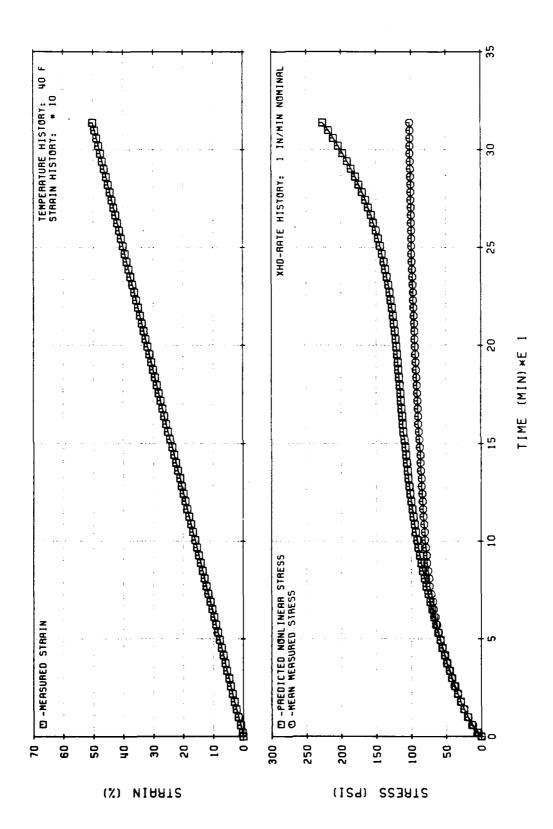
Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 at 0.001 in./min and 74 F (M. Gurtin's Theory) Figure 105.



Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 at 70 (M. Gurtin's Theory) Figure 106.



Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 at 123 F (M. Gurtin's Theory) Figure 107.



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Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 at 40 F (M. Gurtin's Theory) Figure 108.

time-temperature superposition principle was tested with the current version of the theory. The results appear in Figures 107 and 108 for the thermal tests at 123 and  $40^{\circ}\text{F}$ , respectively. The use of the superposition principle breaks down at a strain of about 39%. This was apparently due to the ambient temperature data base being limited to a low strain level. Also, it might be necessary to allow the relaxation function (G_C) to depend on the glass transition temperature (T_g)

$$G_c = G_c(\epsilon, T_g, t)$$

## 4.2.5 Russian Approach to Physically Nonlinear Viscoelastic Solids

The Russians have explored two approaches for characterizing damage effects in solid propellants (References 7 and 8). They are a general functional approach, and a kinetic equation of evolution for damage. Both approaches are based on internal-variable concepts, and either approach appears general enough to also incorporate cumulative damage and propellant response under multiaxial stress states. However, the general functional approach may be of little practical engineering value, because a very extensive testing program may be required to evaluate material parameters. This approach requires introduction of damage measures, which should reflect microstructural damage mechanisms, and a damage functional which characterizes the accumulation of damage or defects. The damage functional is then expanded into a series of multiple integrals in an analogous fashion to that followed by Green and Rivlin for nonlinear materials with fading memory. Herein lies the difficulty. Even assuming isotropy, four to six different types of multi-axial tests are required to evaluate the required material property functions for a first-order theory. Although the approach has theoretical merit and may even have some practical application in the future, its pursuit was abandoned in favor of the kinetic approach.

The essential feature of the kinetic approach is to introduce the degree of damage into the constitutive equations as a reduced-time parameter in the same way that temperature is introduced as a reduced-time parameter for the thermorheologically simple materials in linear thermoviscoelasticity. Damage is then defined in terms of some strength parameter of the material, and the degree of

damage is characterized through an equation of evolution for damage, as explained subsequently.

## 4.2.5.1 Original Model

The one-dimensional constitutive equation taken from the Russian literature by W. L. Hufferd as a means of predicting the response of physically nonlinear viscoelastic materials may be expressed by:

$$\sigma(t) = \int_{0}^{\infty} E(t' - \tau') \frac{d\epsilon}{d\tau}(\tau) d\tau$$
 (121)

where

 $\sigma = stress$ 

 $\epsilon$  = strain due to mechanically applied stress

E(t) = relaxation modulus

$$E(t) = E_e + E_2 t^{-n}$$

and

$$t' - \tau' = \int_{\tau}^{t} \frac{d\epsilon}{a_{\eta} \left[ \eta (\xi) \right]}$$
 (122)

represents the damage-reduced time, which is arrived at in the manner described next.

First, a normalized damage function  $\omega = \omega(t)$  is introduced through the following kinetic equation of evolution:

$$\frac{d\omega}{dt}(t) = h(\omega) f(t)$$
(123)

in which it is further assumed that:

$$f(t) = \int_{0}^{t} F(t - \tau) \phi \left[\sigma_{0}(\tau)\right] d\tau$$
 (124)

together with the conditions:

$$\omega(o) = 0$$

$$\omega(t^*) = 1$$
(125)

indicating that no damage exists at the initial state, and that failure occurs at time t*.

Next, equation (97) is integrated with  $\omega$  (o) = 0, leading to:

$$\int_{0}^{\omega} \frac{d\omega}{h(\omega)} = \int_{0}^{t} f(\tau) d\tau$$
 (126)

Setting  $t = t^{\alpha}$ , so that  $\omega(t^{\alpha}) = 1$ , and substituting equation (98) for  $f(\tau)$ , it is obtained that

$$\int_{0}^{t^{*}} d\xi \int_{0}^{\xi} F(\xi - \tau) \phi \left[\sigma_{0}(\tau)\right] d\tau$$

$$\int_{0}^{1} \frac{d\omega}{h(\omega)}$$
(127)

If now the function F(t) is assumed to have a power-law representation:

$$F(t) = F_0 t^m ag{128}$$

Equation (101) can be written in the form:

$$\int_{0}^{t^{*}} d\xi \int_{0}^{\xi} F_{0}(\xi - \tau)^{m} \phi \left[ \sigma_{0}(\tau) \right] d\tau$$

$$= 1$$

$$\int_{0}^{1} \frac{d\omega}{h(\omega)}$$
(129)

and integrating with respect to  $\xi$ , assuming that the order of integration may be interchanged, one arrives at:

$$\frac{\mathbf{F_0}}{1+\mathbf{m}} \int_{0}^{\mathbf{t}^*} (\mathbf{t}^* - r)^{1+\mathbf{m}} \phi \left[\sigma_0(\tau)\right] d\tau$$

$$\int_{0}^{\mathbf{t}^*} \frac{d\omega}{h(\omega)}$$
(130)

which, for the case where  $\sigma_0$  and  $\phi \left[\sigma_0\right] = \phi \left[\sigma_0\right]$  are constant, becomes:

$$\frac{F_0 \phi_0 (\sigma_0)}{(1+m) (2+m)} \frac{(t_0^*)^{2+m}}{1} = 1$$

$$\int_0^{\infty} \frac{d\omega}{h(\omega)}$$
(131)

where  $t_0^*$  is the time to failure under the constant stress  $\sigma_0$ . Thus equation (104), in this case, may be written as

$$(2 + m) \int_{0}^{t^{m}} (t^{m} - \tau)^{1 + m} \frac{d\tau}{(t_{0}^{m})^{2+m}} = 1$$

If the time to failure under a constant stress,  $\sigma_0$ , has the power-law representation:

$$\sigma_0^{\alpha} t_0 = constant = \beta$$
 (133)

then equation (106) can be put in the form:

$$\int_{0}^{t^{\frac{m}{2}}} (t^{\frac{m}{2}} - \tau)^{1+m} \sigma_{0}^{(\alpha(2+m))} d\tau = \frac{\beta^{2+m}}{2+m}$$
 (134)

so that, motivated by equations (106) and (108), the degree of damage accumulation may be introduced through the expression:

$$\eta(t) = (2 + m) \int_{0}^{t} (t - \tau)^{1 + m} \frac{d\tau}{(t_{0})^{2+m}}$$
(135)

in which, obviously:

$$\eta (0) = 0$$
 $\eta (1) = 1$ 
(136)

The function  $\eta(t)$  can be related to the damage function,  $\omega$ , by:

$$\eta = \frac{\int_{0}^{\omega} \frac{d\omega}{h(\omega)}}{\int_{0}^{1} \frac{d\omega}{h(\omega)}}$$
(137)

This means that  $\eta$  represents the relative damage in the load history for the power-law representation of  $t_0^*$ . From equations (107) and (108), equation (109) may be written as:

$$\eta(t) = \frac{2 + m}{\rho^{2+m}} \int_{0}^{t} (t - \tau)^{1+m} \sigma_{0}^{\alpha(2+m)} d\tau$$
(138)

and finally, the influence of damage is treated as a reduced variable by introducing the modified time, t', defined by

$$dt' = \frac{dt}{a\eta \left[\eta(t)\right]}$$
 (139)

on which equation (96) is based, and where the shift function due to damage,  $a_{ij}$ , depends on the material at hand.

Despite its rather appealing physical and mathematical foundations, the original model did not do any better than Linear Viscoelasticity when it was used to predict the response of either TP-H1011 or UTP-19,360B. In those instances, a linear expression:

$$a\eta = 1 - \eta \tag{140}$$

and an exponential form:

$$a_{\eta} = e^{-\eta}$$

were used for the damage shift function.

The partial failure of the Russian approach to reproduce solid-propellant behavior made it necessary to change certain aspects of the theory, as explained next.

Current Model. One revised version of the Russian stress-strain law takes the form:

$$\sigma(t) = \int_{0}^{t} E\left(\frac{t-\tau}{a\eta}\right) \frac{d\epsilon}{d\tau} (\tau) d\tau$$
 (141)

where  $a_{7}^{*}$  is a damage-related shift function, assumed to depend only on the current state of strain; specifically:

$$\mathbf{a}_{\eta}^* = \mathbf{a}_{\eta}^* \left[ \epsilon(\mathbf{t}), \quad \dot{\epsilon}(\mathbf{t}) \right] \tag{142}$$

Clearly, if:

$$E(t) = E_e + E_2 t^{-n}$$
 (143)

then equation (113) becomes:

$$\sigma(t) = E_{e} \epsilon(t) + (a_{\eta}^{*})^{n} E_{2} \int_{0}^{t} (t - \tau)^{-n} \frac{d\epsilon}{d\tau} (\tau) d\tau$$
 (144)

which resembles the classical approach of the softening function used as a stress-correction factor.

Another revised version of the Russian approach consists of retaining most aspects of the original law, but constant strain-rate data are employed to express the time to failure as:

$$\mathbf{t}_{0}^{*} = \left(\frac{\beta}{\epsilon}\right)^{\alpha} \tag{145}$$

and equation (108) is changed to:

$$\eta(t) = \frac{(2+m)}{\hat{\epsilon} t_0} \int_{0}^{t} (t-\tau)^{1+m} \phi(\tau) d\tau$$
 (146)

where:

$$\phi(t) = \epsilon(t) = \dot{\epsilon}t$$

and

$$\eta(t^*) = 1$$

Thus, evaluation of equation (117) at  $t = t^*$  yields:

$$\frac{(2+m)}{\epsilon_{t_0}} \int_{0}^{t^{\frac{m}{2}}} (t^{\frac{m}{2}} - \tau)^{\frac{1+m}{2}} \phi(\tau) d\tau = 1$$
(147)

which, through a change of variables and after some algebraic manipulations, may be integrated to:

$$(t_0)^{\frac{m}{2}} = 3 + m$$
 (148)

The solution for m, as a function of strain rate, is easy to obtain using equations (116) and (119) (as presented in Figure 109 for UTP-3001).

In much the same way, integration of equation (117) for the relative damage function,  $\eta(t)$ , leads to:

$$\eta(t) = \left(\frac{t}{t_0}\right)^{3 + m} \tag{149}$$

Hence, using constant strain-rate data to express the time to failure does simplify things, but a major assumption would still be needed regarding the form of the damage shift function as it was in the original set of equations. In this context, it is important to note that the linear expression:

$$\mathbf{a}_{\eta} = 1 - \eta \tag{150}$$

and the exponential form:

$$\mathbf{a}\,\eta = \mathbf{e}^{-\,\eta} \tag{151}$$

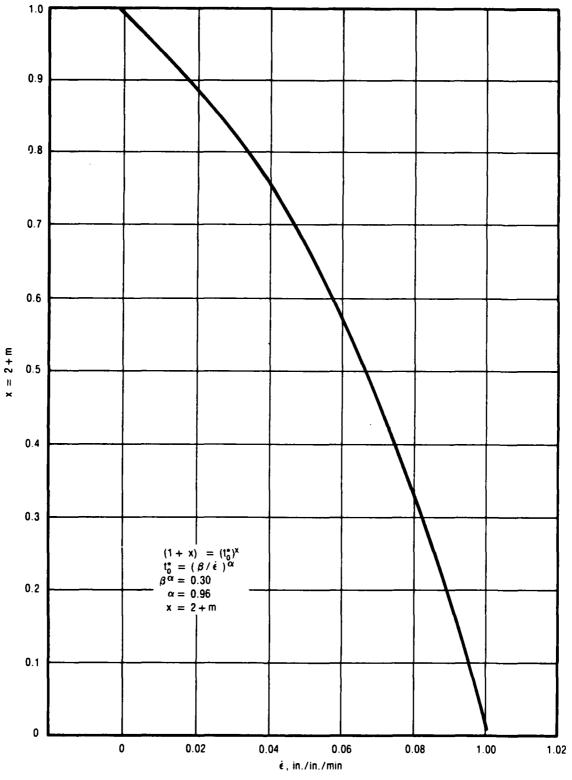


Figure 109. Solution for m as a Function of Strain Rate

were used for the damage shift function without success. For this reason, the modified version used to run the stress predictions included in this report, corresponds to equation (115).

#### 4.2.5.3 Stress Predictions

Figures 110 to 114 show the comparison between the observed response and that calculated using the present theory. As may be seen, the predicted response is quite accurate in all cases considered, which include constant— and dual—rate tests, as well as a short-duration similitude loading.

#### 4.2.5.4 Material Characterization

As may be gathered from equation (115), the simplest version of this theory requires the knowledge of only two material-property functions, to wit:

- 1. The relaxation modulus, and
- 2. The damage shift function:

$$(a_{\eta}^{*}) = a_{\eta} = a_{\eta} \left[ \epsilon(t), \dot{\epsilon}(t) \right]$$
 (152)

which is determined in the following ways:

- a. From constant strain-rate tests, to correct the stress response during loading;
- b. From a relaxation test at a large strain level, to account for healing; and
- c. From a constant strain-rate cycle carried out to a large strain level, to more adequately reproduce the hysteretic behavior of the propellant.

The damage shift functions corresponding to UTP-19,360B are shown in Figures 115 to 119. The first three of these plots represent typical curves of  $a_{\eta}$  for low, intermediate and high strain-rate tests, while Figures 118 and 119 give the correction curves for relaxation and unloading, respectively.

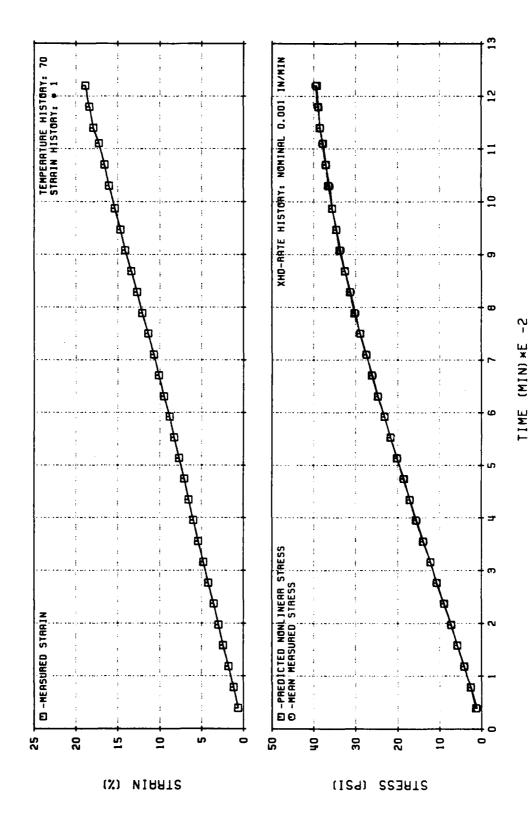
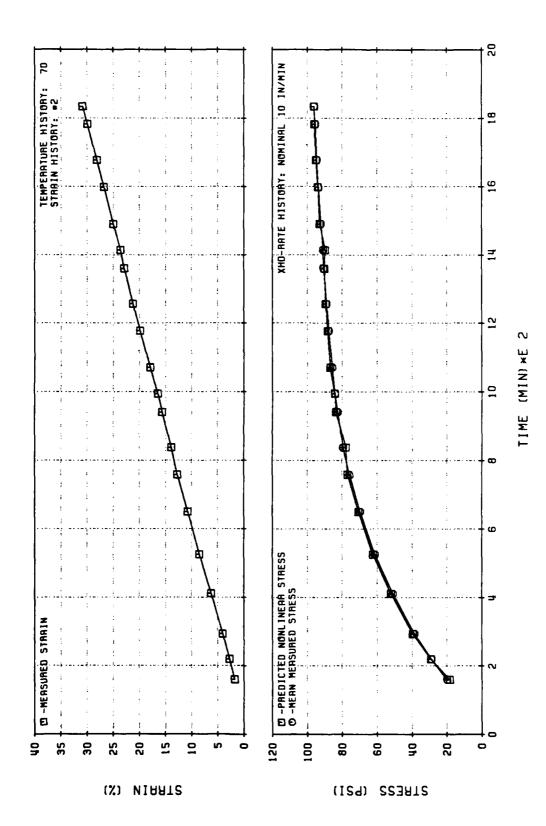
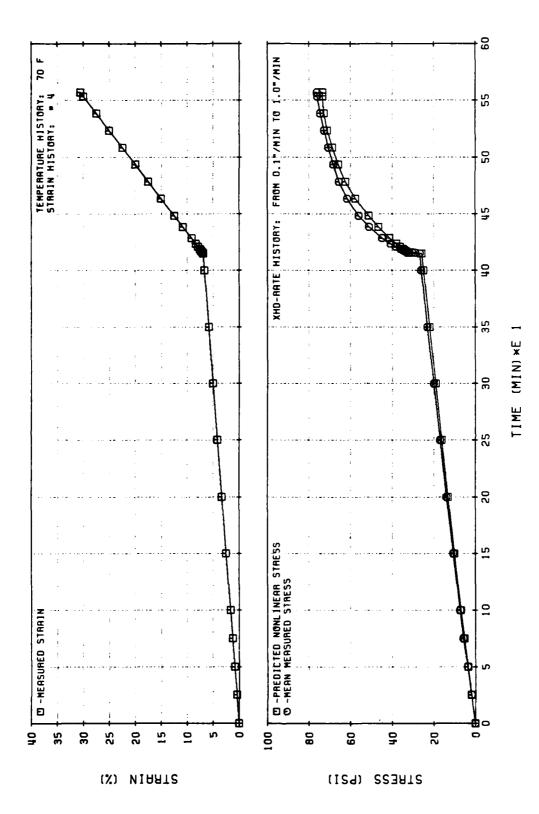


Figure 110. Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777

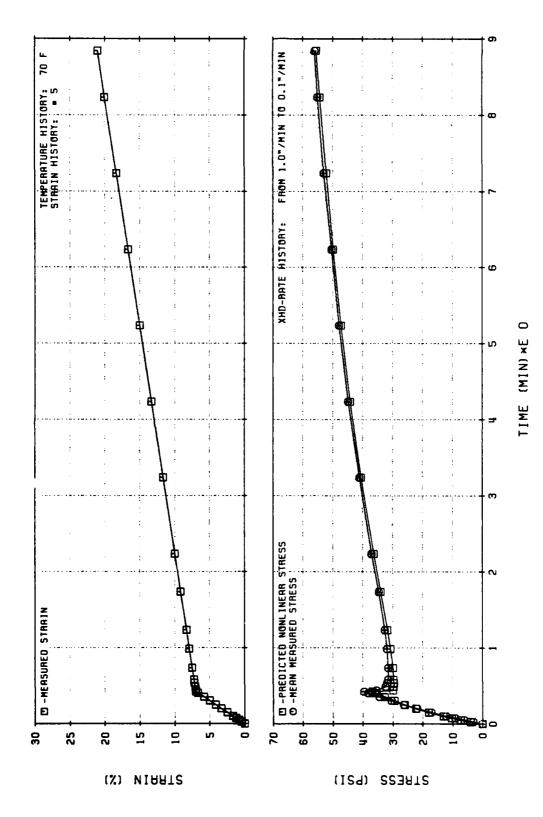


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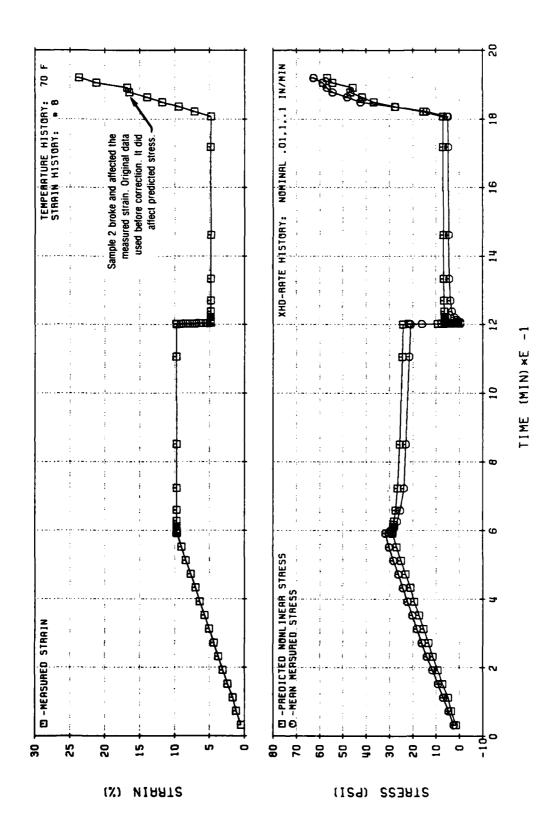
Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 Figure 111.



Nonlinear Viscoelastic Stress Predictions for Two-Rate Test (UTP-19,360B-400/1777) Figure 112.



Nonlinear Viscoelastic Stress Predictions for Two-Rate Test (UTP-19,360B-400/1777)



Nonlinear Viscoelastic Stress Predictions for Short Similitude Test (UTP-19,360B-400/1777) (W. L. Hufferd's Theory) Figure 114.

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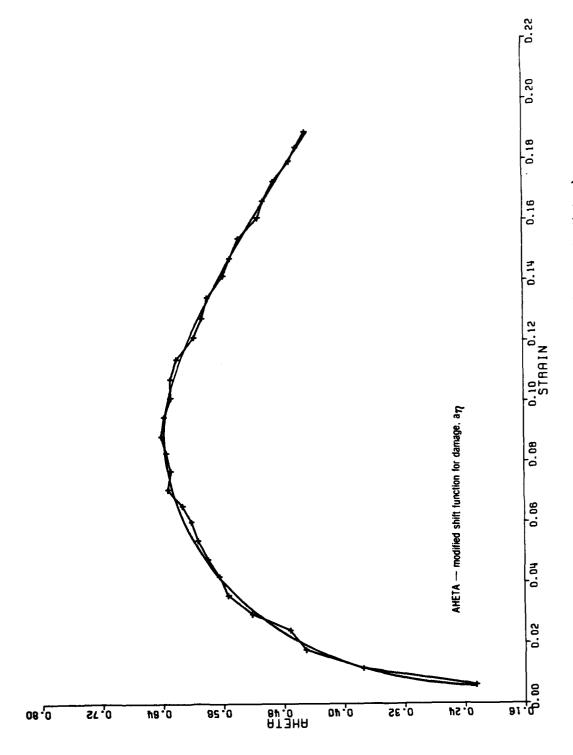


Figure 115. Constant Rate Test (0.001 in./min.)

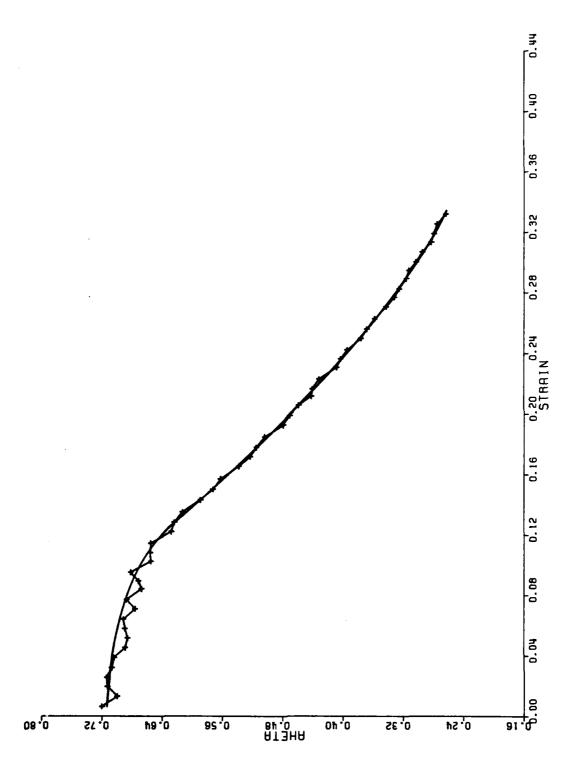
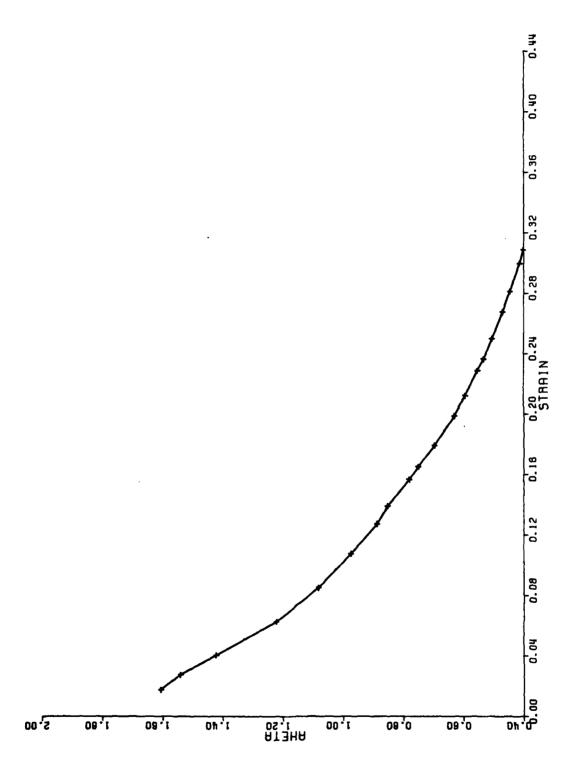


Figure 116. Constant Rate Test (0.1 in./min.)



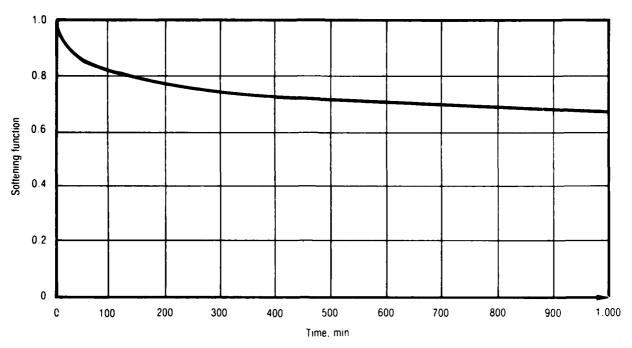


Figure 118. Softening Function During Relaxation

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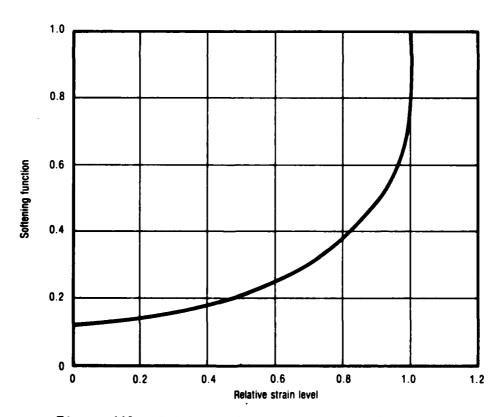


Figure 119. Softening Function During Unloading

28895

# 4.2.6 The Swanson Nonlinear Constitutive Law

## 4.2.6.1 Original Model

The framework for this theory was established by taking into account some typical behavior aspects of high-elongation propellants as indicated in Reference 9. The principal features considered were: (1) the usual viscoelastic dependence of the response on the strain rate, (2) the ability of the solid propellant to sustain large strains, (3) the marked deviation of the solid-propellant response from that associated with Linear Viscoelasticity, as evidenced by the large hysteresis exhibited under cyclic loading by many solid propellants, even at small strains, and (4) the dependence of the stress-strain response on superimposed pressure.

Although it is not essential to have done it this way, the capability of handling large strains was incorporated into the constitutive equations by using the cauchy-stress tensor  $(\sigma)$  as a measure of the state of stress at a point. Its conjugate, the left Cauchy-Green deformation tensor (B) was used as the measure of straining. The Cauchy stresses, defined in terms of force per unit deformed area, are also called "true" stresses. In a principal coordinate system, B takes on the diagonal form:

$$B = \begin{bmatrix} \lambda_1^2 & 0 & 0 \\ 0 & \lambda_2^2 & 0 \\ 0 & 0 & \lambda_3^2 \end{bmatrix}$$
 (153)

in which the  $\lambda_1$ 's are simply the extension ratios in the principal directions.

The remaining aspects of the observed response of solid propellant were modeled through the use of a softening function as a stress correction factor. The major constitutive assumption in this theory relates the second invariants of the deviatoric stress and deformation tensors through the equation:

$$\sqrt{II_{\sigma'}} = (f) (g)$$
 (154)

This separable form has been used previously (References 11 and 29) and is motivated by the fact that the constant strain rate tensile curves are roughly similar.

In equation (122), f is the following viscoelastic function:

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$$f = \int_{0}^{t} G(t - \tau) \frac{\partial \sqrt{IIB'}}{d\tau} d\tau \qquad (155)$$

with G being the relaxation modulus in shear, taken in this theory as one third the tensile relaxation modulus, and:

g = softening function  $\sigma'_{ij} = \sigma'_{ij} - (\sigma_{kk}/3) \, \delta_{ij} = \text{deviatoric stress tensor}$   $B'_{ij} = B_{ij} - (B_{kk}/3) \delta_{ij} = \text{deviatoric deformation tensor}$ (156)

II 
$$\alpha = \left\{ -\left[\alpha_{11} \alpha_{22} + \alpha_{22} \alpha_{33} + \alpha_{33} \alpha_{11}\right] + \alpha_{12}^{2} + \alpha_{23}^{2} + \alpha_{31}^{2} \right\}^{1/2}$$

Second invariant of tensor  $\alpha = \sigma$ , B

Now, g is a function of deformation and pressure (mean stress) and can be considered to be primarily a strain-softening function. It is defined as that function of the invariant  $\sqrt{II_B}$ ' that will force the viscoelastic Cauchy stress to coincide with the experimental results; thus, unloading hysteresis as well as the effects of pressure may be readily incorporated into this theory, simply by obtaining the corresponding forms of the softening function under such conditions.

The softening function corresponding to virgin loading is obtained by fitting the model to uniaxial tensile tests at constant cross-head speed. Under these conditions, the deviatoric stress invariant reduces to:

$$\sqrt{II_{\sigma'}} = \frac{\sigma_{11}}{\sqrt{3}} \tag{157}$$

where, again  $\sigma_{11}$  is Cauchy stress.

Assuming incompressibility:

$$\lambda_1 \lambda_2 \lambda_3 = 1 \tag{158}$$

and noting that:

$$\lambda_2 = \lambda_3 \tag{159}$$

the deformation invariant becomes

$$\sqrt{\text{II}_{B'}} = \frac{1}{\sqrt{3}} \left( \lambda_1^2 - \frac{1}{\lambda_1} \right)$$
 (160)

Taking the rate of change of this invariant as being approximately constant results in:

$$f = \sqrt{3} \lambda \int_{0}^{t} G(t - \tau) d\tau$$
 (161)

so that, from equations (155), (158), and (160), the following is obtained:

$$\frac{\sigma_{11}}{\sqrt{3}} = g \sqrt{3} \lambda^{*} \int_{0}^{t} G(t - \tau) d\tau$$
 (162)

from which the softening function, g, may be obtained. The assumption leading to equation (160), that the time-rate of change of  $\sqrt{II_B}$  is approximately constant, need be guarded against for conditions of changing strain rate. For example, as in dual-rate tests where viscoelasticity does not predict as fast a response to the rate change as is experimentally observed.

The modification to linear viscoelasticity necessary to accommodate this behavior is as follows. The response of the function f in equation (163) to a constant time rate of change of the deformation invariant is defined as  $f_c$ . It can be expressed as:

$$f_{C} = \sqrt{II B'} \int_{0}^{\xi} G_{rel} (\xi - \tau) d\tau$$
where  $\xi = \sqrt{II B'} / \sqrt{IIB'}$ 
(163)

The modification to the f function is done in an incremental manner through:

$$f_{\text{modified}} = f + \beta \left[ f_{\text{c}} - f \right] \sqrt{\text{II }_{\text{B}'}}$$
 (164)

and the following incremental relationship is used:

$$f = f + df_{mod} dt$$

$$t + dt dt$$
(165)

The parameter  $\beta$  governs the response of the f function under changing strain rates. As  $\beta>0$ , the response is analogous to linear viscoelasticity.

The algorithm developed by Herrman and Peterson (Reference 30) has been used to implement the calculation of the convolution integral for f. In brief, let the shear relaxation modulus be represented by a Prony series as

$$G(t) = \sum_{i=1}^{m} G_i e^{-\sigma_i t}$$
 (166)

Then let the f function at time  $t_n$  be given by:

$$f(t_n) = \int_{1}^{t_n} \sum_{i=0}^{t_n} G_i e^{-\alpha_i (t_n - \tau)} \frac{\partial \sqrt{II B'}}{\partial \tau} d\tau$$
 (167)

A recursion relation can be easily developed to compute  $f(t_n)$  (Reference 30). Let

$$f(t_n) = \sum_{i=1}^{m} I_{n,i}$$
 (168)

and

$$I_{n,i} = \int_{0}^{t_{n}} G_{i}e^{-\alpha_{i}(t_{n} - \tau)} \frac{\partial \sqrt{II_{B'}}}{\partial \tau} d\tau$$
 (169)

then

$$I_{n,i} = e^{-\alpha_i \Delta t_n} I_{n-1,i} + \sqrt{II_{B'_n}} \frac{G_i}{\alpha_i} \left[ 1 - e^{-\alpha_i \Delta t_n} \right]$$
 (170)

giving for the change in these terms

$$\Delta I_{n,i} = \sqrt{II B_n^i} \left\{ \frac{G_i}{\alpha_i} \left[ 1 - e^{-\alpha_i \Delta_{t_n}} \right] + I_{n-1,i} + \left[ e^{-\alpha_i t_n} - 1 \right] \right\}$$
 (171)

which is directly analogous to linear viscoelasticity. The modification proposed above can then be implemented as

$$(\Delta I_{n,i})_{\text{modified}} = \Delta I_{n,i} + \beta \left[ I_{\text{en},i} - I_{n,i} \right] \Delta \sqrt{II_{B'}}$$
 (172)

and the I terms can be incremented according to:

$$I_{n,i} = I_{n-1,i} + (\Delta I_{n,i})_{modified}$$

This has the effect of changing each term of the series so that it approaches the value it would have been if it was always at the new strain rate. Note again that (as discussed in Reference 30) varying temperatures can be incorporated into the time scale as usual.

Unloading tests required a further refinement of the model. For lack of more detailed information, the parameter  $\beta$  may be taken as zero for unloading states (i.e., states in which  $\sqrt{II_B}$  is decreasing). The large amount of hysteresis seen in load-unload cycles is then modeled in part by the hysteresis inherent in linear viscoelasticity primarily through the g function. This is accommplished by giving g a different value when the deformation invariant  $\sqrt{II_B}$  is less than its maximum previously achieved during the loading history. If  $\sqrt{II_B}$  hax is the current maximum value, the function:

$$g = g(\sqrt{II_{B'_{max}}}) \left\{ 1 - C_1 \left[ 1 - \sqrt{II_{B'_{max}}} / \sqrt{II_{B'_{max}}} \right] \right\}$$
 (174)

provides plasticity-like behavior.

The behavior of the g function for unloading and reloading conditions is illustrated in Figure 120 (taken from Reference 9).

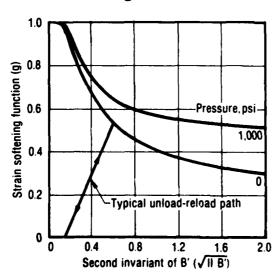


Figure 120. Effect of Deformation and Pressure on the Strain Softening Function 22059

The Swanson approach (Reference 9) was used with an only limited degree of success to predict the stress response of TP-H1011 and UTP-19,360B under several strain histories.

In the case of TP-H1011, the errors in the predictions were believed to be due to uncertainties in the value of the changing-rate coefficient  $(\beta)$ . There was no data available to determine  $\beta$  directly for this propellant.

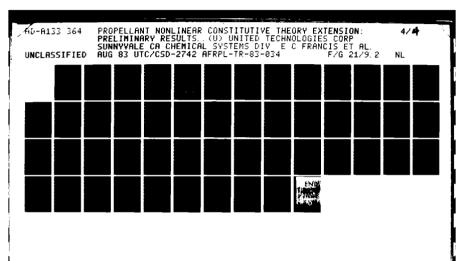
It was possible to characterize UTP-19,360B in a complete fashion. The corresponding predictions were not any better than those obtained for TP-H1011. This led to changing the law as discussed below.

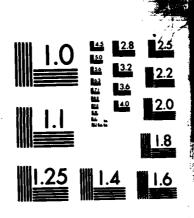
#### 4.2.6.2 Current Model

Analysis of the stress predictions, carried out for UTP-19,360B with the original Swanson theory, revealed the importance of several inadequacies and oversimplifications listed below.

- 1. The softening function (g) should depend not only on the strain and pressure but also on the strain rate.
- 2. The softening function, as defined by equations (155) and (163), should be different for unloading than for reloading.
- 3. The softening function for unloading or for reloading should never become zero for conditions of tensile straining only. A zero value could occur with the softening function defined by equation (174).
- 4. The healing process observed during relaxation in solid propellants like UTP-19,360B was not taken into consideration by the original Swanson theory.
- 5. The reverse-recovery observed in solid propellants during relaxation or rest periods that follow an unloading process, only poorly modeled by classical viscoelasticity, is not considered in the approach by Swanson.
- 6. The changing-rate coefficient ( $\beta$ ) is more a mathematical device than it is a material property. If the softening function is made to depend on the strain rate then  $\beta$  need not be used.
- 7. The use of a softening function as a stress correction factor eliminates the need of using the Cauchy stress  $(\sigma)$  and the nonlinear measure of stretching (B).

All these observations were incorporated into the original stress-strain law but the general form of the corresponding equations remained the same, namely;





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$$\sigma_{11} = \sqrt{3} \quad (g) \quad \int_{0}^{t} G \left( s_{t} - s_{\tau} \right) \partial \sqrt{II_{B'}} d\tau$$
 (175)

valid for one-dimensional loading, with:

$$\mathbf{s_t} - \mathbf{s_7} = \int_{\tau}^{t} \frac{d\xi}{\mathbf{a_T} \left[ \pi(\xi) \right]}$$
 (176)

representing temperature-reduced time; and where the time-temperature shift function was taken in the power-law form:

$$\mathbf{a_T} = \left(\frac{\mathbf{T_R} - \mathbf{T_a}}{\mathbf{T} - \mathbf{T_a}}\right)^{\mathbf{m}} \tag{177}$$

in which  $T_R$  is the shift reference temperature,  $T_a$  and m are material parameters, and T is the current temperature.

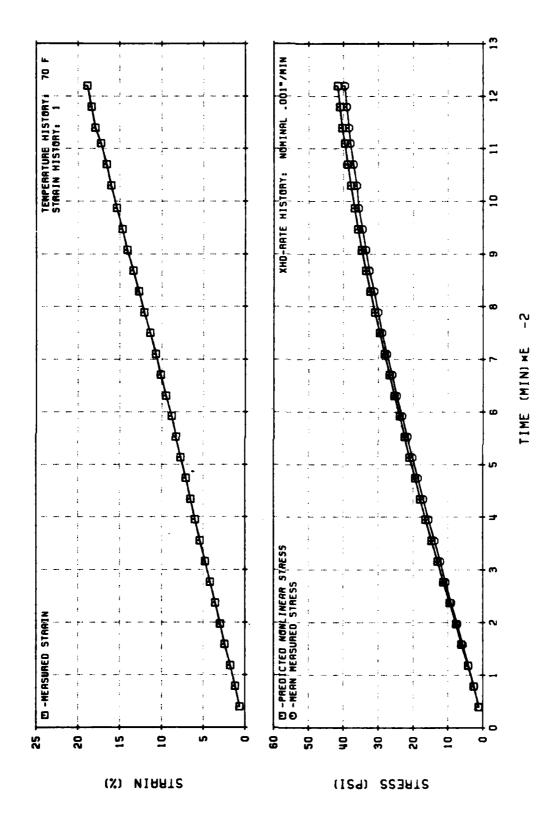
The modified version of the Swanson theory was most successfully used to predict the response of UTP-19,360B, as explained next.

# 4.2.6.3 Stress Predictions

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The degree of accuracy of the predictions made with the current version of the Swanson approach may be realized by examining Figures 121 through 130. The first two figures correspond to the lowest and highest constant-rate tests available. Figure 123 shows the predictions corresponding to a saw-tooth test at constant rate and increasing peak strains. Figures 124 and 125 present the results for the dual-rate tests while Figures 126 and 128 pertain to the long- and short-duration similitude tests, respectively. Figure 127 shows the three-step relaxation test. Finally, Figures 129 and 130 show the results obtained for constant rate tests at 123°F and 40°F.

(Text continued on page 285.)



Nonlinear Viscoelastic Stress Predictions for Constant-Rate Test (UTP-19,360B-400/1777) Figure 121.

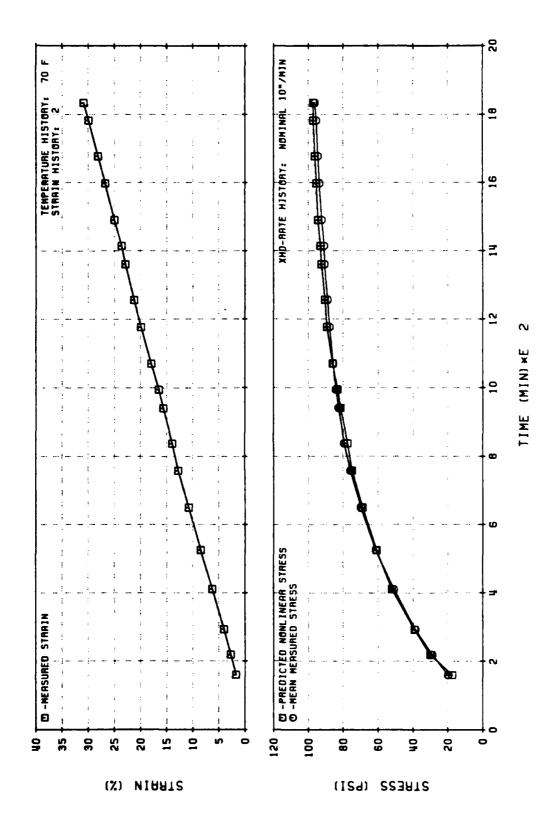
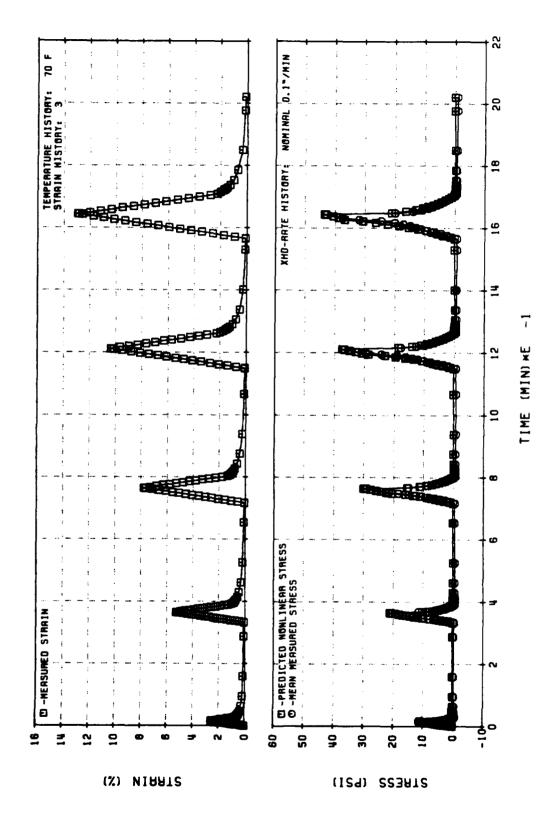


Figure 122. Nonlinear Viscoelastic Stress Predictions for Constant-Rate Test (UTP-19,360B-400/1777)



Nonlinear Viscoelastic Stress Predictions for Saw-Tooth Test (UTP-19,360B-400/1777) Figure 123.

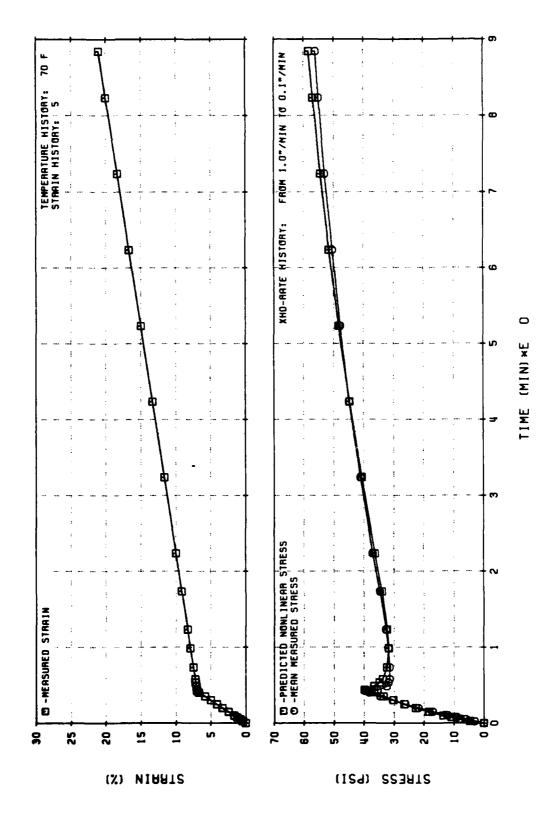
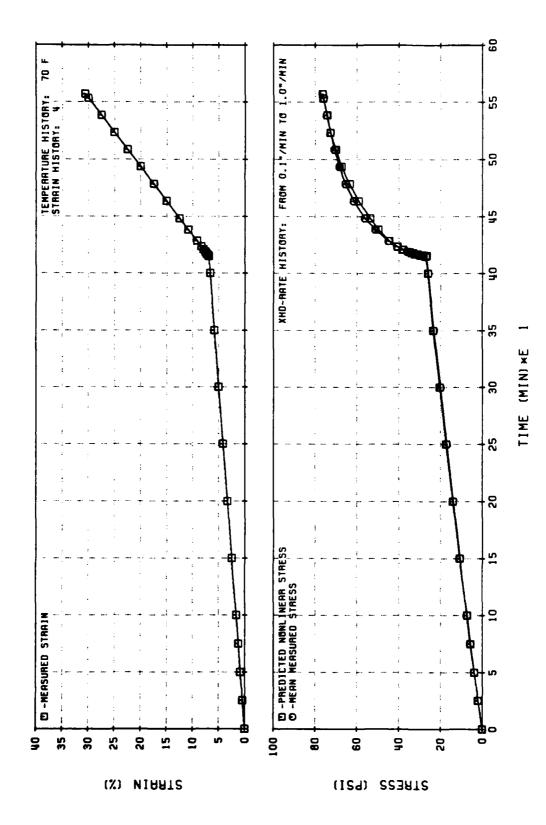


Figure 124. Nonlinear Viscoelastic Stress Predictions for Two-Rate Test (UTP-19,360B-400/1777)



Nonlinear Viscoelastic Stress Predictions for Two-Rate Test (UTP-19,360B-400/1777) Figure 125.

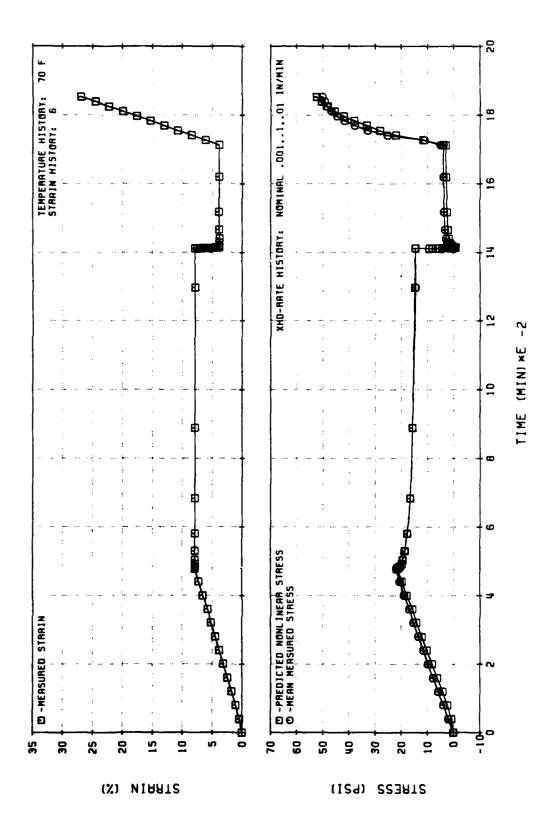
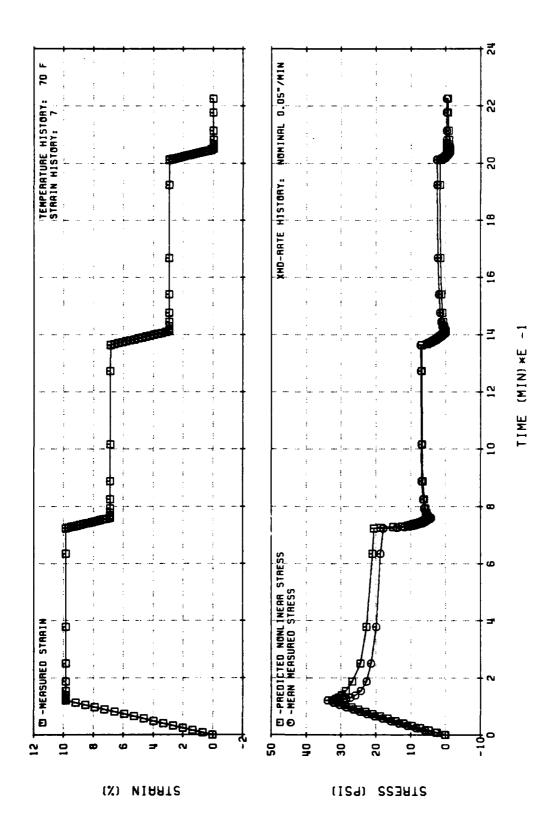


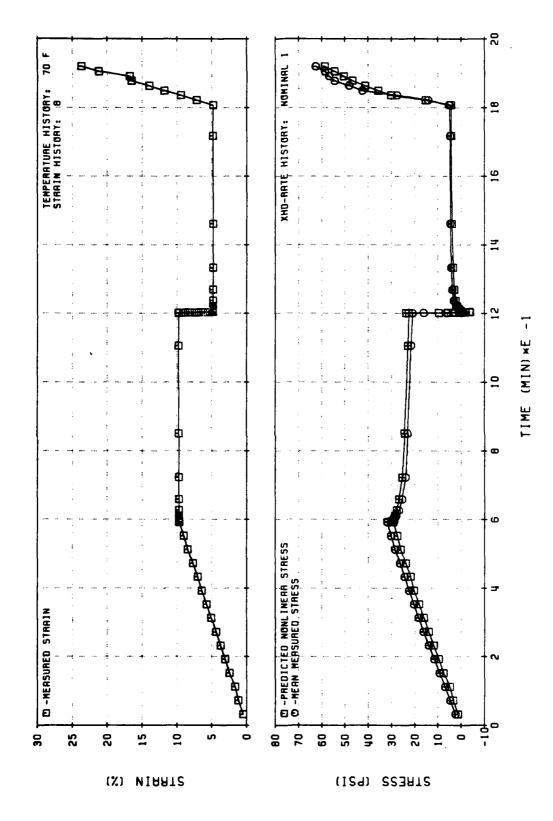
Figure 126. Nonlinear Viscoelastic Stress Predictions for Long Similitude Test (UTP-19,360B-400/1777)



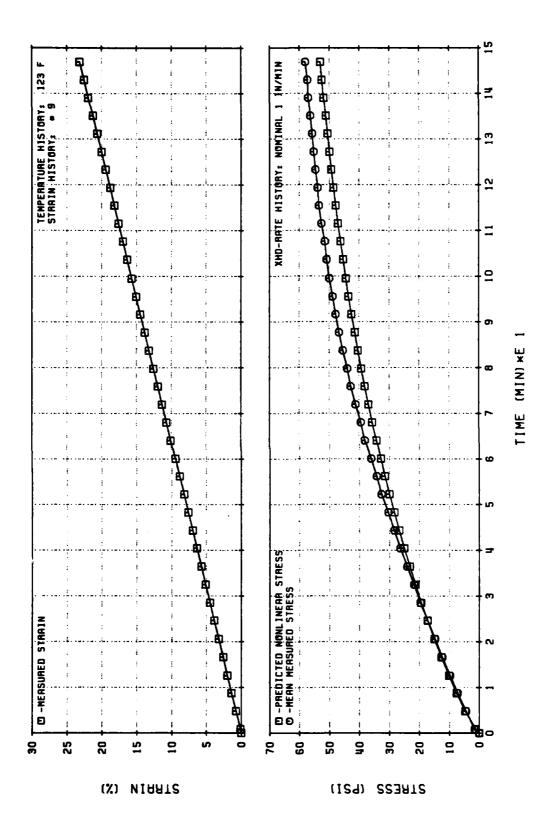
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Figure 127. Nonlinear Viscoelastic Stress Predictions for 3-Step Relaxation (UTP-19,360B-400/1777)



Nonlinear Viscoelastic Stress Predictions for Short Similitude Test (UTP-19,3608-400/1777) Figure 128.



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Figure 129. Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 at 123 F

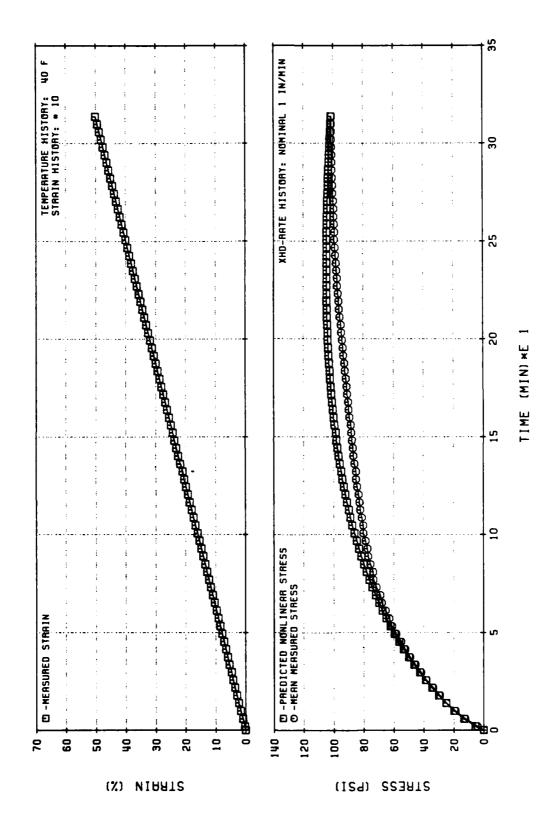


Figure 130. Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 at 40 F

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They testify to the fact that the time-temperature superimposition principal may be used without sacrificing more accuracy than is already lost in fitting equation (143) to the very limited time-temperature shift data.

# 4.2.6.4 Material Characterization

According to this theory, only the following listed properties are needed to characterize a solid propellant completely:

1. The relaxation function, G, defined as:

$$G(t) = \frac{E_{rel}(t)}{3}$$
 (178)

where  $E_{rel}(t)$  is the linear viscoelastic relaxation modulus.

- 2. The softening function, g, defined and obtained as follows:
  - a. For loading conditions:

$$\mathbf{g}_{\mathbf{L}} = \mathbf{g} \ (\boldsymbol{\epsilon}, \ \boldsymbol{\dot{\epsilon}}) \tag{179}$$

and it is obtained from a sequence of constant-rate tests at at least three different rates that span the range expected in the applications.

b. For unloading conditions:

$$g_u = g(\epsilon/\epsilon_{max})$$
 (180)

where  $\epsilon_{\max}$  represents the maximum strain previously achieved during the loading history. The g is determined from the unloading portion of a loading-unloading cycle carried up to an intermediate strain level.

# c. For relaxation conditions:

$$g_r = g_r (t - t_0)$$
(181)

in which  $t_0$  is the time at which the relaxation process begins and g is evaluated from a relaxation test at an intermediate strain level.

In addition, for relaxation after partial unloading or during rest periods starting at  $t = t_0$ :

$$\hat{g_r} = \frac{1}{g_r (t - t_0)} \tag{182}$$

Also, the stress-correction function for reloading is taken as a linear function of the relative strain. It is a straight line from the point where reloading starts to the point of maximum loading over the past history.

#### 4.2.6.5 Addendum to Swanson Theory

# Three-Dimensional Version of the Model

The (general) constitutive assumption used to relate the deviatoric components of the stress and deformation tensors, takes the following form:

$$\frac{\sigma_{ij}^{\prime}}{\sqrt{II_{\sigma'}}} = \frac{B_{ij}^{\prime}}{\sqrt{II_{B'}}}; i, j = 1, 2, 3$$
(183)

together with:

$$\sqrt{\text{II}_{g'}} \approx (g) (f)$$
 (184)

or, equivalently:

$$\sqrt{II_{\sigma'}} = (g) \int_{0}^{t} G(t - \tau) \frac{\partial \sqrt{II_{B'}}}{\partial \tau} d\tau$$
 (185)

where:

 $\sigma_{ij}^{i}$  = i-j component of the deviatoric Cauchy stress tensor  $B_{ij}^{i}$  = i-j component of the deviatoric Left Cauchy-Green deformation tensor  $II_{\sigma}$ ,  $II_{B}^{i}$ , = second invariants of the deviatoric stress and deformation tensors

with:

$$\sigma_{ij}^{i} = \sigma_{ij} + \frac{1}{3} (\sigma_{11} + \sigma_{22} + \sigma_{33}) \delta_{ij}; i, j = 1, 2, 3$$
(186)

$$II_{\sigma'} = \left\{ -\left[\sigma_{11} \ \sigma_{22}' + \sigma_{11}' \ \sigma_{33}' + \sigma_{22}' \ \sigma_{33}'\right] + (\sigma_{12}')^2 + (\sigma_{13}')^2 + (\sigma_{23}')^2 \right\} (187)$$

and similarly for Bij and IIB'; and in which:

$$\delta_{ij} = \begin{cases} 1 & \text{for } i = j \\ 0 & \text{for } i \neq j \end{cases}$$
 (188)

also:

g = softening function that depends primarily on the strain level, the strain rate, and the applied pressure.

and:

$$G(t) \stackrel{\text{def}}{=} E(t)/3 \tag{189}$$

where E(t) represents the tensile relaxation modulus at a small strain.

According to the constitutive assumptions (183) and (185), the distortional behavior of the material is completely characterized through the softening function, g, and the relaxation function, G, which may be evaluated from one-dimensional tests, as explained in the previous section. Indeed, the stress-strain relations set forth in equation, (183) and (185), reduce, as they should, to those employed in the one-dimensional version of the model.

To complete the theory, an assumption is still needed about volumetric behavior; and although time-dependent bulk response may be important in some applications, the elastic relation:

$$\frac{1}{3}(\sigma_{11} + \sigma_{22} + \sigma_{33}) = K(|\lambda_1 \lambda_2 \lambda_3| - 1)$$
 (190)

may be employed; in which K is the bulk modulus and the  $\lambda_{i}$ 's are the stretch ratios.

For an incompressible material (and solid propellants are nearly incompressible):

$$\lambda_1 \lambda_2 \lambda_3 = 1 \tag{191}$$

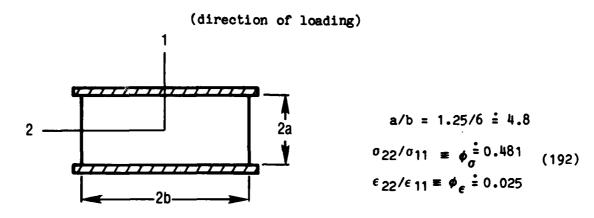
so that equation (190) breaks down, and the stress tensor has to be considered a function of the mean pressure  $(\sigma_{11} + \sigma_{22} + \sigma_{33})/3$ , as well as of the deformation tensor, leading eventually, to a stress-strain law of the form given in equation (183).

# Application of the Model to Two-Dimensional Problems

In order to use the stress-strain law presented in the foregoing section, one must have available the deformation tensor at each point of the continuum where the stresses are desired. This solution in terms of deformation may be arrived at numerically through, say, finite elements, or, analytically.

The accuracy with which the present constitutive theory may predict the two-dimensional response of solid propellants may be seen in Figures 131 to

136, which correspond to constant strain-rate tests of strip-biaxial samples of UTP-19,360B. The first three figures belong to tests performed at a nominal crosshead displacement rate of 0.02 in./min. at 40°F, 70°F, and 120°F, respectively; while Figures 134 to 136 show the results for a crosshead displacement rate of 0.2 in./min. at the same low, intermediate and high temperatures of 40°, 70°, and 120°F. The plotted data refer to the direction of applied loading, which is also the direction of maximum principal stress (and strain). The geometry of the strip-biaxial sample used is as presented in the following sketch.



The stress- and strain-axiality factors,  $\phi_{\sigma}$  and  $\phi_{\epsilon}$ , were taken from Reference (31), and are valid at the center of the sample for small strains only.

The constitutive relations given in (183) and (185) yield:

$$\sigma_{ij}' = (g) \sqrt{\frac{B_{ij}'}{II_{B'}}} \int_{0}^{t} G(t - \tau) \frac{\partial \sqrt{II_{B'}}}{\partial \tau} d\tau$$
 (193)

where:

$$\frac{\partial \sqrt{II_B}}{\partial t} = \frac{\partial \sqrt{II_B}}{\partial B'_{ij}} \frac{\partial B'_{ij}}{\partial t}; i, j = 1, 2, 3$$
 (194)

with summation implied over repeated indices.

Now, under conditions of plane stress of an incompressible material, and along the principal directions, one has:

$$\begin{bmatrix} \sigma_{1j}' \end{bmatrix} = \frac{1}{3} \begin{bmatrix} (2\sigma_1 - \sigma_2) & 0 & 0 & 0 \\ 0 & (2\sigma_2 - \sigma_1) & 0 \\ 0 & -(\sigma_1 + \sigma_2) \end{bmatrix}$$
(195)

$$\begin{bmatrix} B_{1j} \end{bmatrix} = \frac{1}{3} \begin{bmatrix} (2\lambda_1^2 - \lambda_2^2 - \lambda_3^2 - \lambda_3^2 - \lambda_1^2 & 0 \\ 0 & (2\lambda_2^2 - \lambda_3^2 - \lambda_1^2 & 0 \\ 0 & (2\lambda_3^2 - \lambda_1^2 - \lambda_2^2) \end{bmatrix}$$
(196)

II B' = 
$$-B_{11} B_{22} - B_{11} B_{33} - B_{22} B_{33}$$
 (197)

To evaluate (194), we first write it in unabridged notation, noting that in this case, if  $i \neq j$  then  $B'_{ij} = 0$ ; thus:

$$\frac{\partial \sqrt{II_B}}{\partial t} = \frac{\partial \sqrt{II_B}}{\partial B_{11}'} \frac{\partial B_{11}'}{\partial t} + \frac{\partial \sqrt{II_B}}{\partial B_{22}'} \frac{\partial B_{22}'}{\partial t} + \frac{\partial \sqrt{II_B}}{\partial B_{33}'} \frac{\partial B_{33}'}{\partial t}$$
(198)

and using (197):

$$\frac{\partial \overline{II_{B'}}}{\partial t} = \frac{1}{2\sqrt{II_{B'}}} \left[ (-B'_{22} + B_{33}) \frac{\partial B'_{1j}}{\partial t} + (-B'_{33} - B'_{11}) \frac{\partial B'_{22}}{\partial t} + (-B'_{11} - B'_{22}) \frac{\partial B'_{33}}{\partial t} \right]$$
(199)

where, from (196) one has, for instance, that:

$$\frac{\partial B_{11}'}{\partial t} = \frac{2}{3} \left( 2\lambda_1 \frac{d\lambda_1}{dt} - \lambda_2 \frac{d\lambda_2}{dt} - \lambda_3 \frac{d\lambda_3}{dt} \right)$$
 (200)

and similarly for the derivatives of  $B_{22}^{\dagger}$  and  $B_{33}^{\dagger}$ .

In the previous derivations, the stretch ratios are computed as:

$$\lambda_1 = 1 + \epsilon_1 \quad (t)$$

$$\lambda_2 = 1 + \epsilon_2 \quad (t) = 1 + \phi_{\epsilon} \epsilon_1 \quad (t)$$

$$\lambda_3 = 1/(\lambda_1 \lambda_2) \quad (201)$$

in which  $\epsilon_1(t)$  is the strain history imposed on the sample along coordinate 1, and the last expression of (201) follows from the incompressibility condition, equation (191).

Hence, the first component-equation of the constitutive relation (193) yields:

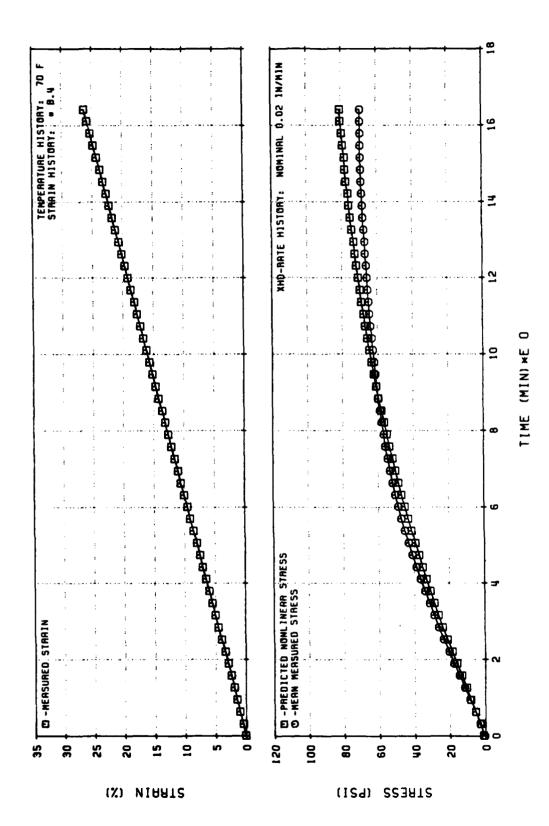
$$\sigma_{11} = \frac{1}{3} (2\sigma_1 - \sigma_2) = (g) \frac{B_{11}'}{\sqrt{II_B'}} \int_0^t G(t - \tau) \frac{\partial \sqrt{II_B}}{\partial \tau} d\tau$$

or, in view of (192):

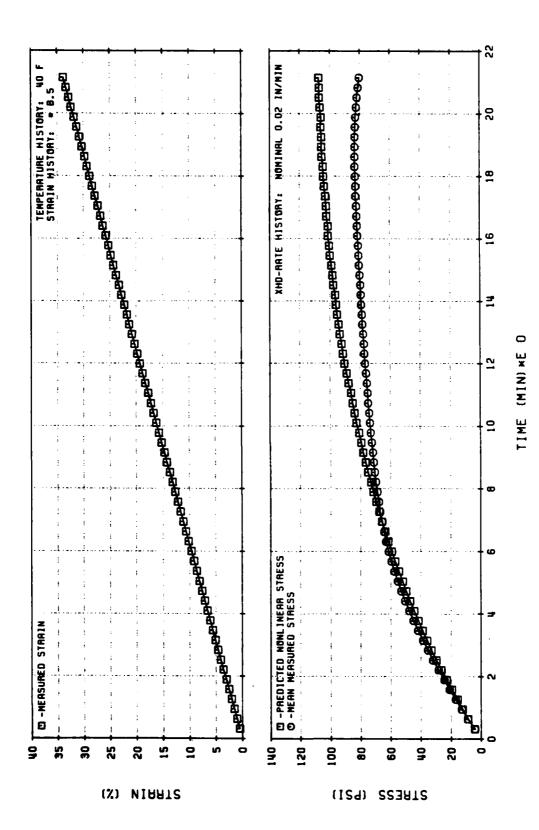
$$\sigma_1 \frac{(2 - \phi_0)}{3} = (g) \frac{B_{11}'}{\sqrt{II_{B'}}} \int_0^t G(t - \tau) \frac{\partial^{V_{II_{B'}}}}{\partial \tau} d\tau$$

and finally: 
$$\sigma_1 = \frac{3}{(2 - \phi_\sigma)} (g) \frac{B_{11}'}{\sqrt{II_{B'}}} \int_0^t G(t - \tau) \frac{\partial \sqrt{II_{B'}}}{\partial \tau} d\tau$$

(202)

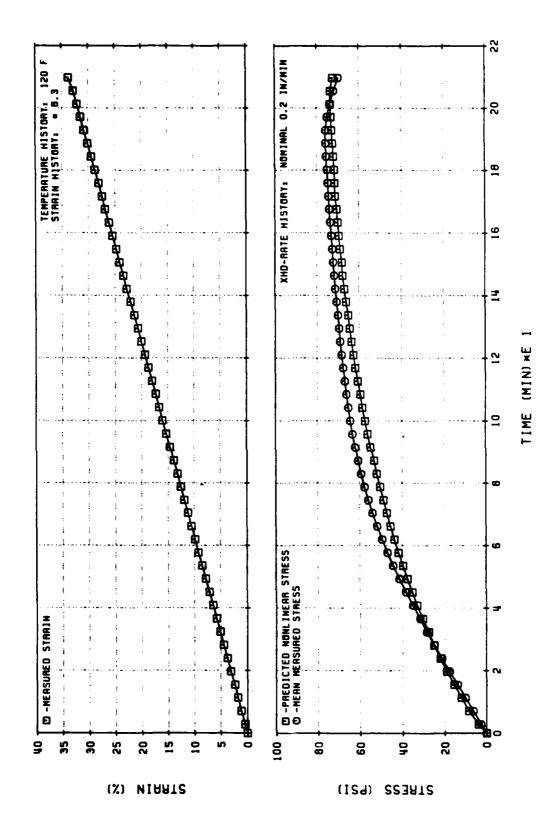


Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 (Biaxial Sample) Hercules Theory Figure 131.

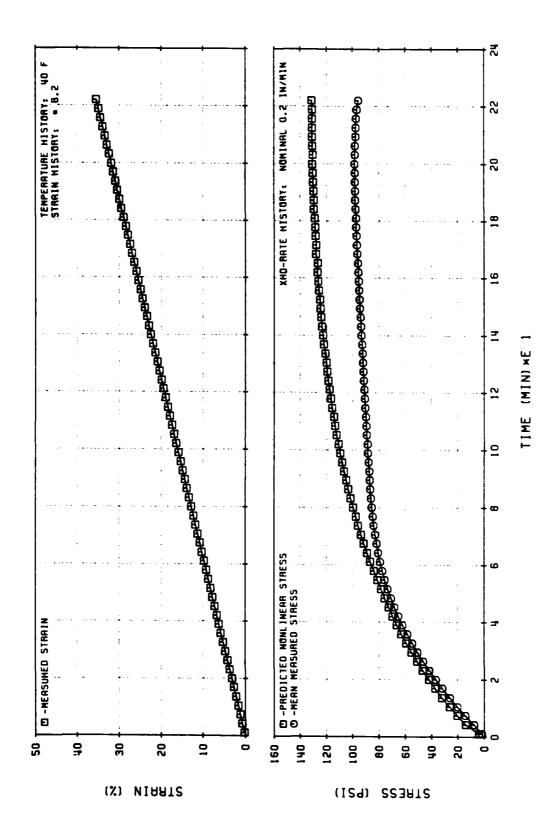


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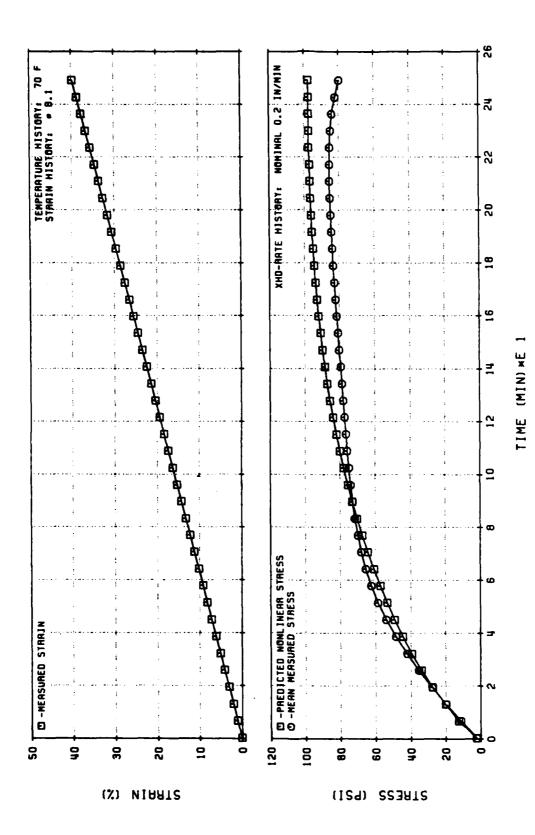
Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 (Biaxial Sample) Hercules Theory Figure 132.



Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 (Biaxial Sample) Hercules Theory Figure 133.



Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 (Biaxial Sample) Hercules Theory Figure 134.



Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 (Biaxial Sample) Hercules Theory Figure 135.

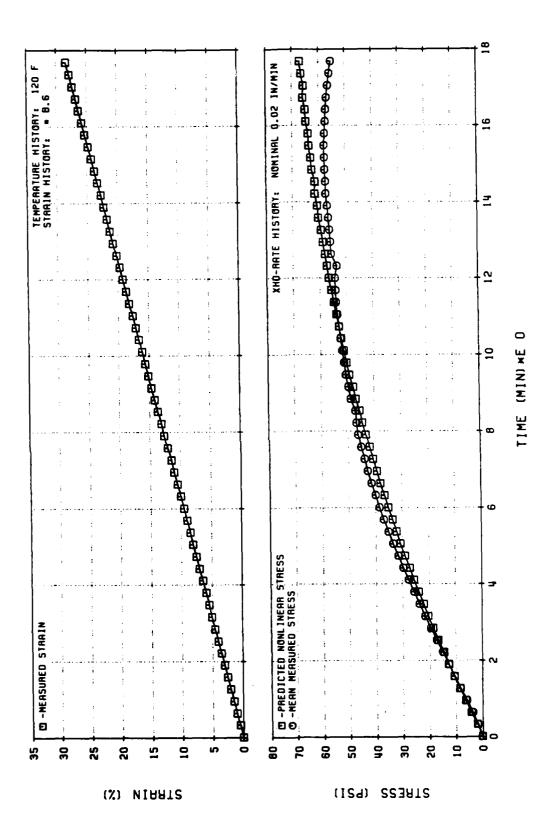


Figure 136. Nonlinear Viscoelastic Stress Predictions for UTP-19,360B-400/1777 (Biaxial Sample) Hercules Theory

Using equations (192), (196), and (199) to (202), we obtained the response of the biaxial sample in the direction of the applied loading. The plots included in this report, show the engineering stress:

$$\sigma_{1} = \sigma_{1}/\lambda_{1} \tag{203}$$

rather than the Cauchy stress,  $\sigma$ .

4.2.7 M. Quinlan's Theory of Materials with Variable Bonding

# 4.2.7.1 Original Model

In developing a mathematical framework for his stress-strain law, Quinlan, in Reference 4, reasoned that since propellants consist of minute rigid particles embedded in a polymer matrix, such materials would respond to a deformation process with a change in the amount of species to species bonding. He thus proposed to correct the deficiencies of fading-memory type theories by introducing a correction term that accounted for the changes in the state of bonding that are induced by a deformation process. His constitutive model then took the form:

$$\sigma = \sigma_{\hat{\mathbf{f}}} + \sigma_{\hat{\mathbf{b}}} \tag{204}$$

in which:

 $\sigma$  = current stress

 $\sigma_f$  = fading-memory type stress

 $\sigma_{\rm b}$  = stress correction due to change in the state of bonding

Motivated to some extent by reaction-rate theory, Quinlan modeled the evolution of the bonding state through the following ordinary differential equation:

$$\dot{\pi} = \alpha \left\{ \dot{\phi} - \mu \left[ 1 - e^{\nu} \left( \phi - \pi \right) \right] \right\}$$
 (205)

subject to the initial condition:

$$\pi\left(0\right)=1\tag{206}$$

in which  $\pi$  represents the state of bonding;  $\alpha$ ,  $\mu$  and  $\nu$  are material parameters, and

$$\phi = 1 + \epsilon \tag{207}$$

is the stretch ratio; with  $\epsilon$ , the strain.

The unique solution of (204) may be readily obtained for piecewise linear stretch histories, as reported in Reference 4.

Taking a linear viscoelastic relation for  $\sigma_f$ , and considering the stress correction term,  $\sigma_b$ , as proportional to the state of bonding, Quinlan arrived at the following stress-strain law:

$$\sigma(t) = \int_{0}^{t} G(t - \tau) \dot{\phi}(\tau) d\tau + B_{0} \dot{\pi}(t)$$
 (208)

with:

$$G(t) = E_0 t^{-n} \tag{209}$$

$$\dot{\pi} (t) = \alpha \left\{ \dot{\phi} - \mu \left[ 1 - e^{\nu} (\phi - \pi) \right] \right\}$$
 (210)

and

$$\pi(t=0)=1 \tag{211}$$

The six parameters:  $E_0$ , n,  $B_0$ ,  $\alpha$ ,  $\mu$  and  $\nu$ , needed in this theory to characterize a propellant, may be obtained by fitting the model to the observed response of the material when subjected to a saw-tooth strain history that has increasing peak strains and sufficiently long rest periods between cycles. Alternatively, the studies reported in the literature on the effects of employing different data bases for characterization, show that the test history should primarily include the maximum expected strain level, the expected range of strain rates, as well as rest and relaxation periods.

The present model was used to predict the response of TP-H1011 under several loading histories; and it reproduced, somewhat accurately, the general trend of solid propellant behavior.

In an attempt to include healing effects, the underlying assumption for the evolution of damage were revised, as explained next.

# 4.2.7.2 Current Model

The theory developed by Quinlan has undergone several changes; mainly in the expression defining the evolution of damage. Thus, the form:

$$\sigma(t) = \int_{-\infty}^{\infty} H(t - \tau) \dot{\phi}(\tau) d\tau + C \dot{\pi}(t)$$
 (212)

has been retained in all versions of this theory, but the rate mechanism underlying damage:

$$\dot{\pi} = P (\pi, \phi, \dot{\phi})$$

for which:

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$$\pi(\phi = 1) = 1$$
 (214)

was assumed to contain a neutral rate,  $\zeta$ , at which damage remains constant, i.e.,

$$P(\pi, \phi, \zeta) = 0 \tag{215}$$

This concept then allowed introducing the notion that at rates higher than 5, bond breakage would take place, while at rates smaller than the neutral rate, bond formation would ensue. The details of the derivations leading to the specific form of equation (213) are presented next.

Equation (215) for the neutral rate, \$\( \), may be rewritten as:

$$\zeta = Q (\pi, \phi)$$

$$\zeta = 0 \text{ if and only if } \pi = \phi$$
(216)

which, upon expansion in Taylor series, becomes:

$$\zeta = Q(\pi, \pi - \phi) \Big|_{\phi = 0} + \frac{\partial Q(\pi, \pi - \phi)}{\partial (\pi - \phi)} \Big|_{\phi = 0} (\pi - \phi) + Q(|\pi - \phi|^2)$$

The first term on the right-hand side of this equation vanishes by virtue of (216), so that, neglecting the higher order terms, and defining:

$$\mu \stackrel{\text{def}}{=} \frac{\partial Q (\pi, \pi - \phi)}{\partial (\pi - \phi)} \Big|_{\phi = 0}$$
(217)

leads to the following first-order expression for the neutral rate:

$$\delta = \mu(\pi - \phi) \tag{218}$$

where for bond breakage:

$$\dot{\phi} = \zeta > 0 \tag{219}$$

while for bond formation:

$$\phi - $<0$$
 (220)

In addition, equation (213) may be cast in the following form:

$$\dot{\pi} = P (\pi, \phi, \dot{\phi}) = R(s, u)$$
 (221)

where:

$$s = s (\pi, \phi)$$

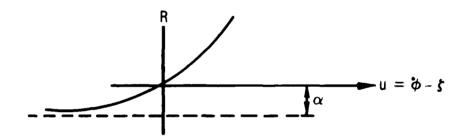
(222)

and:

$$\mathbf{u} = \mathbf{\dot{\phi}} - \mathbf{\dot{\zeta}} \tag{223}$$

with S given by (218).

Now, under the assumption that the process of bond formation is slower than that of bond breakage, the function R defining the evolution of damage must be of the form shown in the sketch below, in which the parameter  $\alpha$  would represent the maximum rate of bond formation.



Hence, R may be defined through the following differential equation:

$$\frac{dR(s, u)}{du} = \nu \left[R(s, u) - \alpha\right]$$
 (224)

where  $\alpha$  and  $\nu$  are positive constants.

Integration of (224) yields:

$$R(s, u) = \alpha(e^{\nu u} - 1)$$
 (225)

Finally, putting (218), (223), and (225) into (221), results in:

$$\dot{\pi} = \alpha \left\{ \exp \left[ \nu \dot{\phi} + \mu \nu (\phi - \pi) \right] - 1 \right\}$$
 (226)

Equations (212) and (226) subject to (214) were used by Quinlan in several ways to characterize the response of UTP-19,360B. One such stress-strain law took the following form:

$$\sigma(t) = E_0 \epsilon(t) + \left[E_1 + E_2 \epsilon(t)\right] \int_0^t (t - \tau)^{-n} \dot{\epsilon}(\tau) d\tau + C\dot{\pi}$$
 (227)

in which Eo,  $E_1$ ,  $E_2$ , n and C are constants and  $\epsilon$  is the strain. Although some aspects of propellant behavior were better modeled than with the original version of the theory, others were not, and further revisions were necessary. In the latest version of his constitutive law, Quinlan used a Prony series to represent the relaxation function and changed strain for stretch in the original equation of evolution for damage, so that, in summary, the current model looks as follows:

$$\sigma(t) = \sigma_{V}(t) + \sigma_{b}(t)$$

$$\sigma_{V}(t) = G_{e} \epsilon(t) + \int_{0}^{t} G(t - \tau) \dot{\epsilon}(\tau) d\tau$$

$$G(t) = \sum_{i=1}^{n} G_{i} e^{-\tau_{i}t}$$

$$i = 1$$

$$\sigma_{b}(t) = Be^{\gamma(\epsilon - \pi)} \epsilon(t)$$

$$\dot{\pi}(t) = \alpha \left\{ e^{\nu(\epsilon - \pi)} - 1 \right\} ; \pi(o) = 0$$
(228)

This final version of the theory has been employed to characterize UTP-19,360B, but has not been used by Quinlan to predict the response of the propellant under any loading history other than the characterization test.

## 4.3 CONCLUSIONS

The best nonlinear constitutive theories available for modeling solid propellant response were selected. Each of these theories was able to model some simple constant strain rate test behavior during the early part of the program, but generally gave poor correlation with the long time complex load histories characteristic of rocket motors.

A broad nonlinear data base was developed with two solid propellants. This data was used both at CSD and at some of the University subcontractors facilities. This data base can be used for evaluating any future nonlinear constitutive theories. All data was collected on an HP-9825 computer and is available in a format for a VAX computer system.

This data was used to evaluate and further develop the nonlinear theories for solid propellant response. Each of the theories has been extensively modified and now fits both simple and complex uniaxial load histories.

These theories will be further developed for multiaxial load histories in the last phase of the program.

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## Appendix A

#### MULTISTATION AUTOMATED DATA REDUCTION

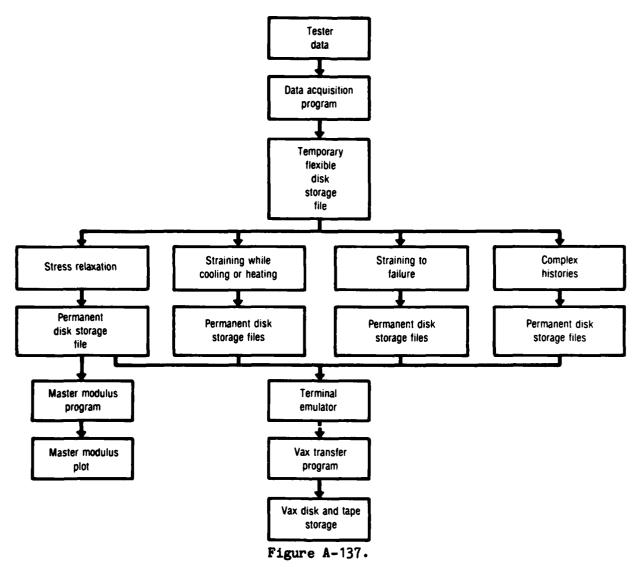
#### INTRODUCTION

Automated handling of multistation tester data is accomplished with a system of interactive programs on the HP 9825 desk top computer (Figure A-137). These programs include data acquisition, stress relaxation-master modulus, straining while cooling or heating, straining to failure, and complex histories. The acquisition of data and test control are functions of the data acquisition program which supplies data to the data reduction programs. The reduction programs reduce and output data for a particular type of test history. In addition, terminal emulating software for the HP 9825 provide a data link for the transfer of data to the VAX-11 mainframe computer. This makes the data directly available to the nonlinear constitutive theory programs.

#### SYSTEM INSTRUMENT CONFIGURATION

The multistation data acquisition instruments are configured to provide load, crosshead position, temperature and elapsed time data to the data acquisition program. The system consists of a Hewlett Packard 9825 desk top computer, 3455 digital voltmeter, 3495 scanner, 98035 programmable clock and 9885 flexable disk drive (Figure A-138).

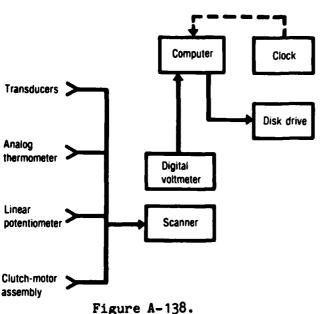
The HP 9825 and data acquisition program act as the system controller. The controller processes incoming test data and crosshead information and responds by sending instructions to other instruments in the system over an HP-IB interface. Output signals from the tester's load transducers, linear potentiometer, and analog thermometers are input into the scanner's programmable relay cards. The scanner's relays under command of the controller can be opened independently to route output data signals individually to the digital voltmeter where they are digitized and read by the program. Crosshead control information from output lines connected to the tester's motor-clutch assembly, is supplied to the program through the scanner in the same way as the data output signals. These



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signals enable the controller to react to changes in crosshead movement and direction without relying on operator intervention. The programmable clock connects directly to one of the computer's I/O ports. It provides the program with elapsed time data and a program interupt capability for controlling the rate at which data is taken. The flexible disk drive provides a mass storage medium where data is stored during testing for latter access by one of the data reduction programs.

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### **PROGRAMS**

Data Acquisition:

As previously stated, the data acquisition program is used to collect data from the multistation tester.

Operation of the program involves steps to initialize the program, calibrate the system, and collect and store test data.

Initialization of the program is accomplished by operator entered information used to identify the particular test and define samples

being tested. In response to prompts from the computer display, the operator inputs descriptions on test material, crosshead rates, strain levels, and temperature levels of the test history. Data input on the test samples include their number, gage length, and individual cross sectional areas, along with their channel locations. In addition, the operator enters pairs of crosshead rates and delta strains for each test interval used to compute sampling rates. The operator also determines how data is taken during relaxation cycles by specifying whether sampling is to be done in a fixed or log time interval.

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Calibration of the system is done by an operator-interactive procedure to determine the tester's transducers and potentiometer sensitivites. This involves the operator queuing the program to take readings from the transducers at differing load conditions. By comparing the change in output signals for a known change in load, the lb/volt sensitivity of each transducer may be determined. Similarly, by moving the linear potentiometer probe a known distance its in./volt sensitivity is determined. The analog thermometers are not calibrated at the time of testing. These units output a 10 mv/°F representation of the test chamber and internal sample temperature. Calibration on them is done periodically by the CSD electronics laboratory. For short time and isothermal tests, calibration is done once before testing begins. For tests

lasting over a long time period, a second calibration is done when testing is complete to enable compensation for drift in the tester's electronics. Since the load transducers are temperature sensitive, for thermal tests two calibrations must be performed at differing temperatures to determine the change in sensitivity per degree change in temperature.

When calibration is complete, the program stops operation until testing is ready to begin. On a cue from the operator, zero load and position data is taken and the system's instruments programmed to their initial conditions. The clock interupt period is set for a sampling rate determined from the initial crosshead rate and delta strain information. Scanner relays are also arranged to monitor the tester's break input voltage.

The program monitors the brake voltage until detecting the brake has disengaged which signifies crosshead motion. The clock's counter and interupt units are then started. Interupt signals are output by the clock at the set sampling rate until changed by the program at the end of the straining interval.

When interupt instructions are received from the clock, program operation branches to a data collection subroutine. The voltmeter and scanner are set to read output signals from each of the transducer, potentiometer and thermometer channels. Fifty milliseconds are required to read each channel. Elasped time, read from the clock counter, is taken as the mean time over which the data set was read. The test data is retained in a memory buffer until transferred to a disk storage file.

The program continues to monitor the crosshead break and clock information channels throughout the test. When a change in crosshead motion is detected, the clock interupt is stopped. From the test description data corresponding to the test interval, a sampling rate is determined and the clock reset. For log time interval samplings, the clock interupt unit is stopped after each reading and the time doubled.

Up to 600 data sets may be retained in the computer's memory at one time. Data is transferred to the disk either between data set samplings, if time allows, or when testing is complete.

Data Reduction Programs:

The reduction programs reduce and output multistation data pertinent to particular types of test histories. The programs are stress relaxation-master modulus, straining while cooling or heating, straining to failure and complex histories. A description of the strain and temperature histories relevant to each is listed in Table A-35.

Test identification, calibration, load, sample extension, thermal, and time data are supplied to the programs from the acquisition data files or entered directly by the operator. In addition, relaxation cathetometer strain measurements and the thermal expansion coefficient may be optionally entered.

Each program reduces stress, strain, modulus, temperature and elapsed time data when applicable. The method by which each is determined depends on the test history and amount of information available to the program.

Calibration sensitivity (S) of the transducer and potentiometer are determined in general by

where load is the transducer calibration weight or potentiometer probe displacement. For tests where multiple calibrations were performed, the sensitivities at time t are corrected for electrical drift and thermal variations with the linear relationships

$$S(t) = S_{initial} + \frac{(S_{final} - S_{initial})}{X_{final} - X_{initial}} X (t)$$

TABLE A-35

	Program	Test Strain History	Output
1.	Stress relaxation		Tabular - time, modulus Graphic - modulus vs time
2.	Straining while cooling or heating		Tabular - time, strain, temperature, stress Graphic - stress vs time and temperature
3.	Straining to failure		Test History - Tabular - time, strain and stress  Graphic - stress vs time and straining
			Mech Properties Tabular - initial modulus, maximum stress and strain - corrected stress and strain - rupture strain
4.	Complex histories	Combination of strain- ing, relaxation and temperature intervals	Tabular - time, strain temperature and stress Graphic - stress vs time and strain and temperature vs time

where X is time for isothermal and temperature for nonisothermal tests.

Temperature corrections are not made on the potentiometer sensitivity since it is located outside the environmental test chamber.

Zero load outputs (20) for the transducers are also corrected for electrical drift and thermal variation by the same method as the sensitivities.

Corrections on the zero position output of the potentiometer cannot be made since the crosshead can't be accurately returned to its initial position.

Sample stresses ( $\sigma$ ) at time t are calculated by  $\sigma(t) = \left[ \text{transducer output - ZO (t)} \right] S(t)/\text{cross-sectional area}$  Sample strain ( $\epsilon$ ) at time t is determined by  $\epsilon(t) = \left[ \text{pot output - ZO} \right] S(t)/\text{gage length}$ 

For nonisothermal tests a strain correction may be applied using the thermal expansion coefficient  $(\alpha)$ . In this case, the total sample strain becomes

$$\epsilon(t)$$
 = mechanical strain +  $T(t)$  -  $T_{initial}$   $\alpha/gage$  length

An additional correction for effective gage length may be made using cathetometer measurements of actual sample strains. The correction factor is determined as the ratio of the mean intervals in the test history. For histories where multiple cathetometer measurements were made, the correction factors are linearized to measured strain between them.

Relaxation and secant modulus (E) at time t is determined by

$$E(t) = \sigma(t) \left[ 1 + \epsilon(t) / \epsilon(t) \right]$$

where  $\epsilon(t)$  is held constant over relaxation test intervals.

Temperature is reduced from analog thermometer readings by converting the millivolt output to volts.

Elapsed time is calculated as the difference between when the data was taken and when initial loading occurred  $(t_0)$  since loading times may vary from sample to sample,  $t_0$  is approximated by the time of initial straining.

Once data is reduced, a tabular and graphic summary of the test is output by each program. A description of the outputs is listed in Table A-35. To retain data for future reference and reuse, identification, calibration and raw test data are stored on permanent diskette data files.

Terminal Emulator

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The program is used to transfer data between the HP 9825 and VAX-11, desk top and mainframe computers.

A link is created between the computer types utilizing the VAX-11's dialines and an RS-232 interface which connects the 9825 to an acoustic coupler. The emulator software then supplies the capability of using the 9825 as an intelligent terminal through which data may be read from the flexible disks and sent over phone lines to the VAX.

Data transfer is accomplished with a VAX program which reads data sent from the terminal and retains it in storage files for access by the nonlinear theory programs.

# SYMBOLS

Ac	microcrack growth rate shift factor
AF	temperature dependent material function
Aij	expansion coefficients of bulk stress in terms of octahedral strains
An	constant
Ao	initial area
A _T	temperature shift factor $(a_T)$
аη	(AHETA) damage related shift function
a	half sample width
a _F	softening function
^a k	constant
a _{np}	expansion coefficients of correction modulus
A1, A2, A3, A4, A6, Ai	constants
2a	biaxial sample width
_	
В	Cauchy-green deformation tensor
В	bulk modulus and a constant
B*	deviatoric deformation tensor
2b	biaxial sample height (gage length)
C	softening function
CSD	Chemical Systems Division
c*	rehealing parameter
c	constant
c, c ₁ , c ₂ , c ₁	constants

d constant

 $D_m$ ,  $D_5$ ,  $D_6$ ,  $D_7$  constants

E modulus

E_e equilibrium modulus

E_R reference modulus and normalized coefficient for

modulus

E_R relaxation modulus

E(t),  $E_{rel}(t)$  linear viscoelastic relaxation modulus

e product of F and virgin response function g

eij deviatoric strain tensor

 $E(\xi)$  linear viscoelastic modulus

F force

F(t) constant rate modulus

 $F(\xi, \epsilon_m)$  damage curve at  $\epsilon_m$  damage level

f material parameter

(f) deformation function

f_c constant time rate of change of deformation invariant

f(t) viscoelastic type function

^OF degrees Fahrenheit

F damage function or softening function

F strain magnification factor

G shear modulus

G. L. gage length

G_c corrected modulus

G_r relaxation modulus

G_{rel} shear relaxation modulus

G(t) shear relaxation modulus

g virgin response function

g() function of

(g) strain softening function

h function of damage in kinetic equation of evolution

HTPB Hydroxy Terminated Polybutadiene

I_d volume dilatation

In contribution to stress at time tn

 $I\gamma$  octahedral shear strain

 $\parallel$  I_{$\gamma$}  $\parallel$  pi L_p norm

J creep function

JANNAF Joint Army Navy NASA Air Force

K constant

 $K_{\mathrm{I}}$  stress intensity factor

K_X rehealing parameter

k constant

 $\mathbf{L}_{\mathbf{X}}$  constant

M constant

 $M_x$ ,  $M_2$ ,  $M_4$  constants

e constant

millivolts

m, m, m_i material parameters

number of cycles constant terms of equation under summation hydrostatic pressure **PBAN** Polybutadiene Acrylonitrile constant P2, P4 constants P₁₅ used to normalize Y3 to 1 terms of equation under summation constant RH relative humidity S virgin stress and damage parameter constant Se damage parameter constant So S_r certain measure of damage St-ST temperature shifted time  $S_{\mathbf{x}}$ constant damage parameter temperature T peak stress time material property and shift temperature temperature at t = 0To TR reference temperature

t time time time to failure under constant load UTP United Technologies Propellant  $= \epsilon/\epsilon_{\rm m}$ x X width position from center of biaxial sheet root of Ya X_r y ± e_{iñ} Y1, Y1, Y2, Y3 functions related to damage coefficient of thermal expansion α constant β constant parameter γ shear strain ΔL change in length δ11 Kronecker delta strain principal strain € 1 lateral strain € 2 € 11, €22, €33 principal strains € 0 pseudo strain maximum strain € m, € max € (t) strain at time t strain of unfilled polymer  $\epsilon_{\mathbf{u}}$ strain due to mechanical stress  $\epsilon_{\sigma}$ 

η	compressibility ( $\eta = 0$ is incompressibility)
7(t)	related to damage function,
λ	extension ratio $(1 + \epsilon)$
λ	healing correction factor
λ	softening function
λ	width to height ratio
ξ	reduced time
ξ	ratio of position to half sample
ξ	width (x/a)
<b>π</b> α	second invariant of tension $(\alpha = \sigma, B)$
Σ	summation
σ	engineering stress or stress
σ	Cauchy-stress tensor
σ _B	bulk stress
σ _e	stress correction
σ¢j	deviatoric stress
· σ¹	deviatoric stress
σχ	linear viscoelastic stress
σo	constant stress
σ _{kk}	bulk stress
$\sigma_{f r}$	linear viscoelastic stress
o(t)	stress at time t
σ _f (t)	fading memory stress
7	reduced time
7	shear stress

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shifted time  $\phi \qquad \qquad \text{function of loading}$   $\Psi_{n} \qquad \qquad \text{nth component of stress correction}$   $\omega \qquad \qquad \text{normalized damage function}$   $\text{II}_{\alpha} \qquad \qquad \text{second invariant of tensor } (\alpha = \sigma, \ B)$   $\sqrt{\text{II}_{B}}, \qquad \qquad \text{second invariant of deformation tensor}$   $\sqrt{\text{II}_{\sigma}}, \qquad \qquad \text{second invariant of deviatoric stress}$   $|\cdot| \qquad \qquad \text{denotes absolute value}$ 

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